

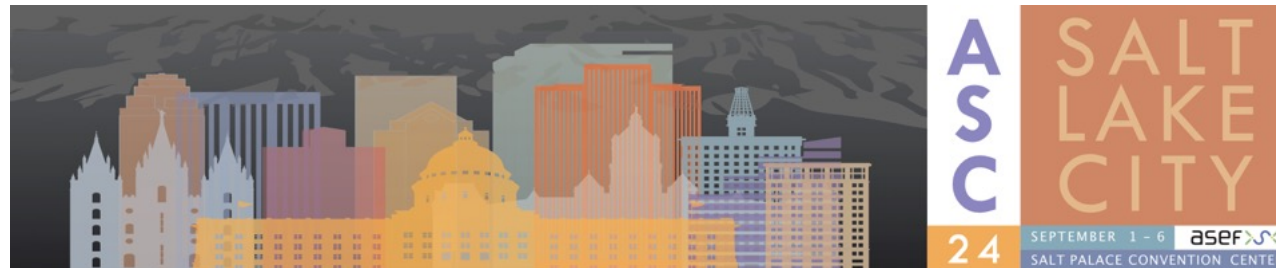
ASC2024 Salt Lake City

Status of Iron Based Superconductors: characteristics and relevant properties for applications

Kazumasa Iida, Nihon University



2024.9.3, Salt Lake City



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S. Eley

Overview

1. Iron-based superconductors (IBSs)

- Physical properties

2. Tuning of the superconducting properties

- SC transition temperature (strain, monolayer, intercalation, EDLT)
- Grain boundary, GB
- Critical current density (natural defects, APC, thermodynamic approach)

3. Progress Toward applications

- Use of IBS wires and bulks in magnets, and perspective

Discover of Iron based superconductors (IBSs)

First Fe-based superconductor in 2006

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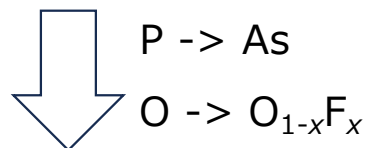
Published on Web 07/15/2006

Iron-Based Layered Superconductor: LaOFeP

Yoichi Kamihara,[†] Hidenori Hiramatsu,[†] Masahiro Hirano,^{†,‡} Ryuto Kawamura,[§] Hiroshi Yanagi,[§]
Toshio Kamiya,^{†,§} and Hideo Hosono^{*,†,‡}

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4259 Nagatsuta, Midori-ku, Yokohama 226-8503, Japan, Frontier Collaborative Research Center, Tokyo Institute of
Technology, Mail Box S2-13, 4259 Nagatsuta, Midori-ku, Yokohama 226-8503, Japan, and Materials and Structures
Laboratory, Tokyo Institute of Technology, Mail Box R3-4, 4259 Nagatsuta, Yokohama 226-8503, Japan

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Published on Web 02/23/2008

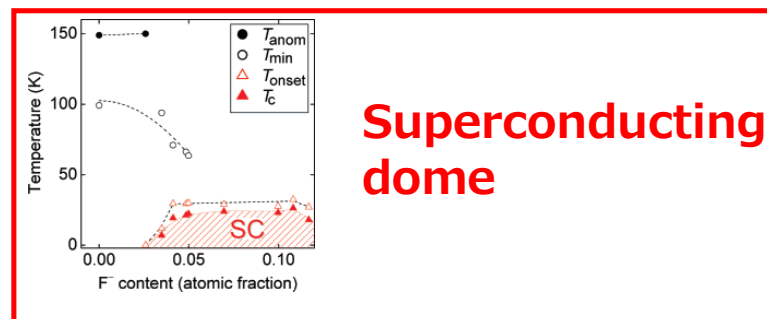
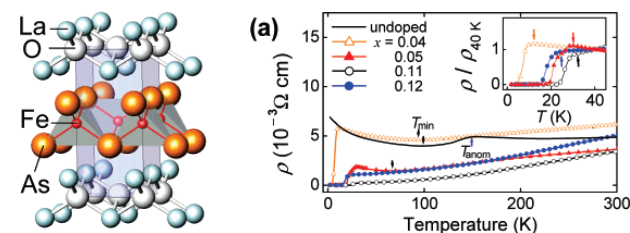
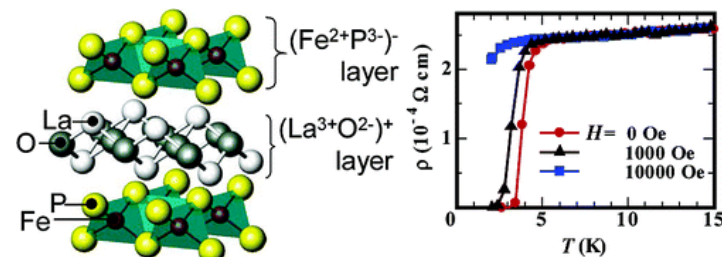
In 2008

Iron-Based Layered Superconductor La[O_{1-x}F_x]FeAs (x = 0.05–0.12) with T_c = 26 K

Yoichi Kamihara,^{*,†} Takumi Watanabe,[‡] Masahiro Hirano,^{†,§} and Hideo Hosono^{†,‡,§}

ERATO-SORST, JST, Frontier Research Center, Tokyo Institute of Technology, Mail Box S2-13, Materials and
Structures Laboratory, Tokyo Institute of Technology, Mail Box R3-1, and Frontier Research Center, Tokyo Institute
of Technology, Mail Box S2-13, 4259 Nagatsuta, Midori-ku, Yokohama 226-8503, Japan

Received January 9, 2008; E-mail: hosono@mssl.titech.ac.jp

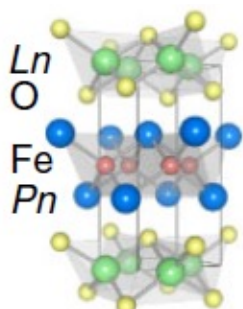


Superconducting
dome

Iron based superconductors (IBSs): FeAs or FeCh tetrahedron

1111

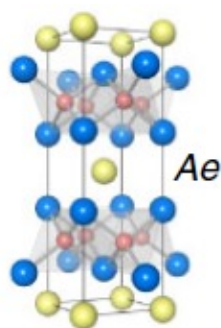
$T_c=55$ K



$LnFePnO$
 $AeFePnF$
 ($Ln=La, Ce$
 $Ae=Ca, Sr$
 $Pn=P, As$)

122

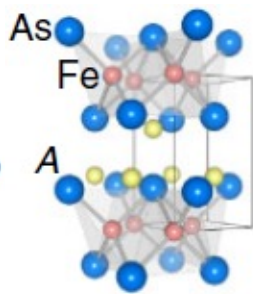
$T_c=38$ K



$AeFe_2Pn_2$
 ($Ae=Ca, Sr,$
 Ba, K, Eu)

111

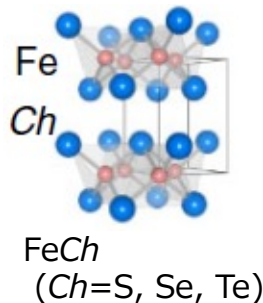
$T_c=18$ K



$AFeAs$
 ($A=Li, Na$)

11

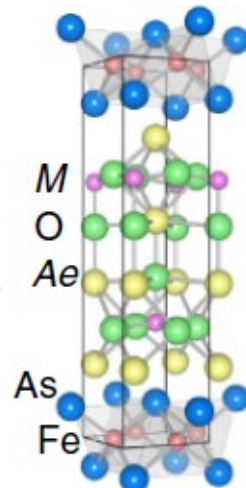
$T_c=14$ K



$FeCh$
 ($Ch=S, Se, Te$)

21113

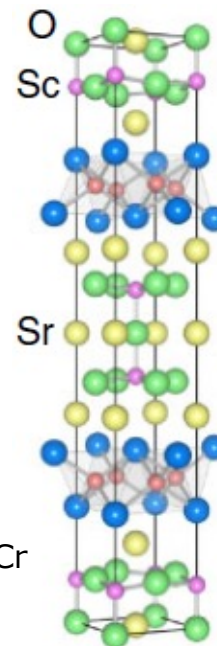
$T_c=37$ K



$Ae_2MFePnO_3$
 ($M=Sc, Ti, V, Cr$
 $Ae=Ca, Sr$
 $Pn=P, As$)

32225

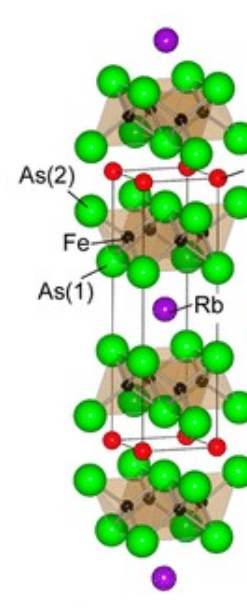
$T_c=45$ K



$Sr_3Sc_2Fe_2As_2O_5$

1144*

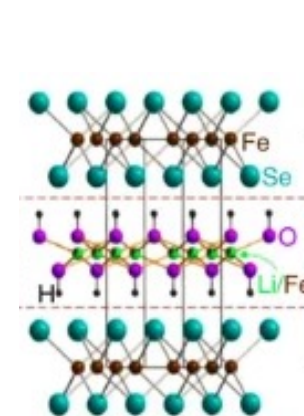
$T_c=36$ K



$CaRbFe_4As_4$

11111

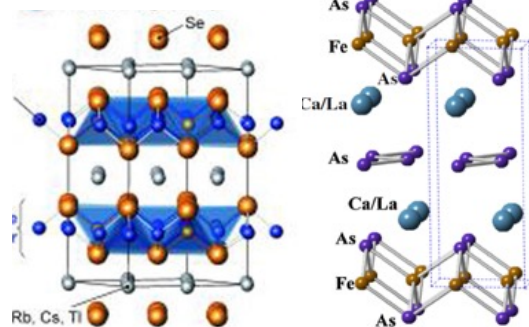
$T_c=41$ K



245

112

$T_c=34$ K



$AeFeAs_2$
 ($Ae=Ca, Sr, Ba$)

FeAs and FeCh
 tetrahedron:
 common structure

*A. Iyo et al., *J. Am. Chem. Soc.* **138** 3410 (2016).

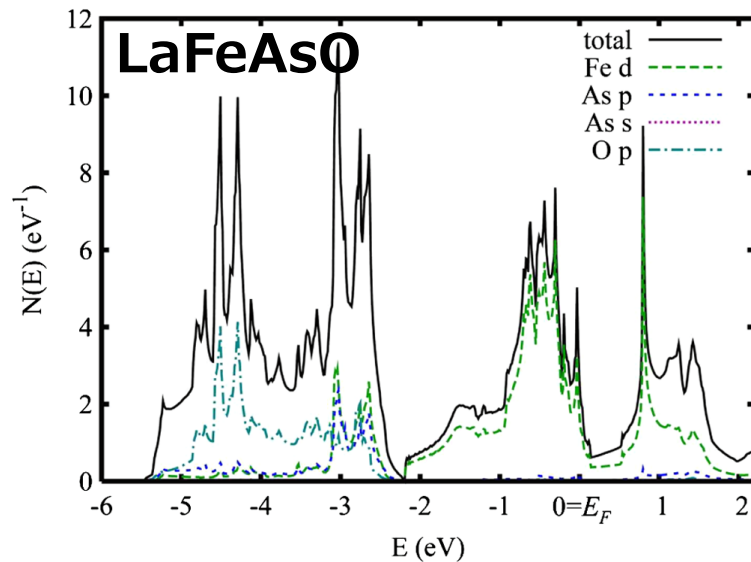
K. Tanabe and H. Hosono, *Jpn. J. Appl. Phys.* **51** 010005 (2012).

Electronic structure: Multiband superconductors

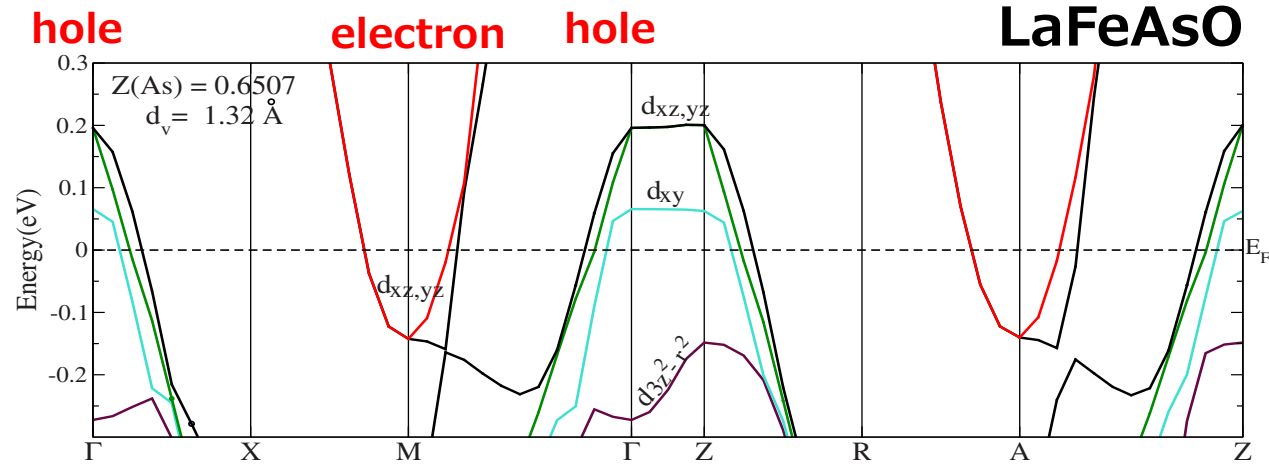
Fe3d orbitals play an important role for superconductivity

Fe 3d orbitals dominate the total DOS around E_F

**Five Fe 3d bands across E_F
→ Multiband superconductors**



D. J. Singh and H.-M. Du, *Phys. Rev. Lett.* **100** 237003 (2008).

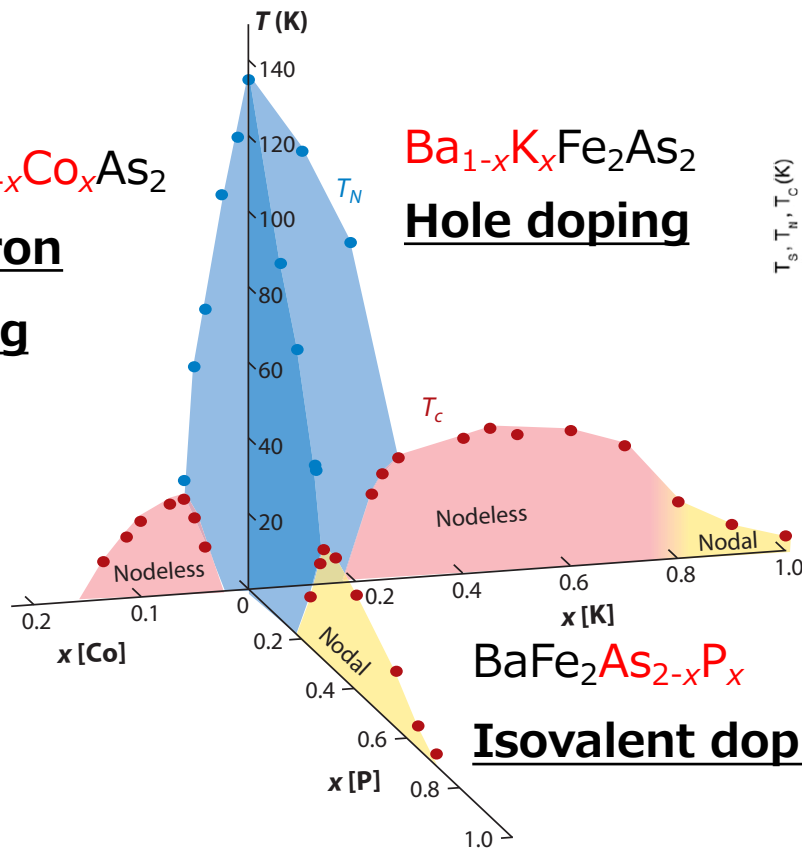


V. Vildosola et al., *Phys. Rev. B* **78** 064518 (2008).

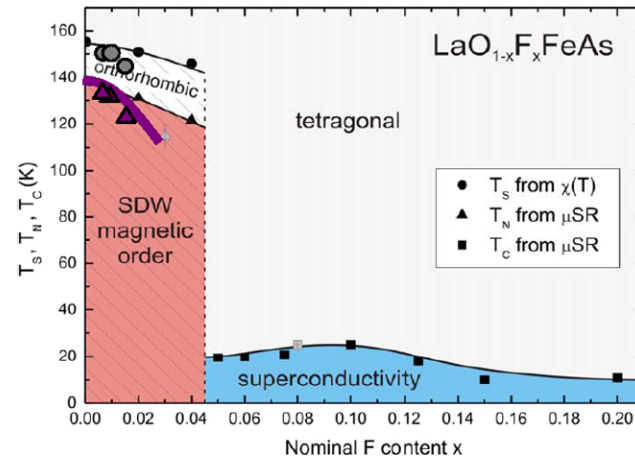
Magnetic phase diagrams of 122, 1111 and 11

BaFe₂As₂ (Ba122) system

BaFe_{2-x}Co_xAs₂
Electron
doping

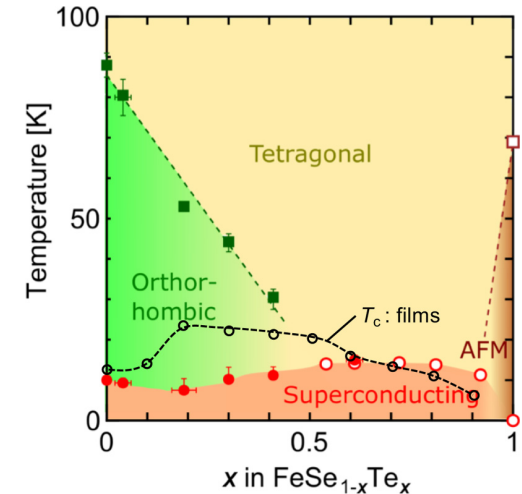


LaFeAsO_{1-x}F_x



A. Köhler et al., *J. Supercond. Nov. Magn.* **22** 565 (2009).

FeSe_{1-x}Te_x



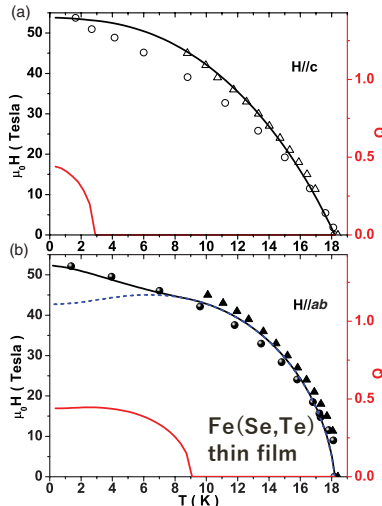
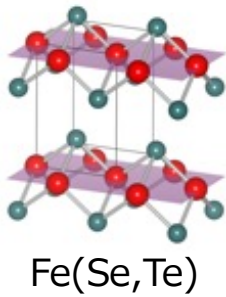
K. Terao et al., *Phys. Rev. B.* **100** 224516 (2019).

- **AFM is suppressed by doping**
- **Presence of the SC domes**
- **Direct doping to the FeAs layer is not harmful to SC (c.f. Cuprates)**

T. Shibauchi et al., *Annu. Rev. Condens. Matter Phys.* **5** 113 (2014).

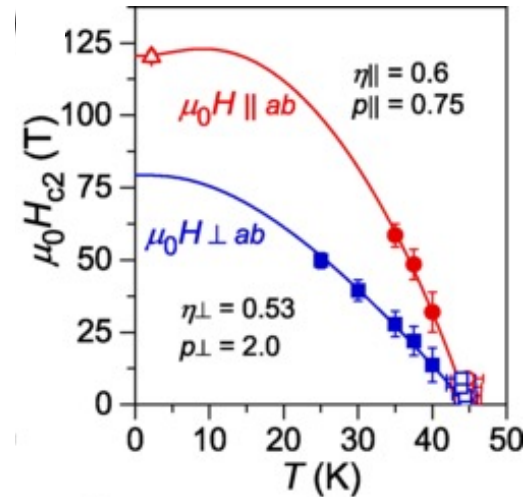
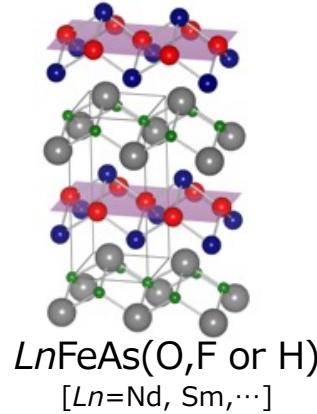
High upper critical field, H_{c2} , and low anisotropy

11



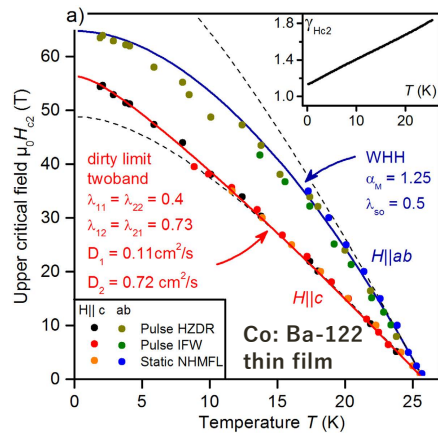
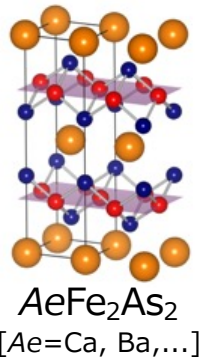
C. Tarantini *et al.*, *Phys. Rev. B* **84** 184522 (2011).

1111



K. Hanzawa *et al.*, *Phys. Rev. Materials* **6** L111801 (2022).

122



J. Hänisch *et al.*, *Sci. Rep.* **5** 17363 (2015).

| | 11 | 122 | 1111 |
|----------------|-----|-----|------|
| γ_{Hc2} | 1-2 | 1-2 | 1-5 |

- Large H_{c2} over 50 T at low T even $H||c$
- Low anisotropy

Comparison between cuprates and IBSSs

| | Cuprates | IBSSs |
|---|--|---|
| Degree of freedom in material design | high (many compounds) | high (many compounds) |
| Parent compound | Mott insulator | AFM bad metal |
| Gap symmetry | d-wave, single band | extended s-wave (s_{\pm} or s_{++}), 5 bands |
| Doping | Hole: $T_c=154$ K ⁱ⁾ Electron: $T_c=30$ K ⁱⁱ⁾ | Hole: $T_c=38$ K ⁱⁱⁱ⁾ , isovalent: $T_c=31$ K ^{iv)} Electron: $T_c=55$ K ^{v)} , 65 K ^{vi)} |
| H_{c2} anisotropy | ~ 5 : RE-123 ~ 150 : Bi-2223 | 1~2: 11 and 122 1~5: Ln-1111 |
| Pairing mechanism | spin fluctuations? | spin fluctuations? orbital fluctuations? |

i) $\text{HgBa}_2\text{Ca}_2\text{Cu}_3\text{O}_{8+\delta}$ w/pressure

ii) $\text{La}_{1-x}\text{Ce}_x\text{CuO}_4$

iii) $\text{Ba}_{0.6}\text{K}_{0.4}\text{Fe}_2\text{As}_2$

iv) $\text{BaFe}_2(\text{As}_{0.66}\text{P}_{0.33})_2$

v) $\text{LnFeAs}(\text{O},\text{F})$ ($\text{Ln}=\text{Nd}, \text{Sm}$)

vi) FeSe monolayer

Overview

1. Iron-based superconductors (IBSs)

- Physical properties

2. Tuning of the superconducting properties

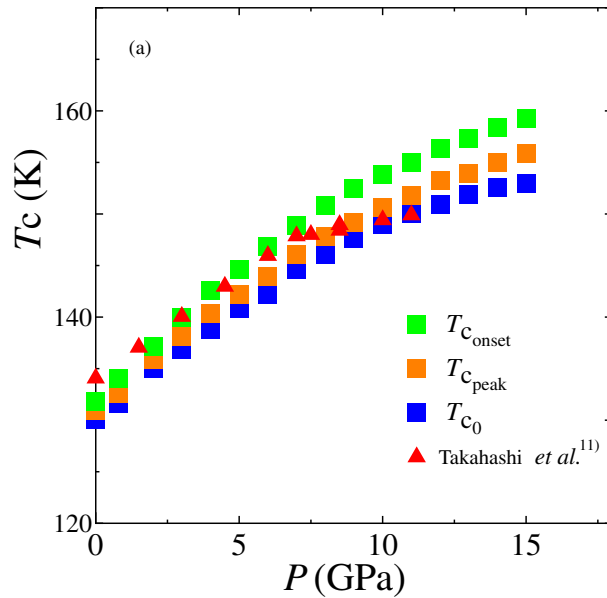
- SC transition temperature (strain, monolayer, intercalation, EDLT)
- Grain boundary, GB
- Critical current density (natural defects, APC, thermodynamic approach)

3. Progress Toward applications

- Use of IBS wires and bulks in magnets, and perspective

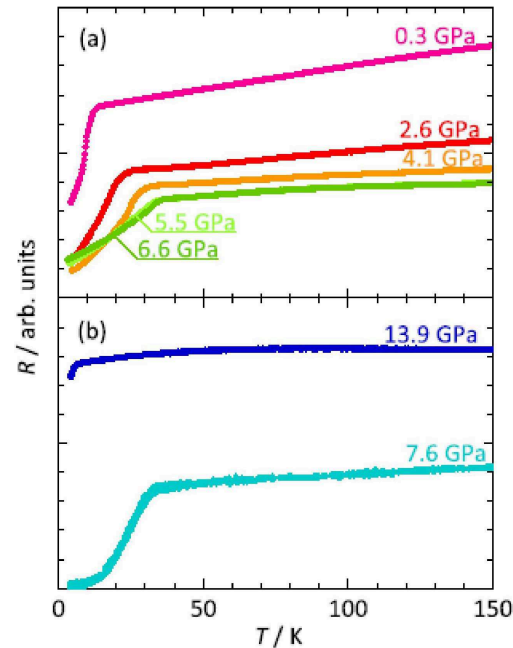
A significant jump of T_c by pressure

HgBa₂Ca₂Cu₃O_{8+δ}:
130 K -> 153 K (15 GPa)

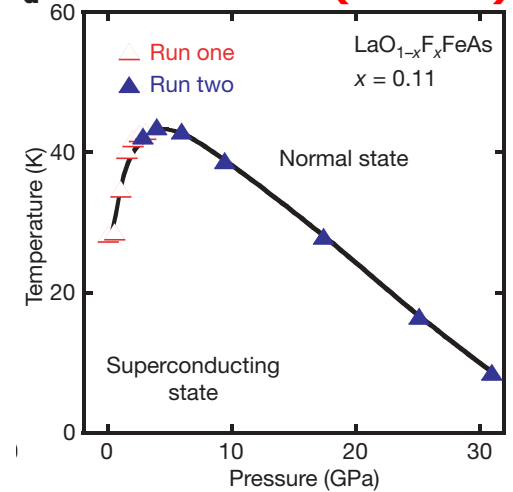


N. Takeshita *et al.*, *J. Phys. Soc. Jpn.* **82** 023711 (2013). S. Margadonna *et al.*, *PRB* **80** 064506 (2009).

FeSe:
9 K -> 37 K (7 GPa)

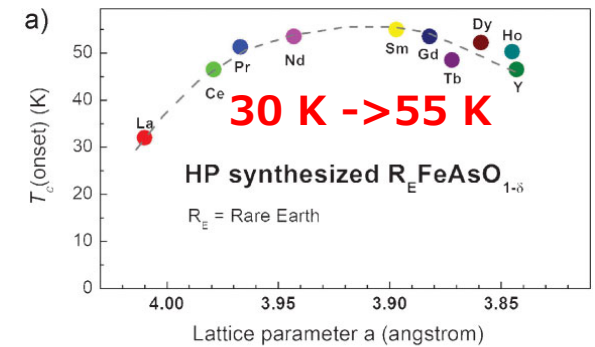


LaFeAsO_{0.89}F_{0.11}:
25 K -> 43 K (4 GPa)



H. Takahashi *et al.*, *Nature* **453** 376 (2008).

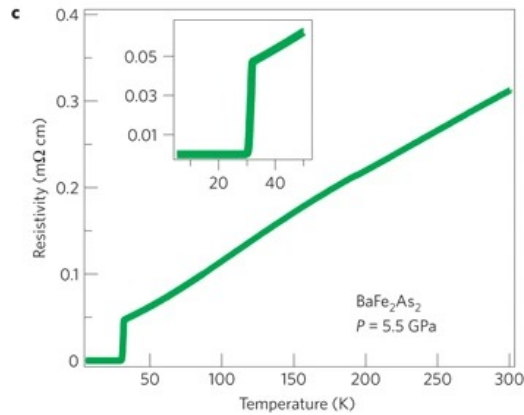
- Isotropic pressure enhances T_c
- Chemical pressure also enhances T_c



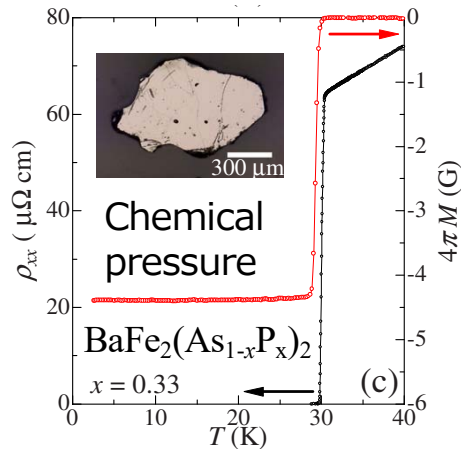
Z. Ren, Z. Zhao, *Adv. Mater.* **21** 4584 (2009).

Pressure & strain induced superconductivity in Ba122

Bulk



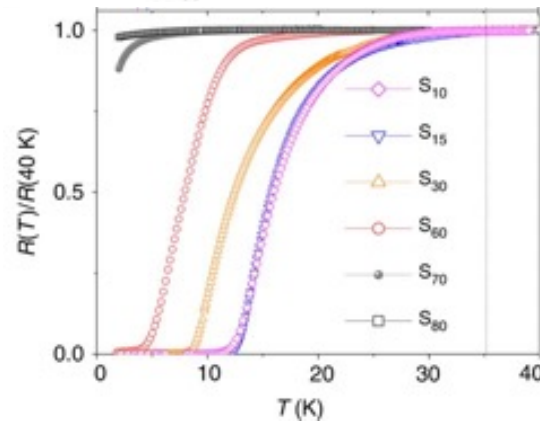
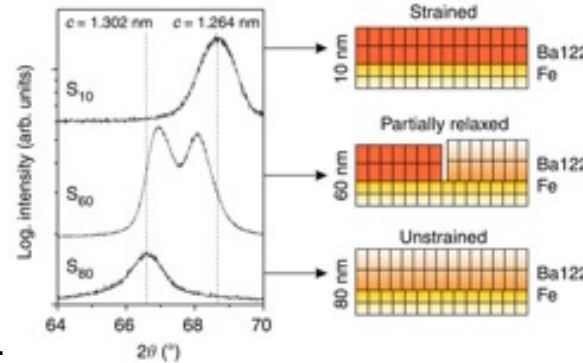
S. J. A. Kimber *et al.*, *Nat. Mater.* **8**, 471 (2009).



S. Kasahara *et al.*, *PRB* **81** 184519 (2010).

Bilayer

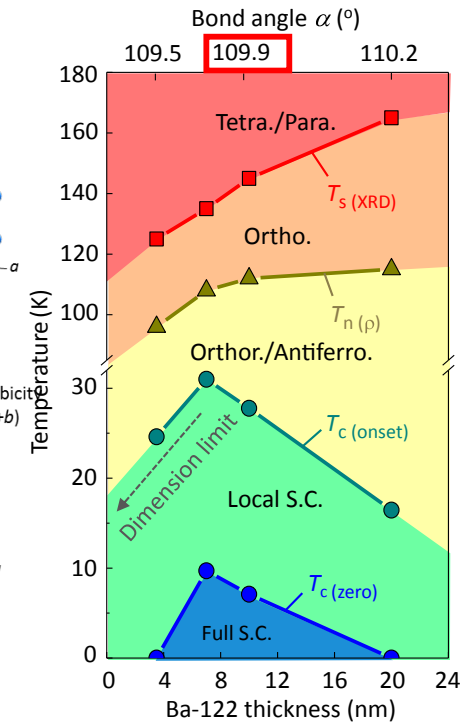
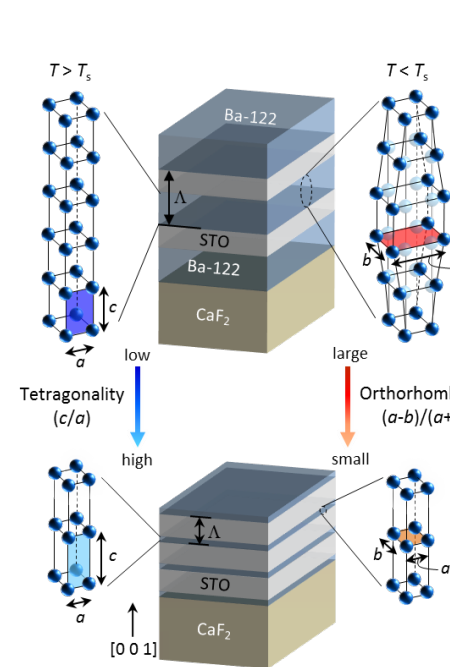
Fe: 30 nm/Ba122: 10~80 nm



J. Engelmann, K.I. *et al.*, *Nat. Commun.* **4** 2877 (2013).

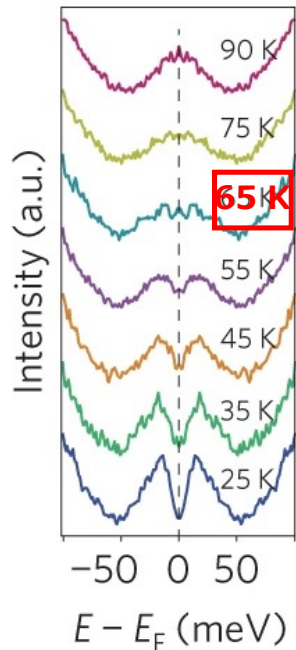
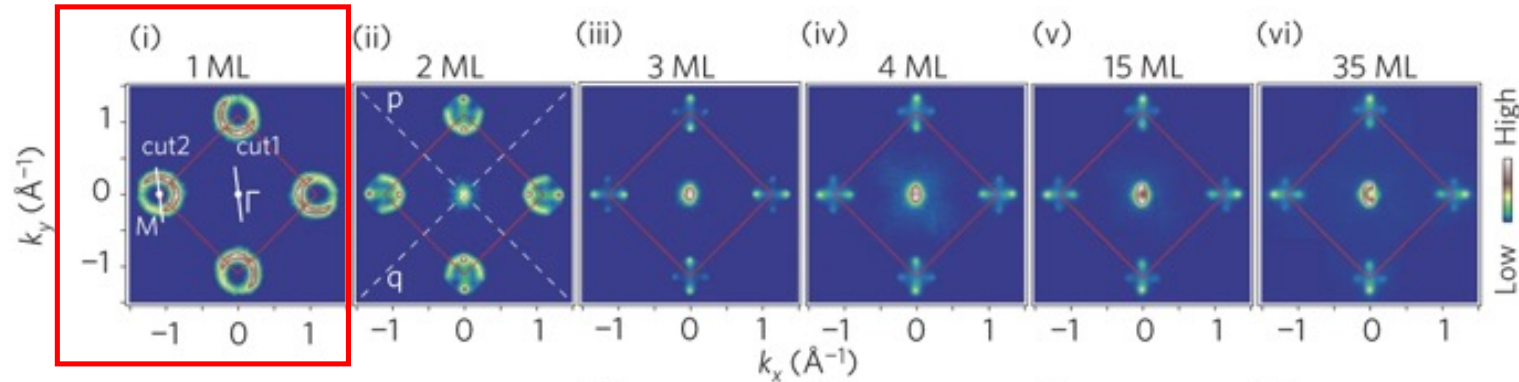
Superlattice

SrTiO_3 : 14 nm/Ba122: 4~20 nm



J. Kang *et al.*, *PNAS* **117** 21170 (2020).

$T_c = 65$ K in a monolayer of FeSe



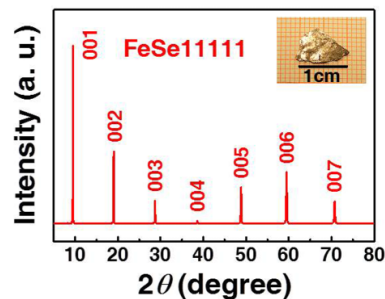
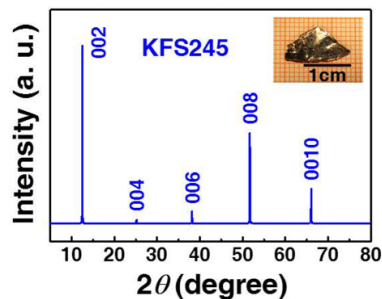
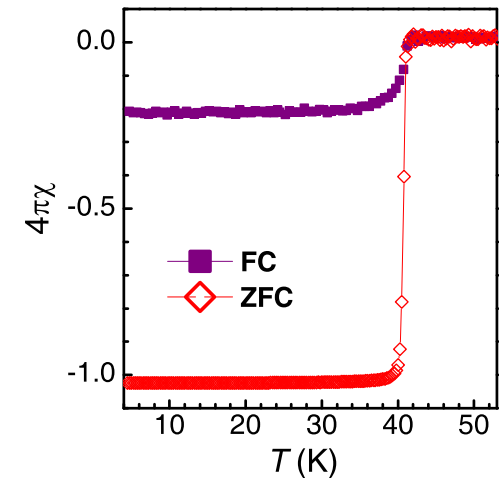
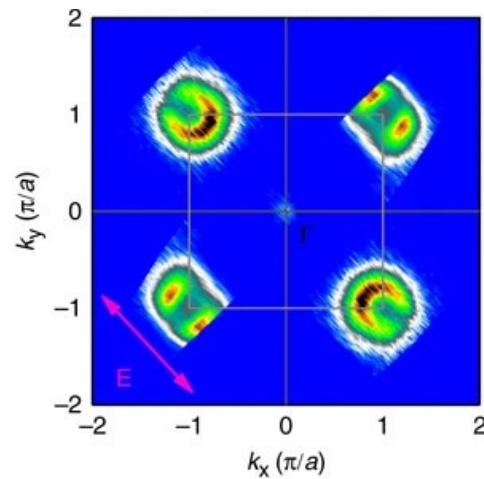
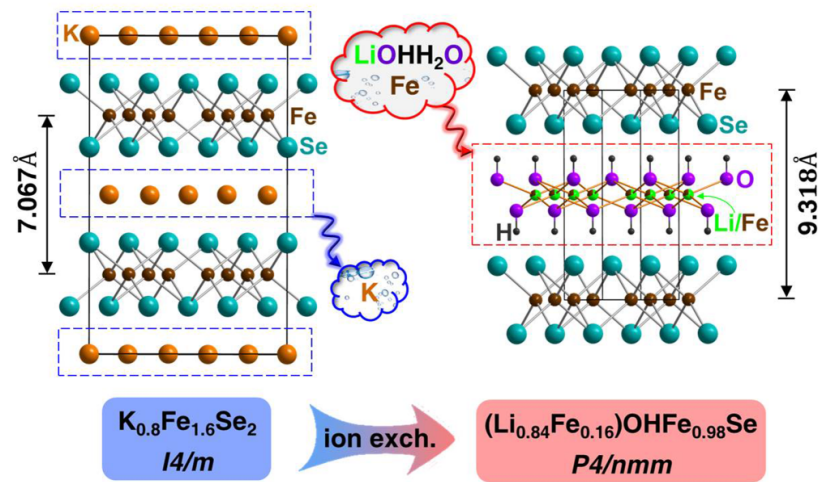
- SC gap opened at between $55 \text{ K} \leq T \leq 65 \text{ K}$ for 1 ML
- Only electron-like pockets appeared at M-point for 1 ML
- Hole-like pocket started to appear at Γ -point for over 2 ML
- FeSe ML is very sensitive to air -> quickly degraded

Wang *et al.*, *Chin. Phys. Lett.* **29** 037402 (2012).

Tan *et al.*, *Nat. Mater.* **12** 634 (2013).

Tuning superconducting properties, T_c , by intercalation

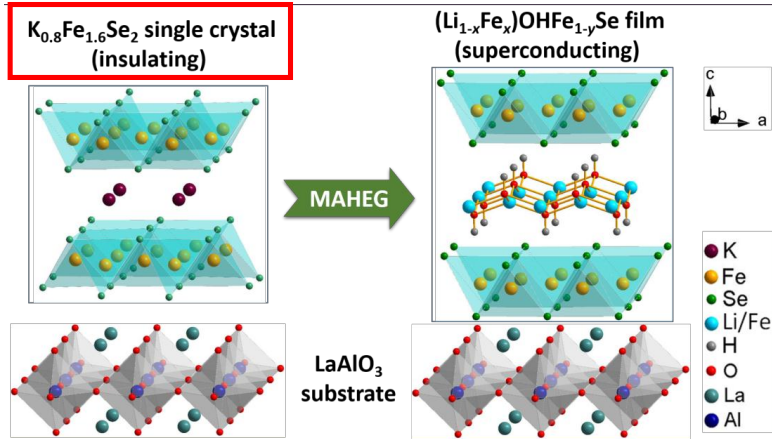
(Li,Fe)OHFeSe (1111) single crystal



- (Li,Fe)OHFeSe single crystals grown by ion exchange
- Electronic structure is very similar to the FeSe ML ($T_c \sim 42$ K)

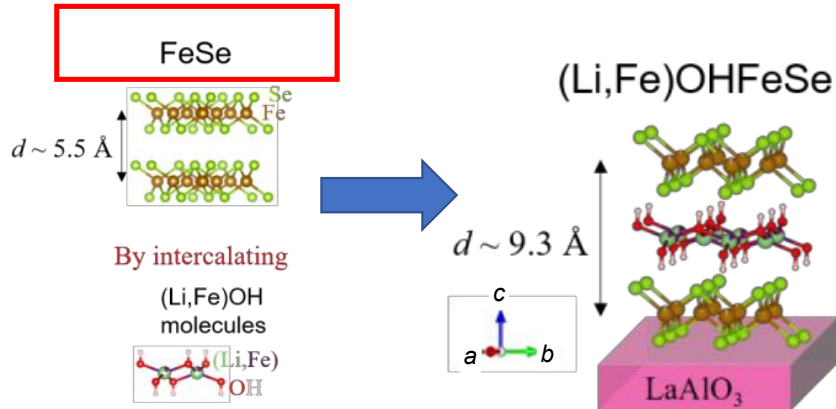
Dong *et al.*, *Phys. Rev. B*, **92** 064515 (2015).

Tuning superconducting properties, T_c , by intercalation



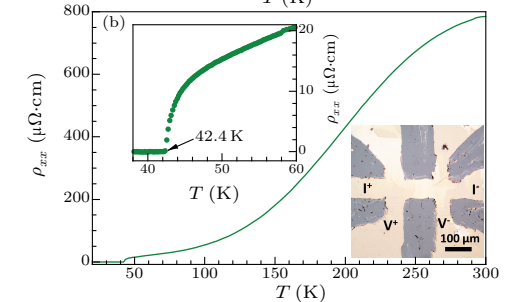
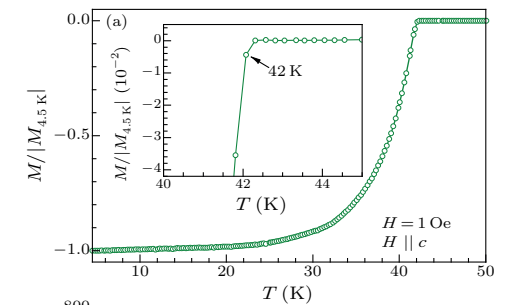
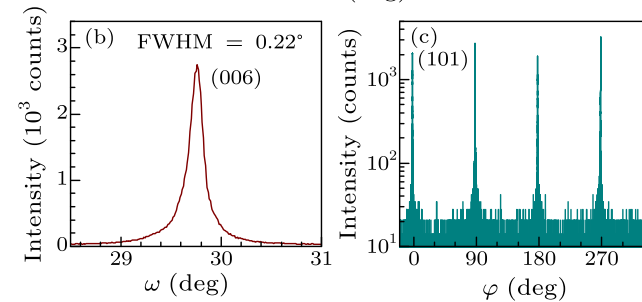
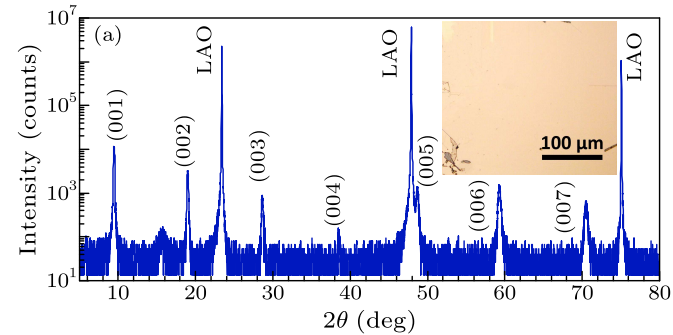
matrix-assisted hydrothermal epitaxial growth

Huang et al., arXiv:1711.02920 (2017).



Dong et al., Chinese Phys. Lett. **39** 127402 (2022).

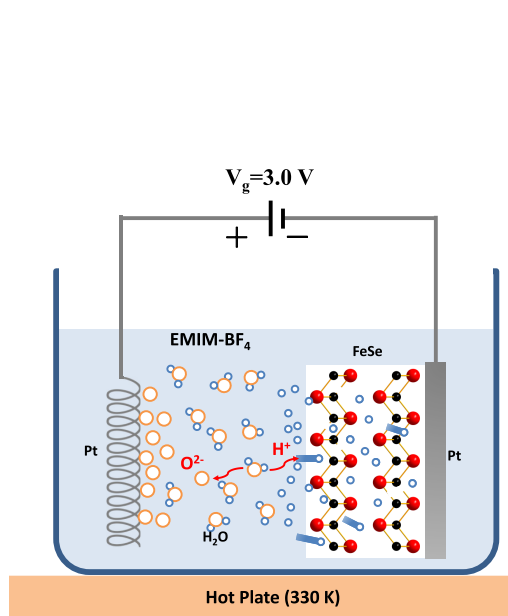
Epitaxial (Li,Fe)OHFeSe thin film



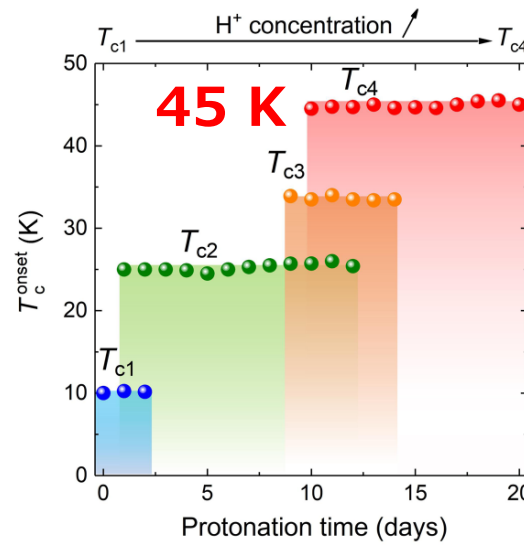
- (Li,Fe)OHFeSe epitaxial thin films
- $T_c \sim 42 \text{ K}$, comparable to single crystal

Tuning superconducting properties, T_c , by protonation

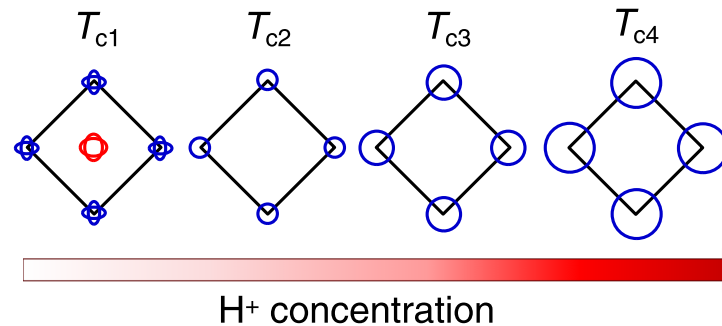
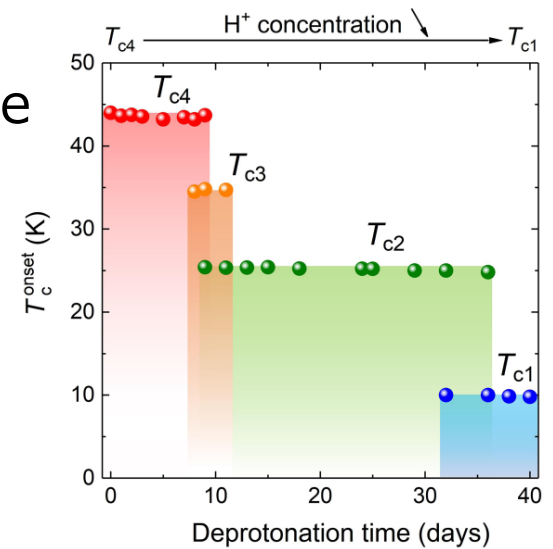
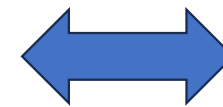
Protonation-induced SC in FeSe



H₂O in an ionic liquid



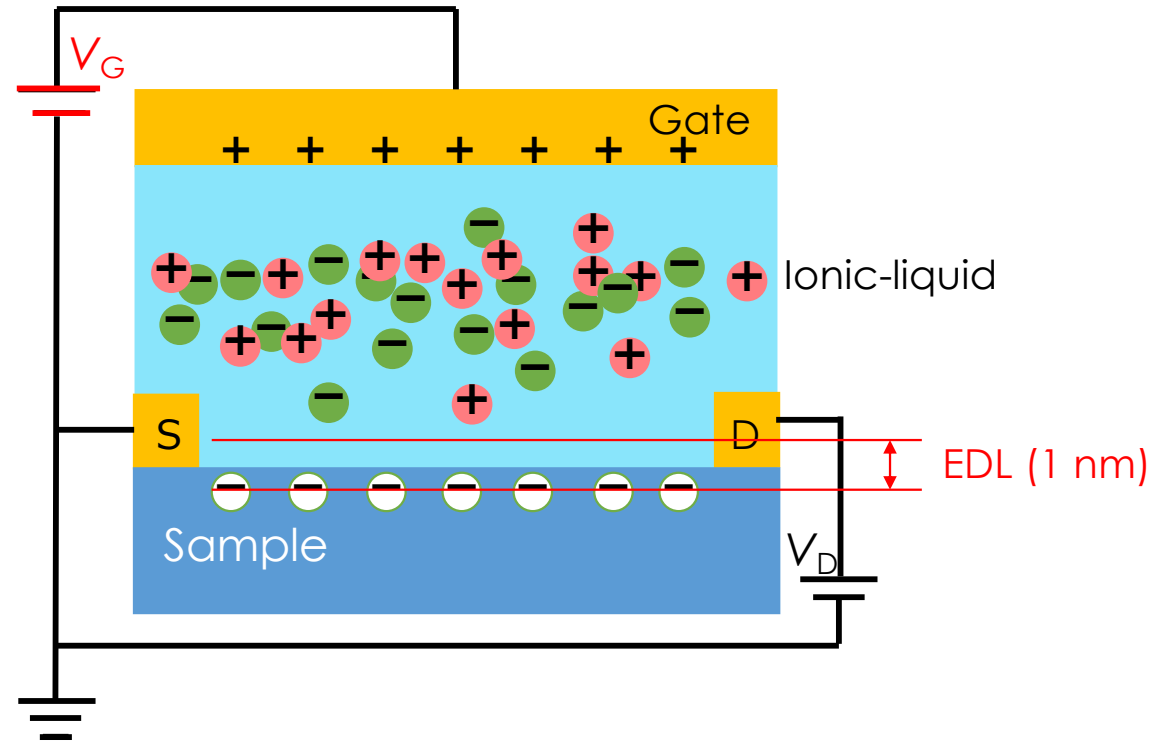
Reversible process



Heavily electron doping induces high T_c

Y. Meng *et al.*, *Phys. Rev. B* **105** 134506 (2022).

Electric double layer transistor (EDLT)



Electro-static
carrier doping

$V_G > 0 \dots$ electron doping
 $V_G < 0 \dots$ hole doping

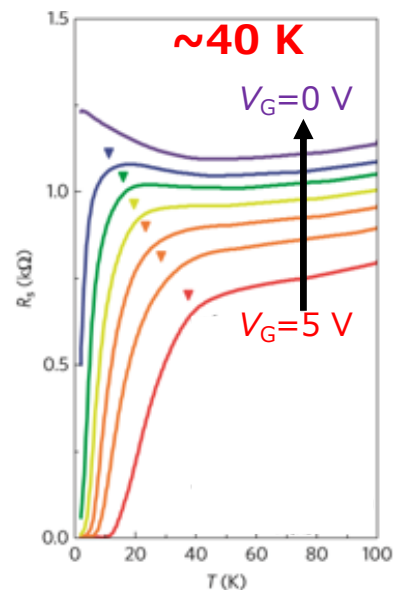
Tuning superconducting properties by EDLT

SC induced by EDLT

first demonstration

in IBS [$\text{FeSe} \sim 0.6 \text{ nm}$]

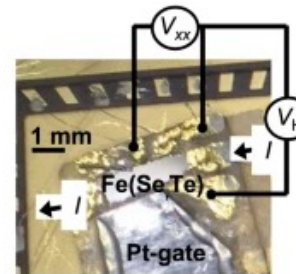
J. Shiogai *et al.*, *Nat. Phys.* **12** 42 (2016).



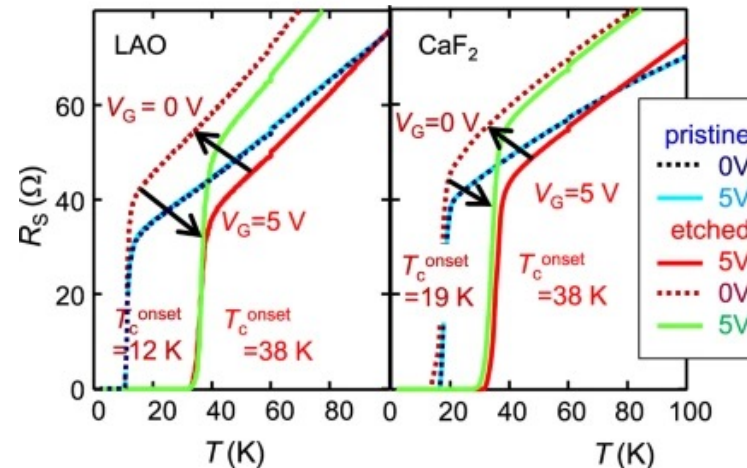
T_c enhancement by EDLT

$\text{FeSe}_{0.8}\text{Te}_{0.2}$ ($> 10 \text{ nm}$)

S. Kouno *et al.*, *Sci. Rep.* **8** 14731 (2018).



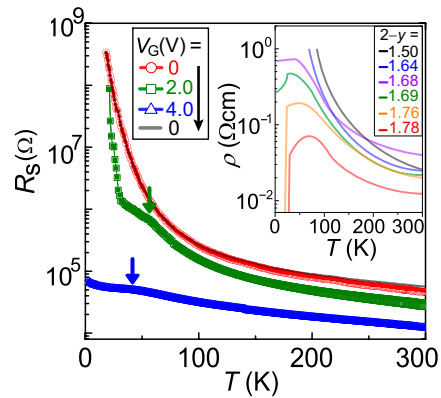
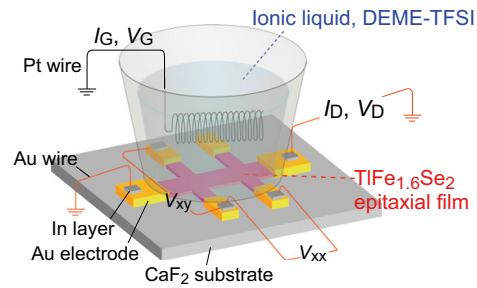
12 K \rightarrow 38 K



So far, enhancement of T_c by EDLT only for FeSe & Fe(Se,Te)

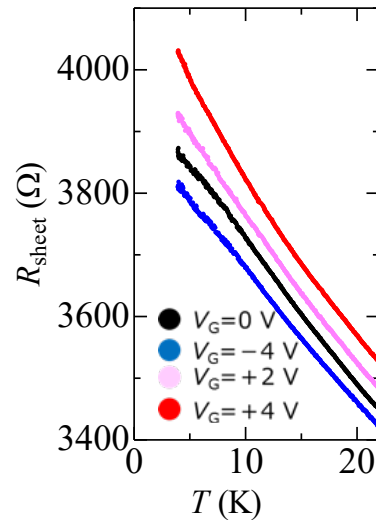
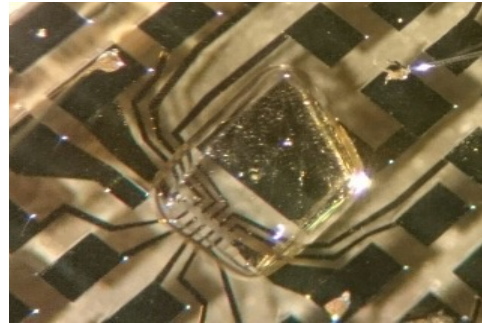
TlFe_{1.6}Se₂ (~20 nm)

T. Katase *et al.*, *PNAS* **111** 3979 (2014).



NdFeAsO (~7 nm)

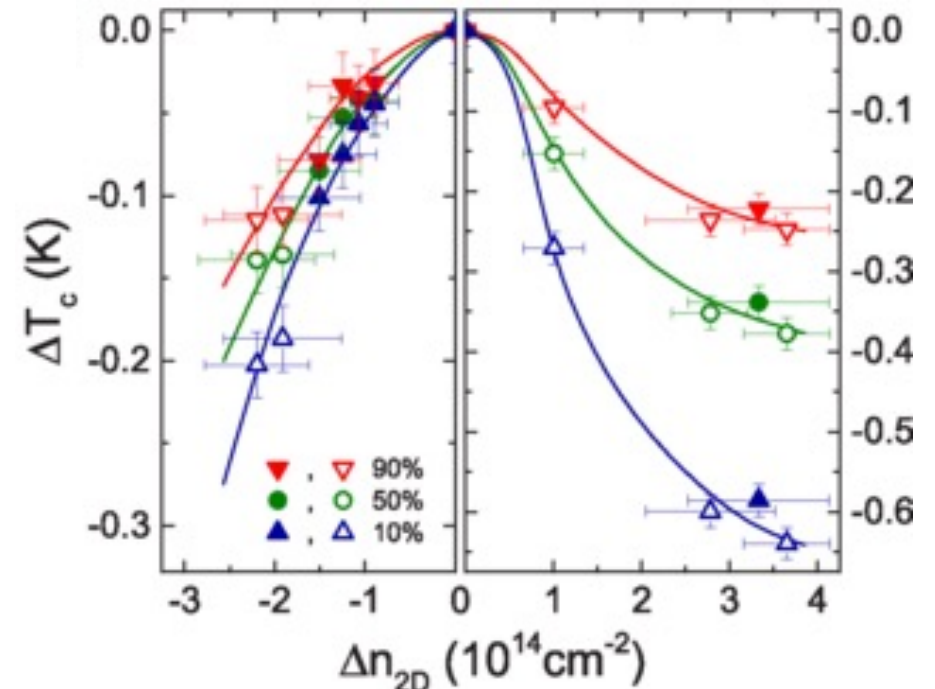
K. Iida *et al.*, *unpublished*



Ambipolar suppression of T_c

BaFe₂(As,P)₂ (~10 nm)

E. Piatti, K.I. *et al.*, *Phys. Rev. Mater.* **3** 044801 (2019).



Overview

1. Iron-based superconductors (IBSs)

- Physical properties

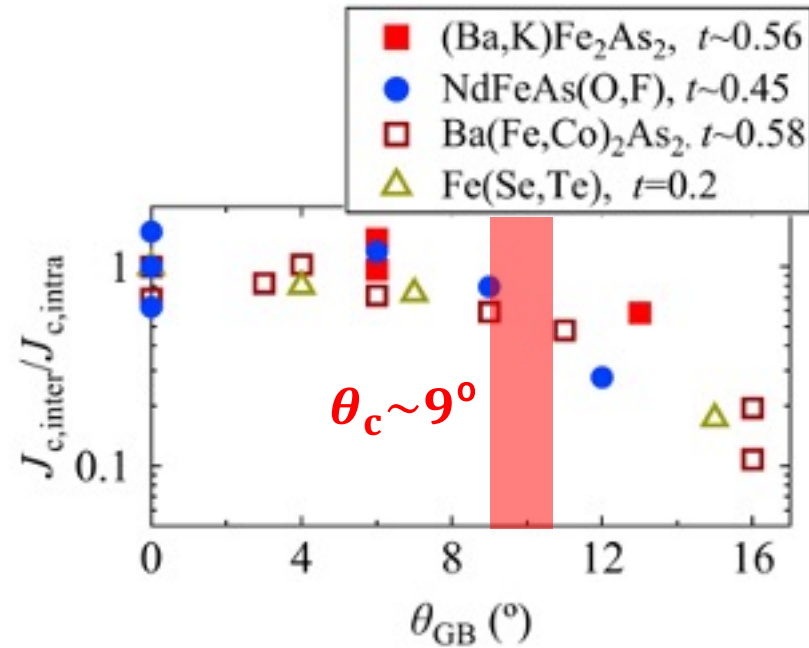
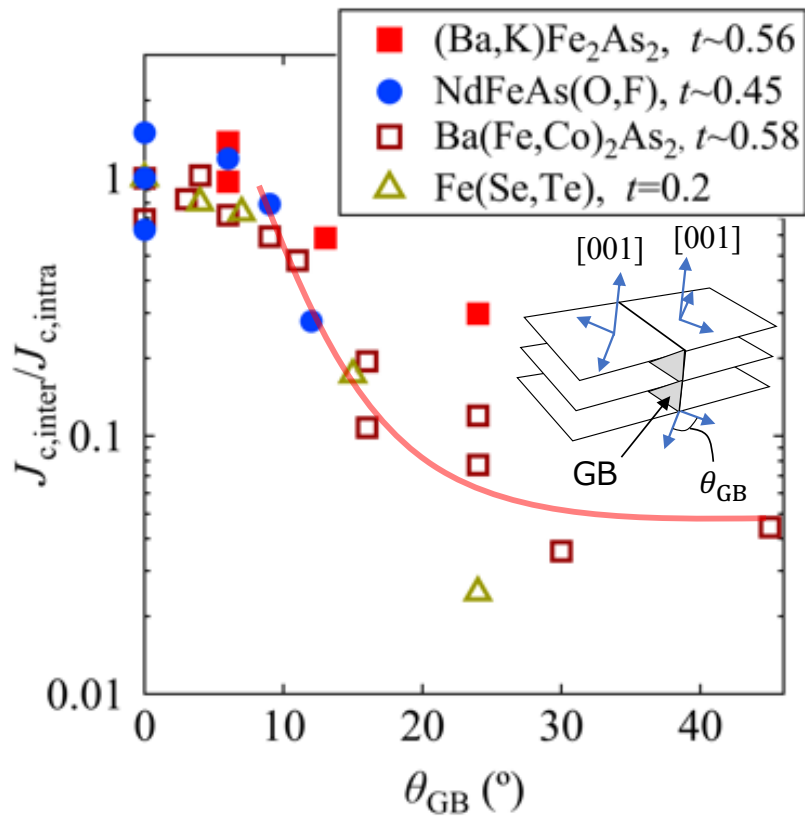
2. Tuning of the superconducting properties

- SC transition temperature (strain, monolayer, intercalation, EDLT)
- Grain boundary, GB
- Critical current density (natural defects, APC, thermodynamic approach)

3. Progress Toward applications

- Use of IBS wires and bulks in magnets, and perspective

Large critical angle θ_c & constant $J_{c,inter}$ ($\theta_{GB} > \sim 15^\circ$)

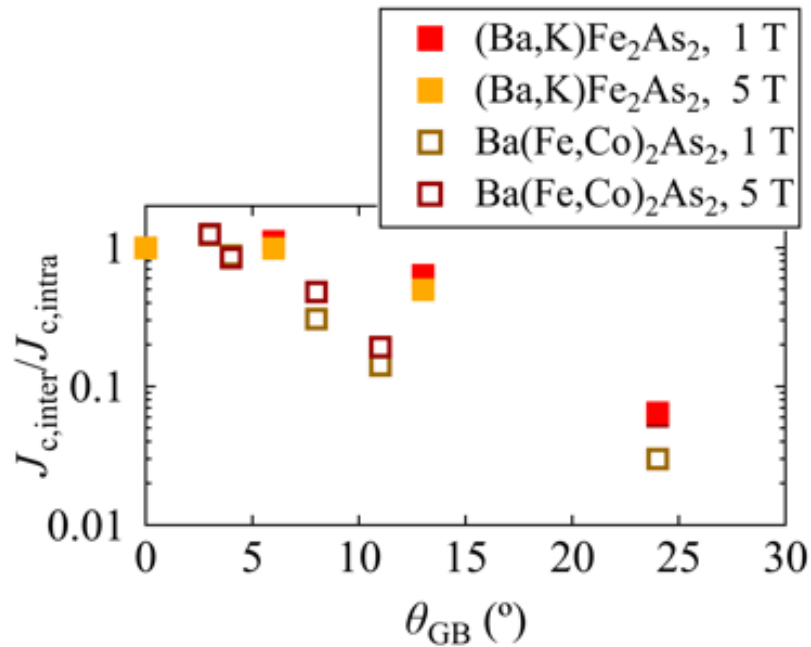


- A critical angle θ_c of 9° , which is larger than cuprate
- The $J_{c,inter}$ is constant at $\theta_{GB} > \sim 15^\circ$

T. Hatano, K.I. et al., *NPG Asia Mater.* **16** 41 (2024).
 K. Iida et al., *Supercond. Sci. Technol.* **32** 074003 (2019).
 T. Katase et al., *Nat. Commun.* **2** 409 (2011).
 W Si et al., *Appl. Phys. Lett.* **106** 032602 (2015).
 E. Sarnelli et al., *IEEE Trans. Appl. Supercond.* **27** 7400104 (2017).

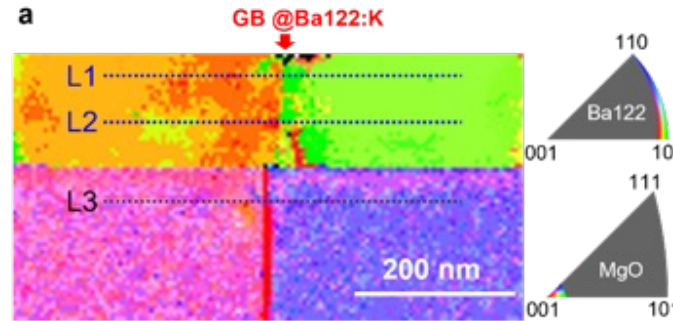
Excellent GB properties in (Ba,K)122

Unchanged θ_c for (Ba,K)122 even in field

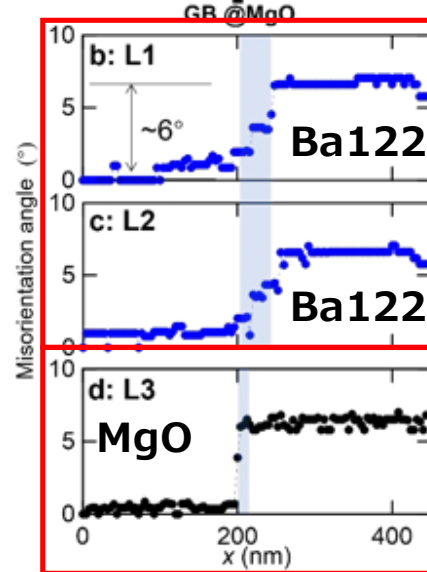


T. Hatano, K. I. *et al.*, *NPG Asia Mater.* **16** 41 (2024).

SPED (scanning precision electron diffraction)

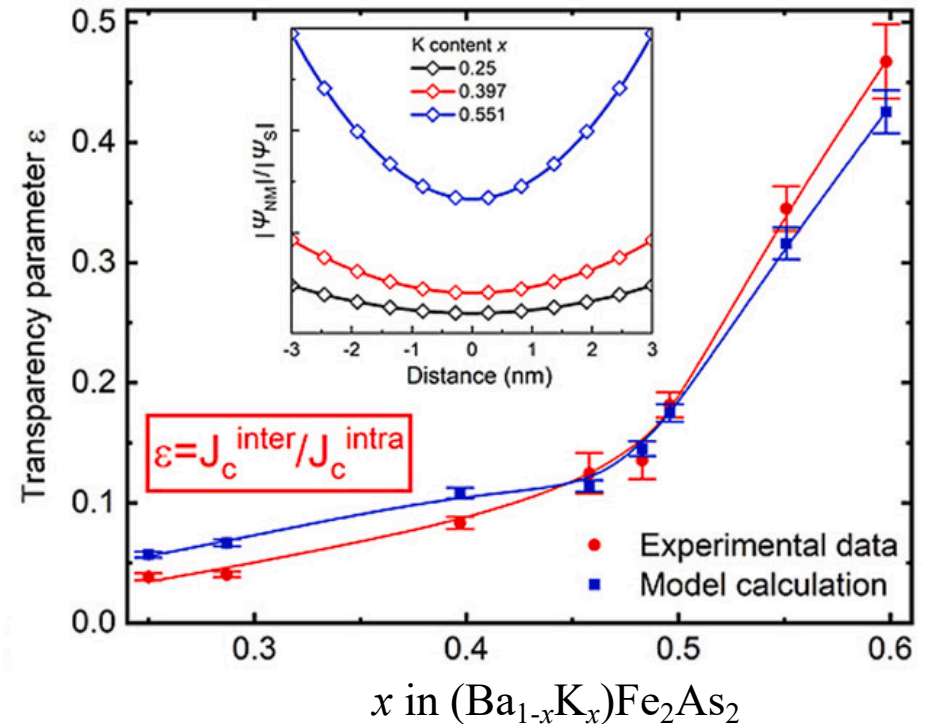
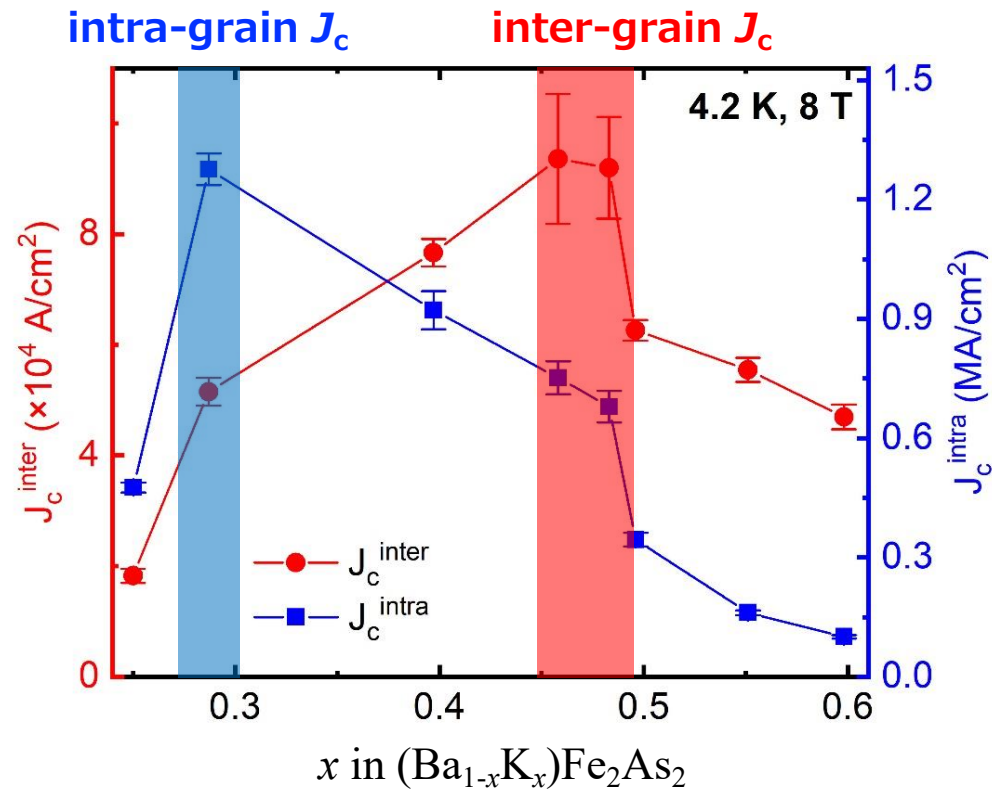


(Ba,K)122 having $\theta_{GB}=6^\circ$



- As expected, GB is sharp for MgO
- GB angles of K:Ba122 gradually change

GB transparency is increased by over-doping (Ba,K)122

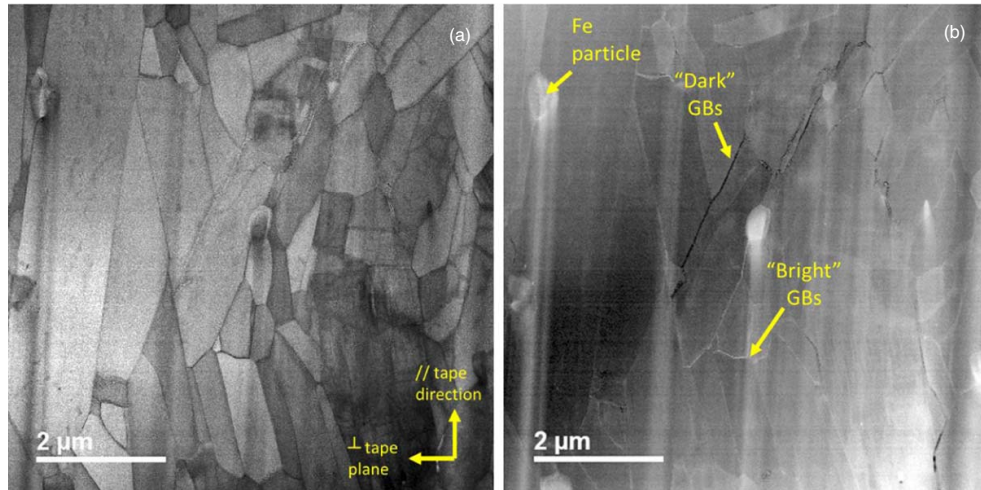


- Over-doped grains enhanced the proximity effect
- SNS JJ model described the data

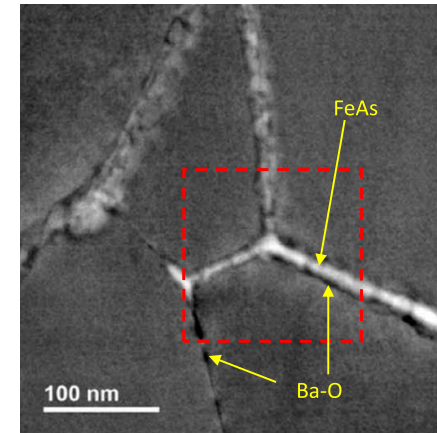
Z. Cheng *et al.*, *Mater. Today Phys.* **28** 100848 (2022).

Detailed analyses on polycrystalline (Ba,K)122

Hot-pressed (Ba,K)122 tape



FeAs wetting phase and BaO (Blocking the supercurrent flow)



- Grain alignment is important (GB connectivity)
- Clean GB is important (the number of GB connections)

Detailed in 4MOr2A-01
by F. Kametani (4/9)

F. Kametani *et al.*, *Appl. Phys. Express* **17** 013004 (2024).

Overview

1. Iron-based superconductors (IBSs)

- Physical properties

2. Tuning of the superconducting properties

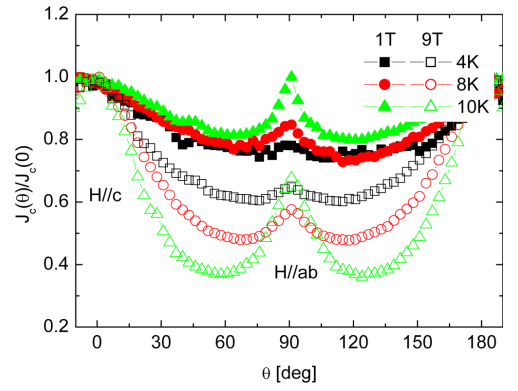
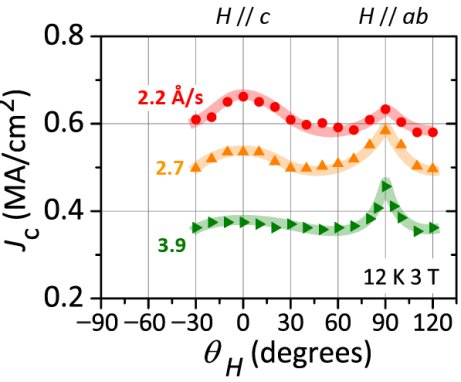
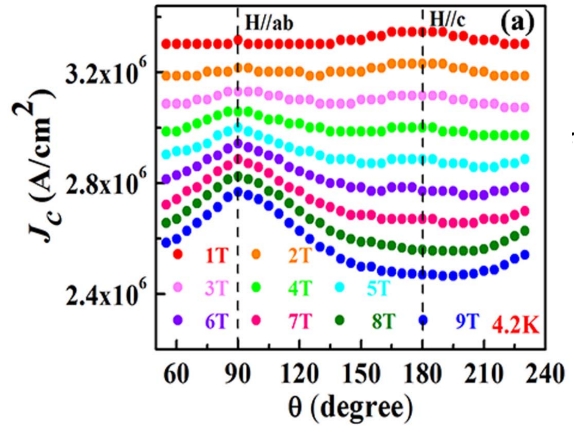
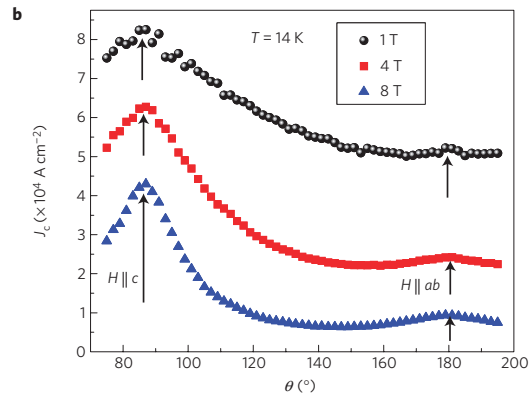
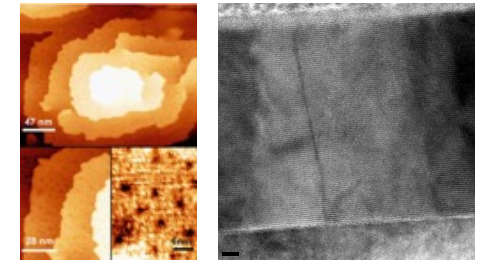
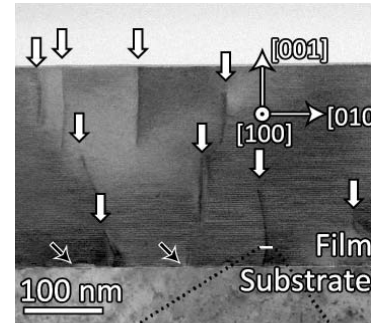
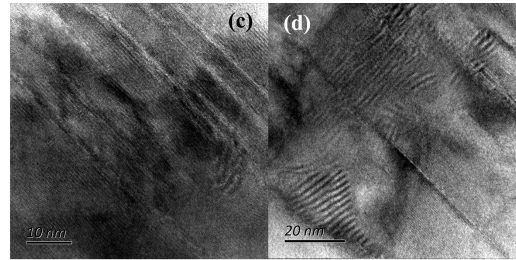
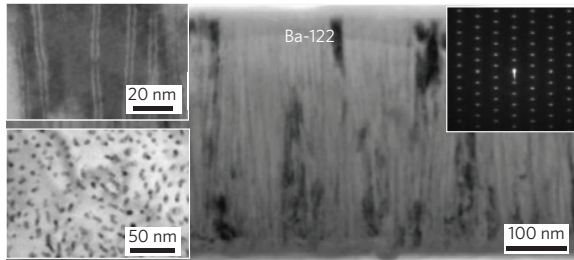
- SC transition temperature (strain, monolayer, intercalation, EDLT)
- Grain boundary, GB
- Critical current density (natural defects, APC, thermodynamic approach)

3. Progress Toward applications

- Use of IBS wires and bulks in magnets, and perspective

Natural defects in IBS thin films

- **Co-doped Ba-122 on SrTiO₃ template** ➤ **Co-doped Ba-122 on CaF₂** ➤ **P-doped Ba-122 on MgO** ➤ **Fe(Se,Te) on SrTiO₃**
- Ba-Fe-O nano pillar** (columnar defects) **Stacking faults** **Dislocations** **Threading dislocations**

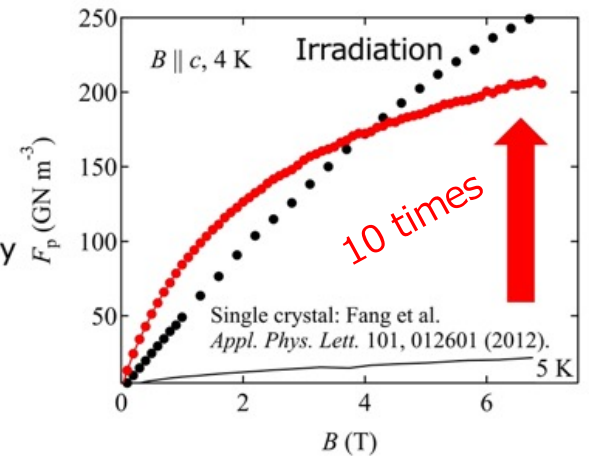
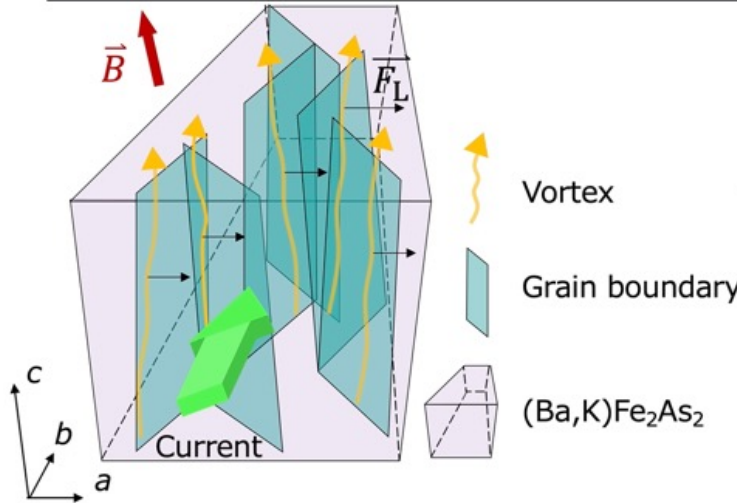
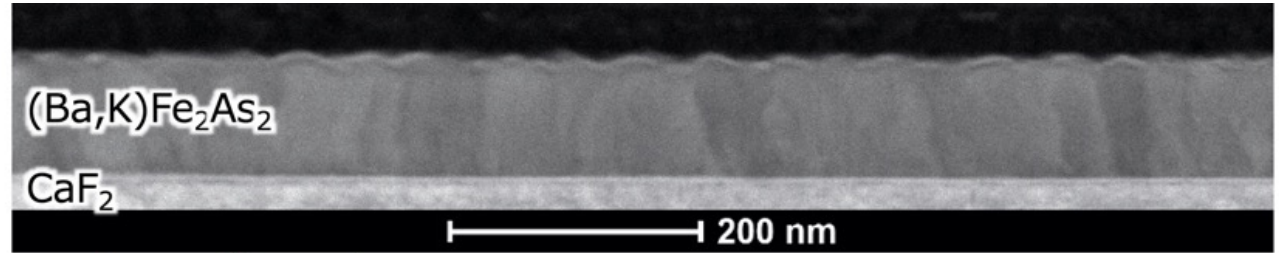
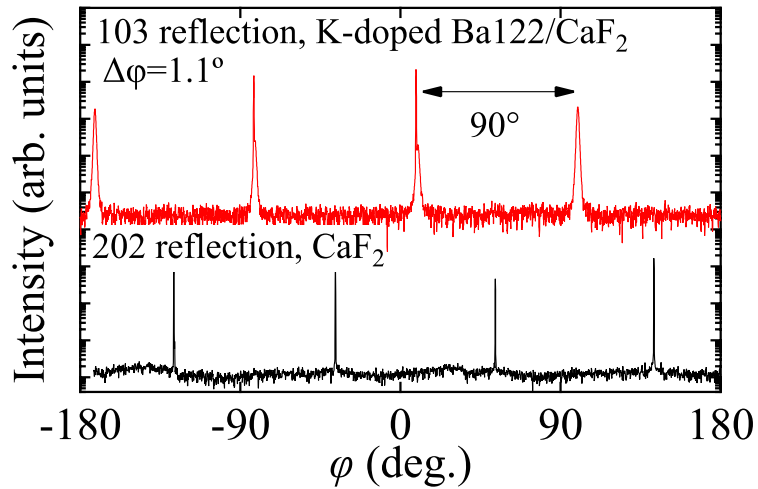
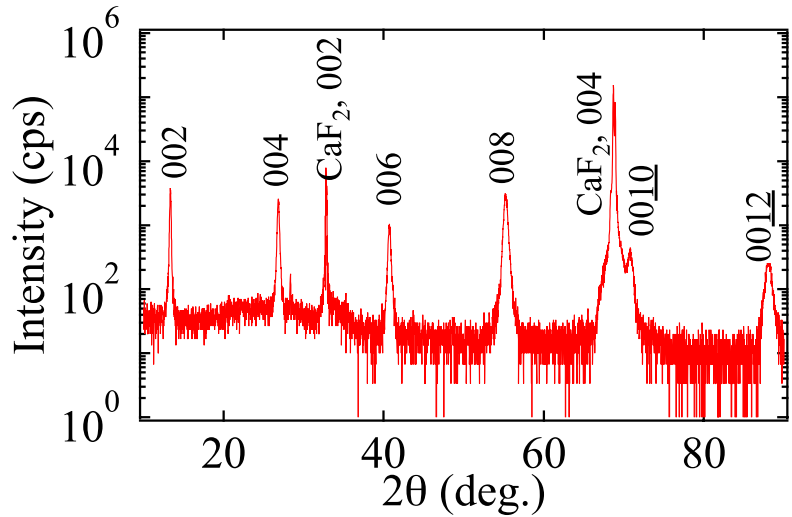


S. Lee *et al.*, *Nat. Mater.* **9** 397 (2010).

P. Yuan *et al.*, *SuST* **30** 025001 (2017).

H. Sato *et al.*, *APL* **104** 182603 (2014). E. Bellingeri *et al.*, *APL* **100** 082601 (2012).

Natural defects in IBS thin films



**Fluoride sub. & low growth temperature
($\sim 400^\circ\text{C}$) are the key to epitaxial growth**

D. Qin, KI et al., *Phys. Rev. Materials* **5** 014801 (2021). ²⁷
K. Iida et al., *NPG Asia Materials* **13** 68 (2021).

APC in IBS thin films

| Materials | Methods | Microstructure | T_c (K) | Refs. |
|----------------|--|---------------------------------------|---|----------|
| Co-doped Ba122 | Multilayer or Quasi-multilayer | Ba122 or SrTiO ₃ insertion | 25.4 -> 26.0 | [1], [2] |
| Fe(Se,Te) | | CeO ₂ insertion, strain | 21.3 -> 20.4 | [3] |
| Co-doped Ba122 | BaZrO ₃ addition to PLD targets | Nano BaZrO ₃ rods | 27.1 -> 24.6 (2 mol% BZO) | [4] |
| P-doped Ba122 | | Nano BaZrO ₃ particles | 26.3 -> 25 (3 mol% BZO) | [5] |
| Co-doped Ba122 | BaHfO ₃ addition to PLD targets | Nano BaHfO ₃ particles | 22.0 -> 19.5 (1 mol%BHO) | [6] |
| FeSe | SrTiO ₃ addition to PLD targets | Nano SrTiO ₃ rods | not shown | [7] |
| Fe(Se,Te) | Proton irradiation | Splayed cascade defects | 18 -> 18.5 ($1 \times 10^{15} \text{ cm}^{-2}$) | [8] |
| NdFeAs(O,F) | α -particle irradiation | No microstructure | 49 -> 46 ($5 \times 10^{15} \text{ cm}^{-2}$) | [9] |
| (Li,Fe)OHFeSe | TM (Mn) doping | No microstructure | 42 -> 37 | [10] |

- ✓ Unlike cuprates, **IBSs are robust against irradiation (disorder)**
- ✓ For Ba-122, T_c decreases with **-1 K/mol%** (cf. -0.2~-0.1 K/mol% for REBCO)

[1] S. Lee *et al.*, *Nat. Mater.* **12** 392 (2013).

[2] C. Tarantini *et al.*, *Sci. Rep.* **4** 7305 (2014).

[3] S. Seo *et al.*, *NPG Asia Materials* **12** 7 (2020).

[4] J. Lee *et al.*, *SuST* **30** 085006 (2017).

[5] M. Miura *et al.*, *Nat. Commun.* **4** 2499 (2013).

[6] S. Meyer *et al.*, *J. Phys.: Conf. Ser.* **1559** 012052 (2020).

[7] T. Horide *et al.*, *Thi. Sol. Films* **733** 138802 (2021).

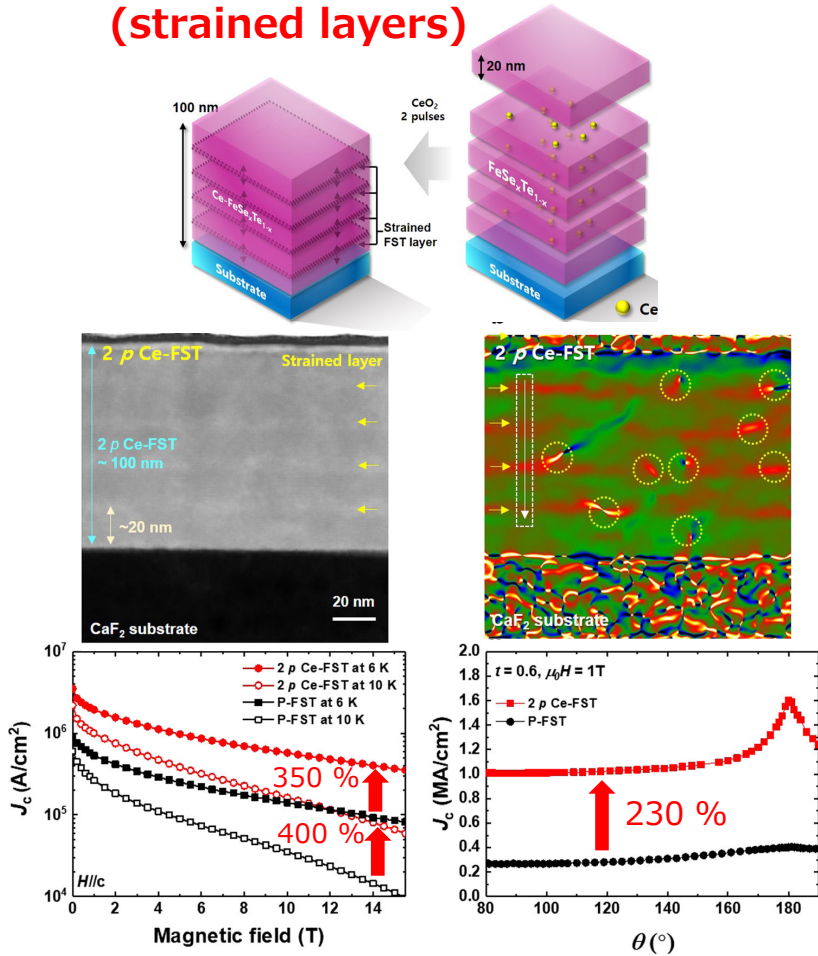
[8] T. Ozaki *et al.*, *Nat. Commun.* **7** 13036 (2016).

[9] C. Tarantini *et al.*, *SuST* **31** 034002 (2018).

[10] D. Li *et al.*, *Sust* **32** 12LT01 (2019).

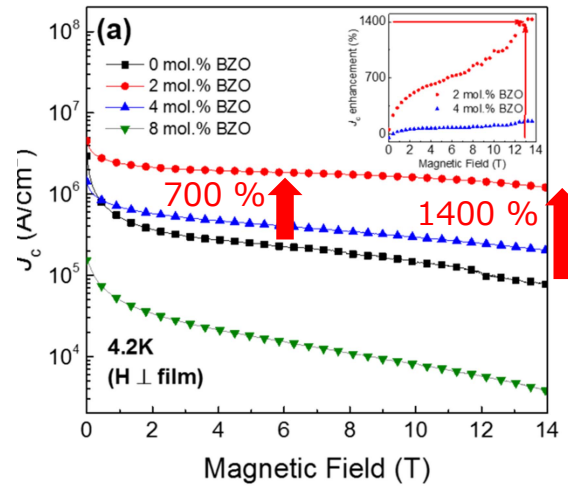
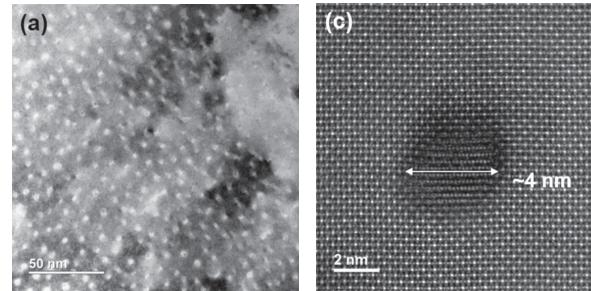
APC in IBS thin films

Quasi-multilayer (strained layers)



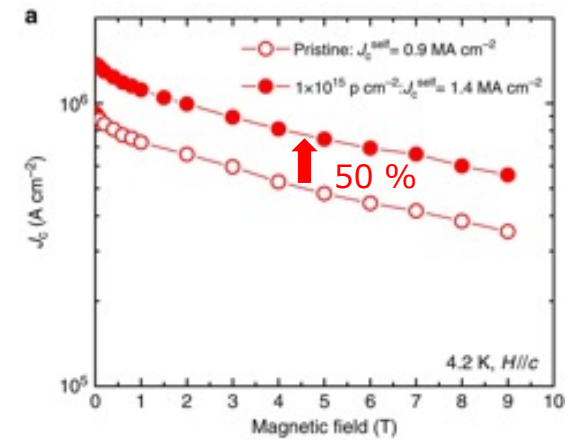
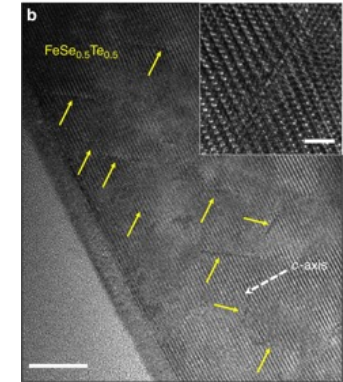
S. Seo *et al.*, NPG Asia Materials **12** 7 (2020).

Target modifications (BZO nano cylinders)



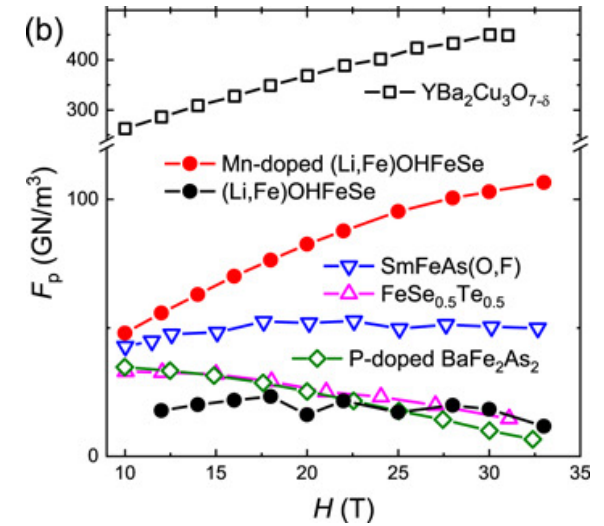
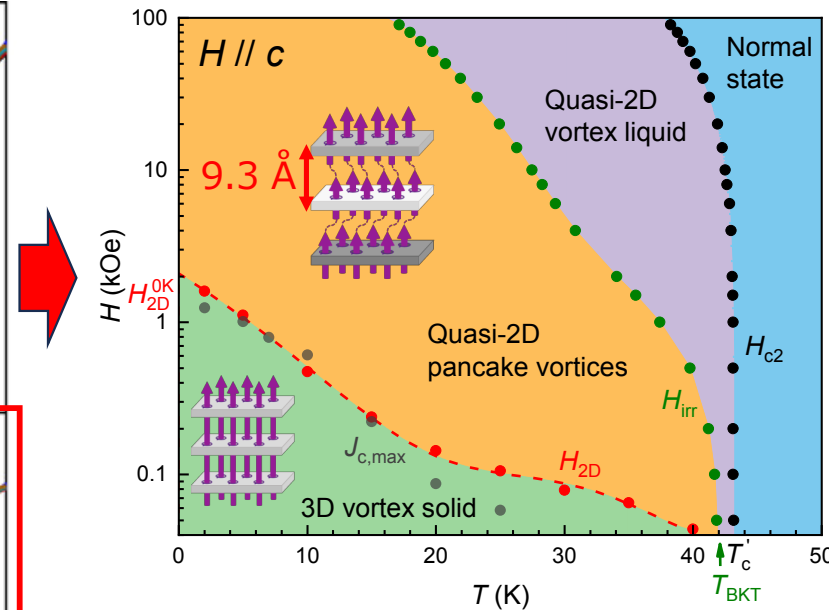
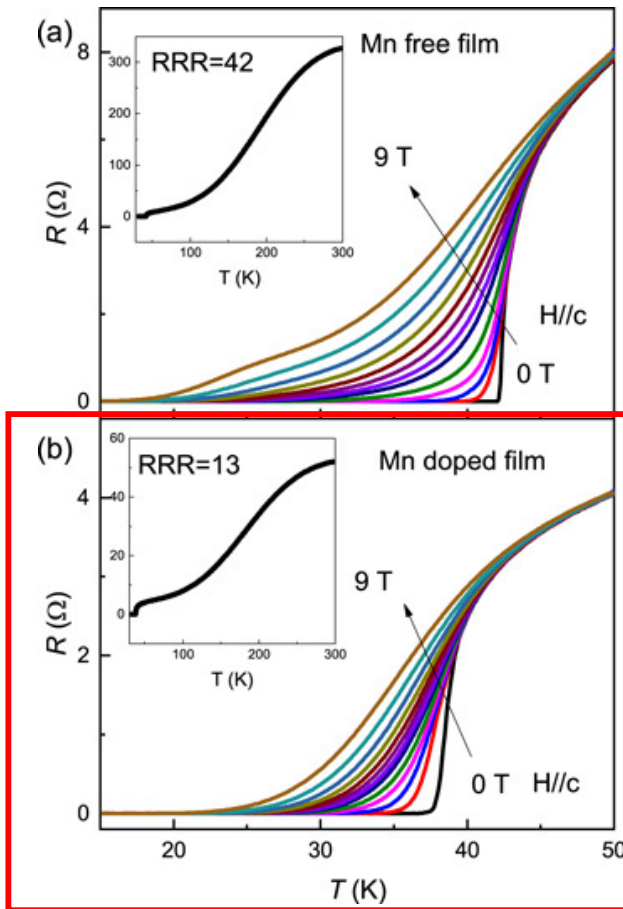
J. Lee *et al.*, SuST **30** 085006 (2017).

Proton irradiation (splayed cascade defects)



T. Ozaki *et al.*, Nat. Commun. **7** 13036 (2016).

Mn-doped (Li,Fe)OHFeSe (1111) thin films



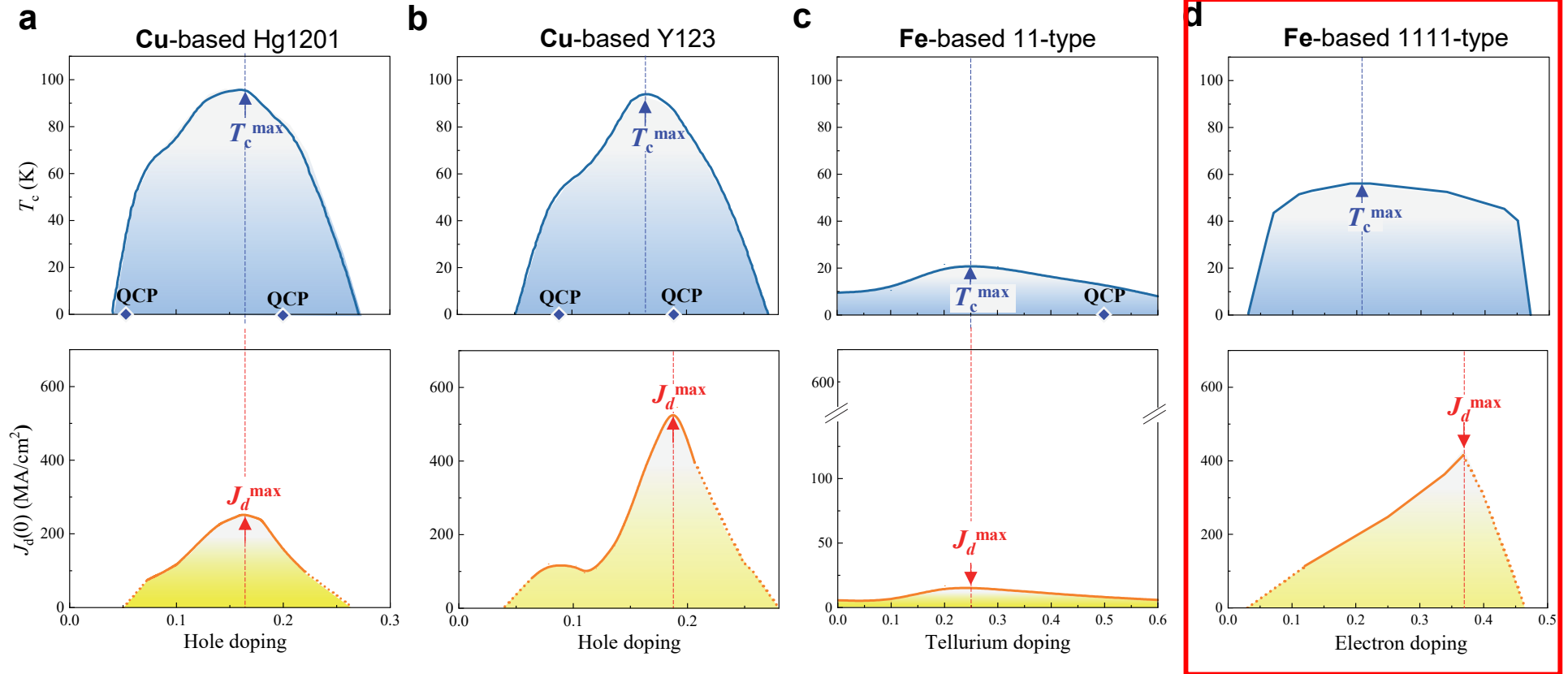
Mn-doping
($\text{Fe}_{1-x}\text{Mn}_x$, $x \sim 0.1$)

- Suppression of broadening
- Significant enhancement of F_p
- Decrease in T_c (42 K \rightarrow 37 K)
- Low RRR (42 \rightarrow 13)

Li et al., *Supercond. Sci. Technol.* **32**, 12LT01 (2019).

30

Thermodynamic approach + APC



The doping levels of $T_{c, \max}$ is different from that of $J_{d, \max}$

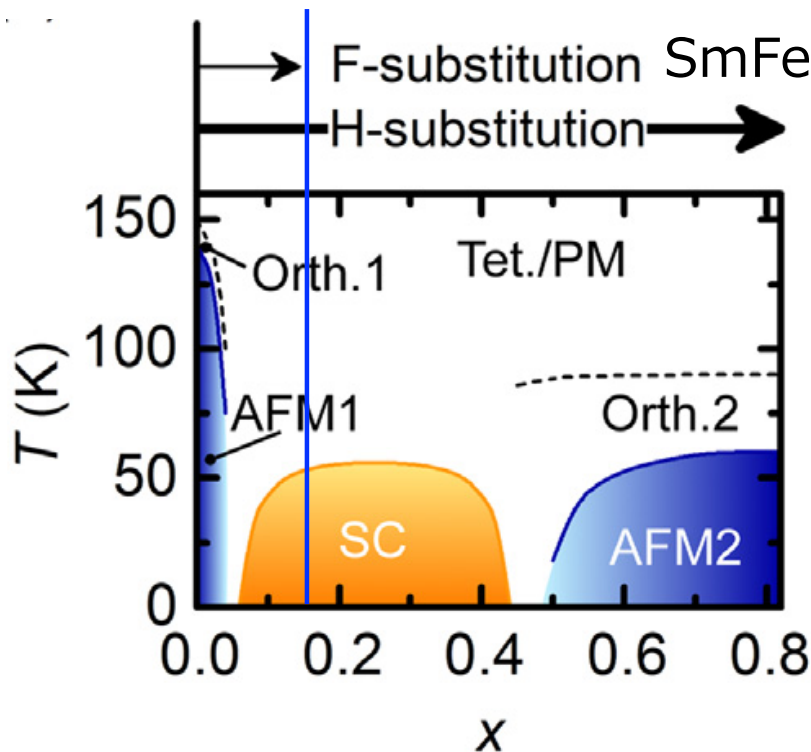
$$J_d(T) = \frac{\phi_0}{3\sqrt{3}\pi\mu_0\lambda^2(T)\xi(T)}$$

M. Miura, K. I. et al., Nature Materials (2024).

Solubility limit in SmFeAsO

$LnFeAsO$ (Ln : lanthanoide)

$O^{2-} \rightarrow F^-$ or $H^- + e^-$ (electron doping)



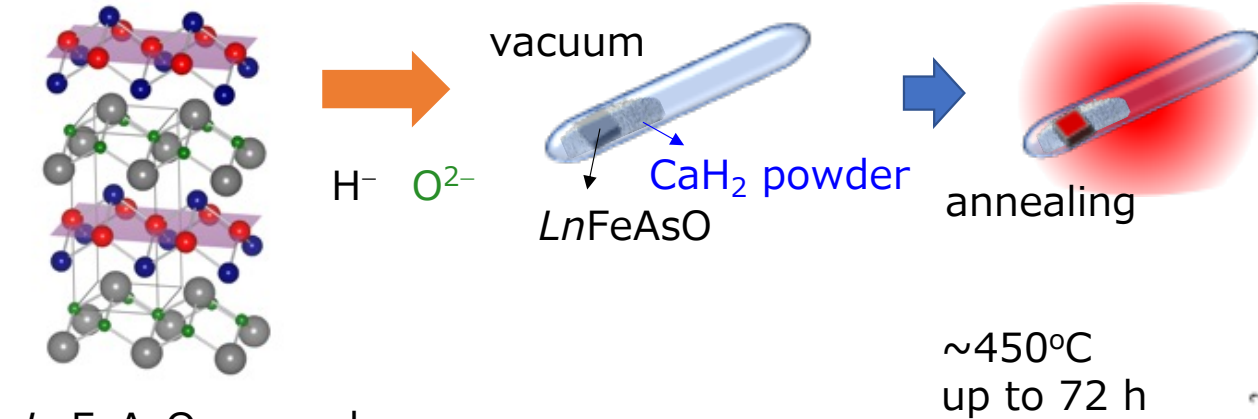
- Substitution level is limited **up to ~ 0.2** (For $SmFeAsO_{1-x}F_x$)
- For H, the substitution level is increased **up to ~ 0.8**
- Heavily electron doping can be achieved

S. Iimura *et al.*, *J. Asia Ceramic Societies* **5**, 357 (2017).

32

Thermodynamic approach + APC for SmFeAs(O,H)

Topotactic reaction



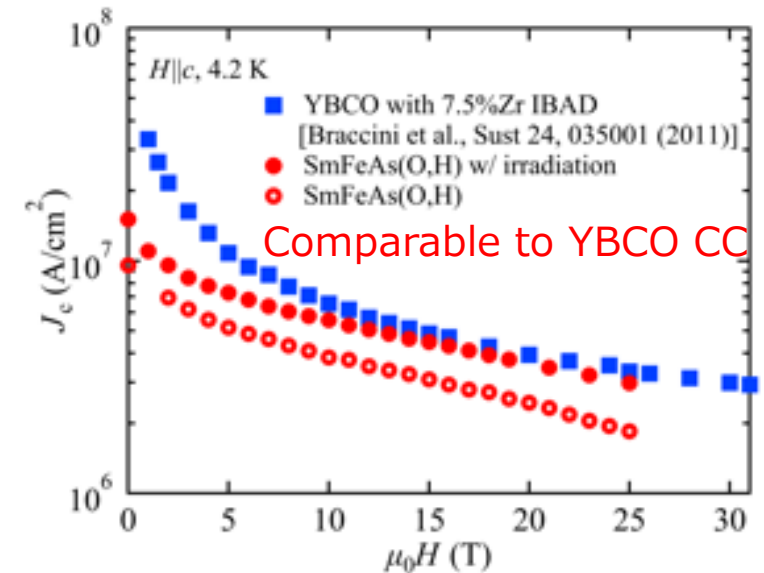
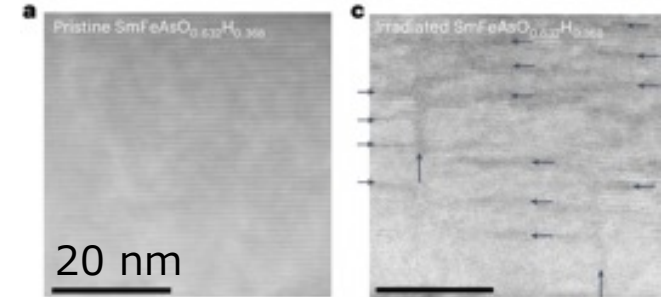
LnFeAsO grown by
PLD or MBE

K. Kondo, K.I. et al., *SuST* **33**, 09LT01 (2020).

~450°C
up to 72 h

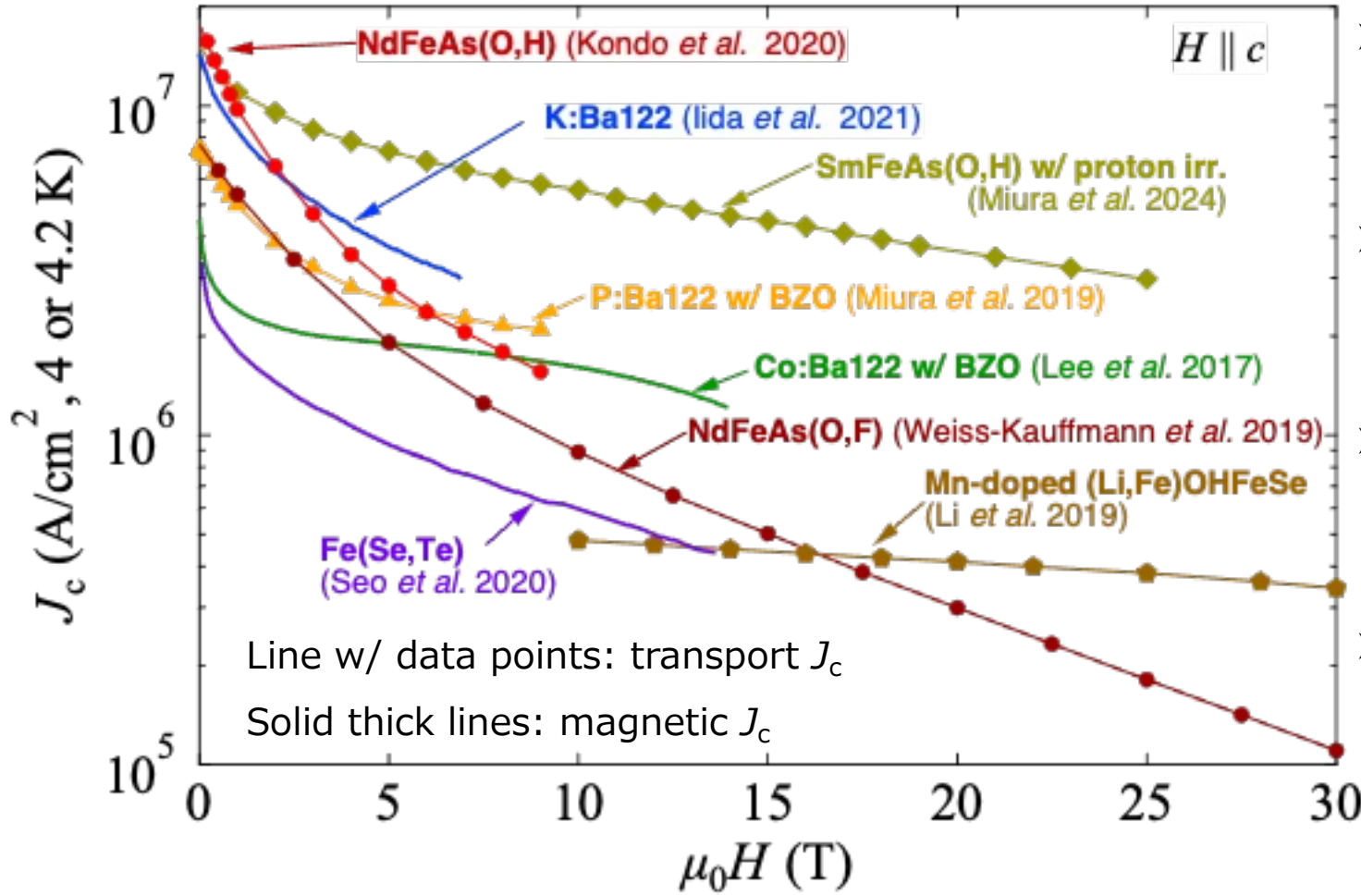
Pristine

Proton irradiation



M. Miura, K. I. et al., *Nature Materials* (2024).

Current status of the best-performing J_c - H (single xtal. sub.)



- Self-field J_c for all IBS exceed 3 MA/cm², even for Fe(Se,Te) with $T_c=20$ K
- The highest self-field J_c of over 17 MA/cm² was achieved with NdFeAs(O,H)
- In-field J_c was remarkably improved in SmFeAs(O,H) w/ proton irradiation
- Constant J_c of 0.4 MA/cm² up to 30 T for Mn-doped (Li,Fe)OHFeSe

Overview

1. Iron-based superconductors (IBSs)

- Physical properties

2. Tuning of the superconducting properties

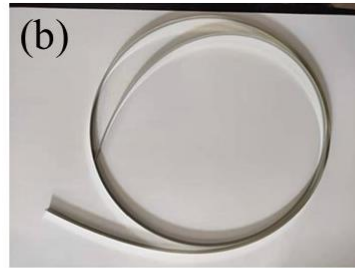
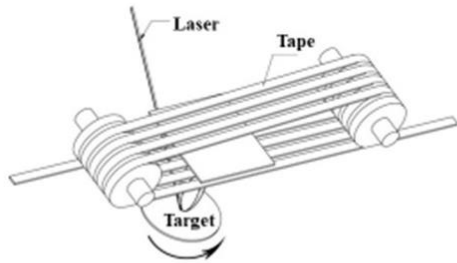
- SC transition temperature (strain, monolayer, intercalation, EDLT)
- Grain boundary, GB
- Critical current density (natural defects, APC, thermodynamic approach)

3. Progress Toward applications

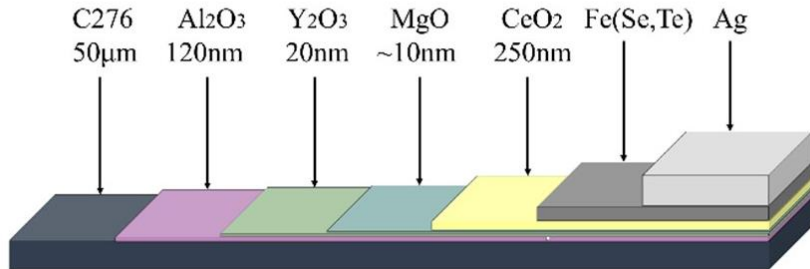
- Use of IBS wires and bulks in magnets, and perspective

Fe(Se,Te) tapes

1 m long Fe(Se,Te) coated conductor

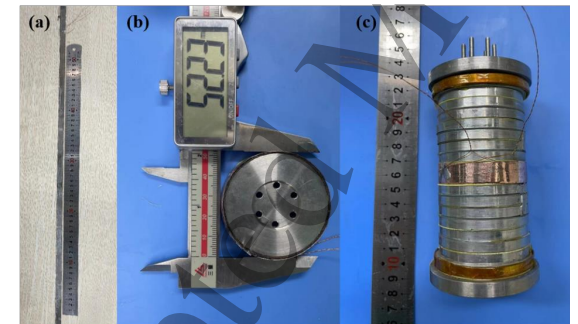
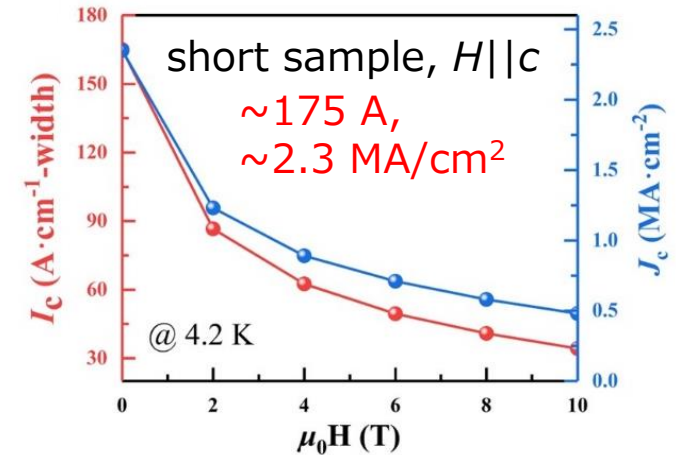
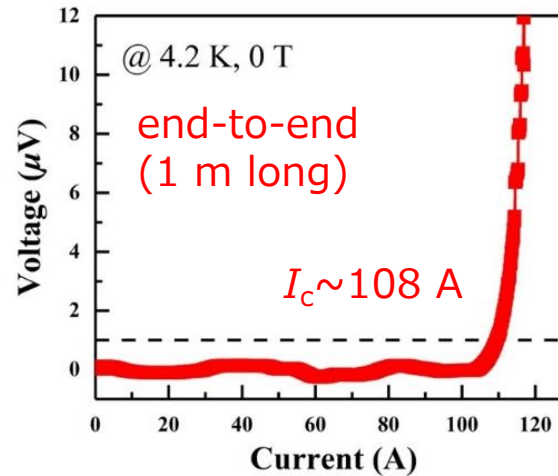


~320 nm



- multi turn PLD system, operating at $f=20-60$ Hz
- $\text{FeSe}_{0.4}\text{Te}_{0.6}$ target, $\phi = 60$ mm, $t = 8$ mm
- Self-field $I_c \sim 108$ A @ 4.2 K (1 m tape, $T_c = 17.5$ K)
- Short sample: Self field $I_c \sim 175$ A corresponding to $J_c \sim 2.3$ MA/cm²

L. Liu *et al.*, *Adv. Eng. Mater.* **25** 2201536 (2023).



S. Wei *et al.*, *Sust* **36** 04LT01 (2023).

Detailed in 2MOr2A-06,-07
by H. Liu & Y. Li

Fe(Se,Te) tapes

Realising a cheap, Fe(Se,Te) Coated Conductor

PRIN **TEAM HIBISCUS**

 **PRIN HIBISCUS**

 **M. Putti**, M. Iebole, P. Manfrinetti, M. Meinero, G. Sylva
University of Genova

 **V. Braccini**, E. Bellingeri, C. Bernini, M. Cialone, G. Grimaldi, A. Leo, M. Lisitskiy,
A. Malagoli, N. Manca, A. Martinelli, I. Pallecchi, L. Piperno, A. Provino
CNR - SPIN

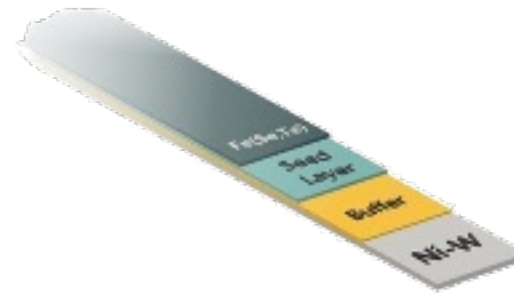
 **E. Silva**, A. Alimenti, N. Pompeo, K. Torokhtii
University of Roma Tre

 **L. Gozzelino**, M. Fracasso, R. Gerbaldo, G. Ghigo, F. Laviano, A. Napolitano, D.
Torsello
Politecnico of Torino

 **G. Celentano**, A. Vannozzi, A. Augieri, A. Angrisani, A. Mancini, A. Rufoloni
ENEA Frascati

courtesy by M. Putti & V. Braccini

- **Thick films** of Fe(Se,Te) by e-depo



Reported by L. Piperno
(1MOrB-02, 2/9)

- **Irradiation effects** on Fe(Se,Te) films

Reported by M. Iebole & F. Rizzo
(1MOr1B-03, -04, 2/9)

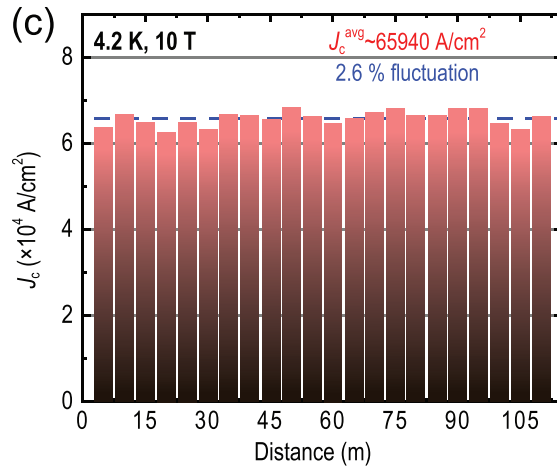
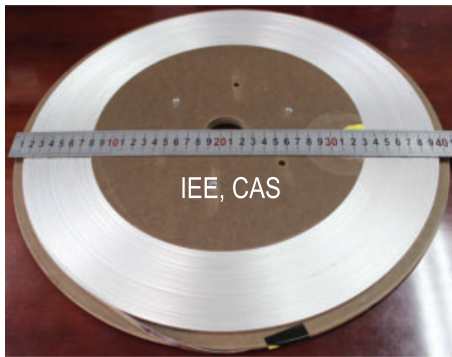
- **Pinning mechanism**

Microwave vortex motion in FeSe

Reported by N. Pompeo
(1MOr1B-05, 2/9)

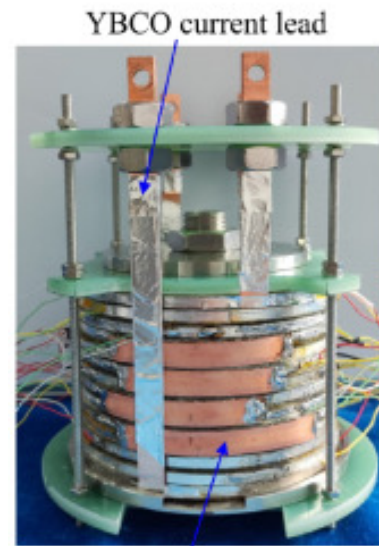
(Ba,K)122 wires and magnets

100 m-long, 7-core Ba122 tapes

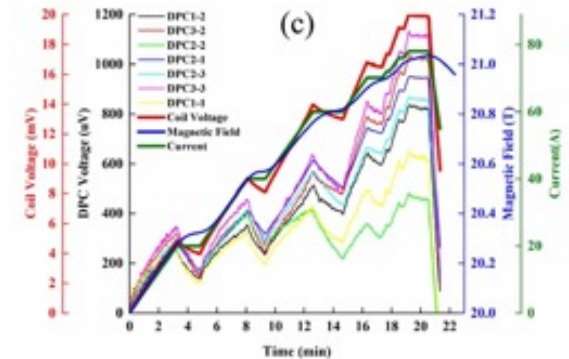


C. Dong et al., *National Science Rev.* **11** nwae122 (2024).

(Ba,K)122 high field insert coil



Measured field: 21.03 T
Background field: 20 T



H. Ding et al., *Sust* **36** 11LT0101 (2023).

Double Pancake Coils



7 DPCs

- DPC 1-1 $I_c \geq 100\text{A}$
- DPC 3-3 $I_c \geq 85\text{A}$
- DPC 2-3 $I_c \geq 90\text{A}$
- DPC 2-1 $I_c \geq 100\text{A}$
- DPC 2-2 $I_c \geq 90\text{A}$
- DPC 3-2 $I_c \geq 85\text{A}$
- DPC 1-2 $I_c \geq 100\text{A}$

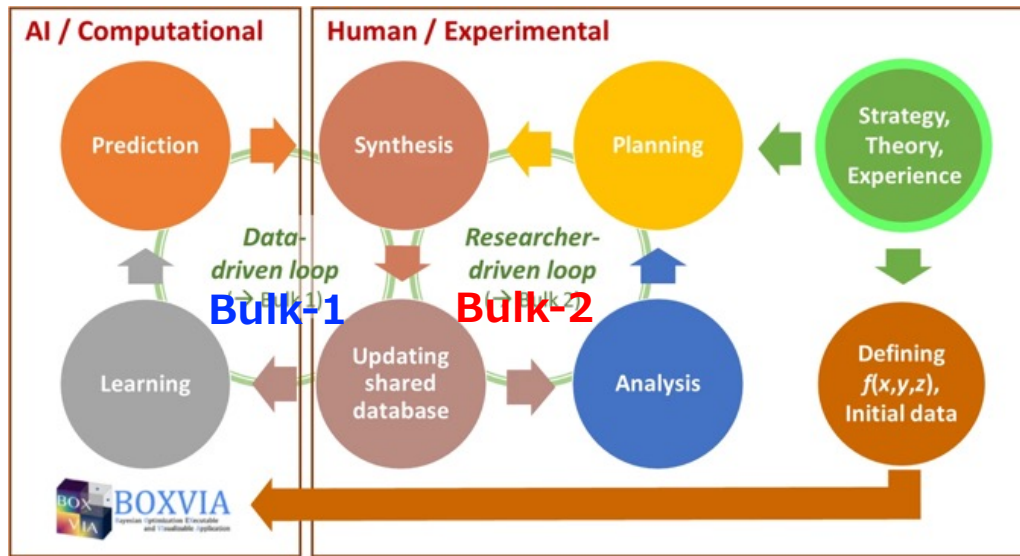
Courtesy by Prof. Dr. Yanwei Ma

World first Tesla class coil using IBSs

Detailed in 5MOr1B-04
by C. Dong

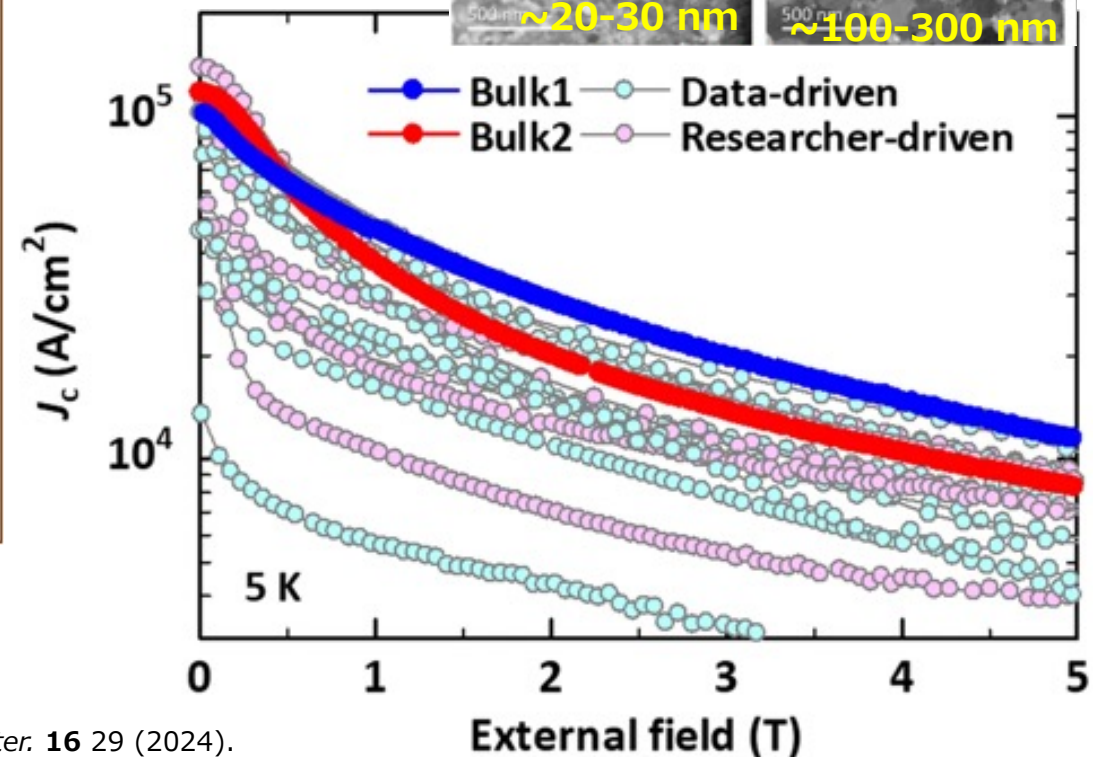
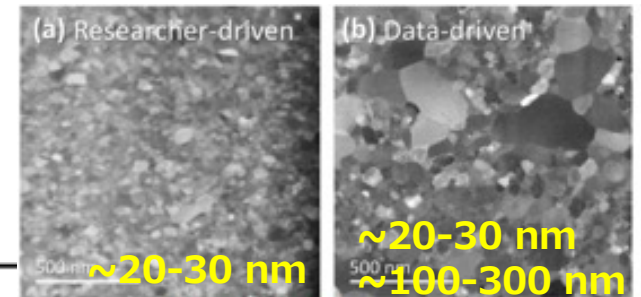
(Ba,K)122 bulks fabricated by data- & researcher driven process design

Conceptual schematic of the complementary data- and researcher-driven process designs



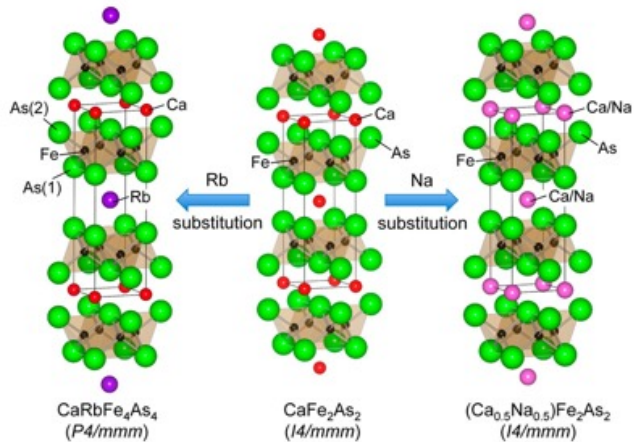
Detailed in 2MOr2A-01
by A. Yamamoto

A. Yamamoto *et al.*, *NPG Asia Mater.* **16** 29 (2024).

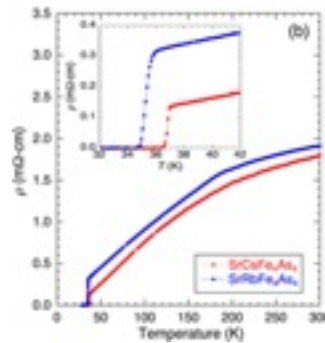


CaKFe₄As₄ (1144) bulk samples

Hybrid phase between AeFe₂As₂ (Ae = Ca, Sr) and AFe₂As₂ (A=K, Rb, Cs)

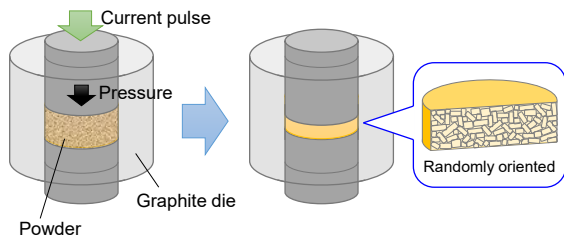


$T_c \sim 36$ K w/o doping

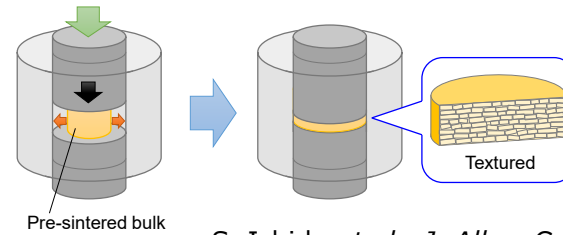


A. Iyo et al., *J. Am. Chem. Soc.* **138** 3410 (2016).

(a) Spark Plasma Sintering (SPS)

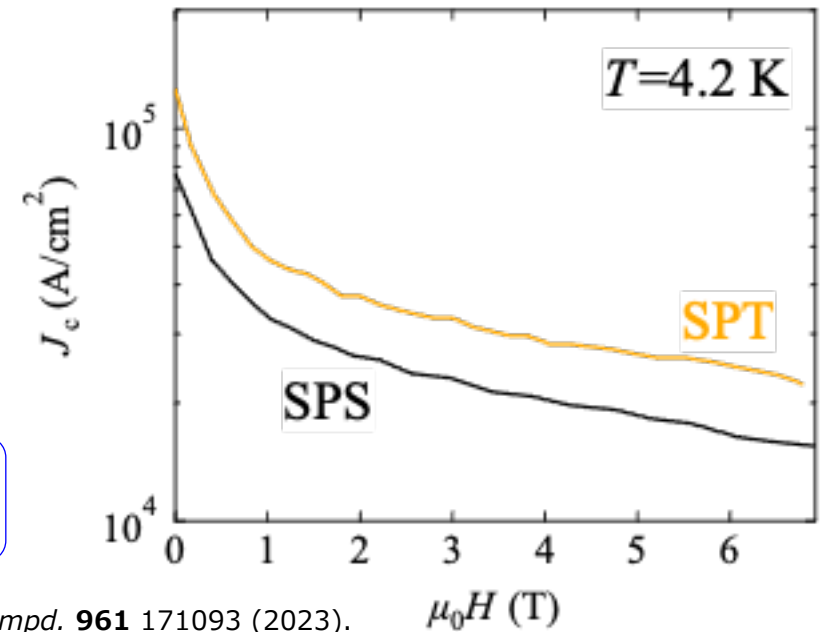


(b) Spark Plasma Texturing (SPT)

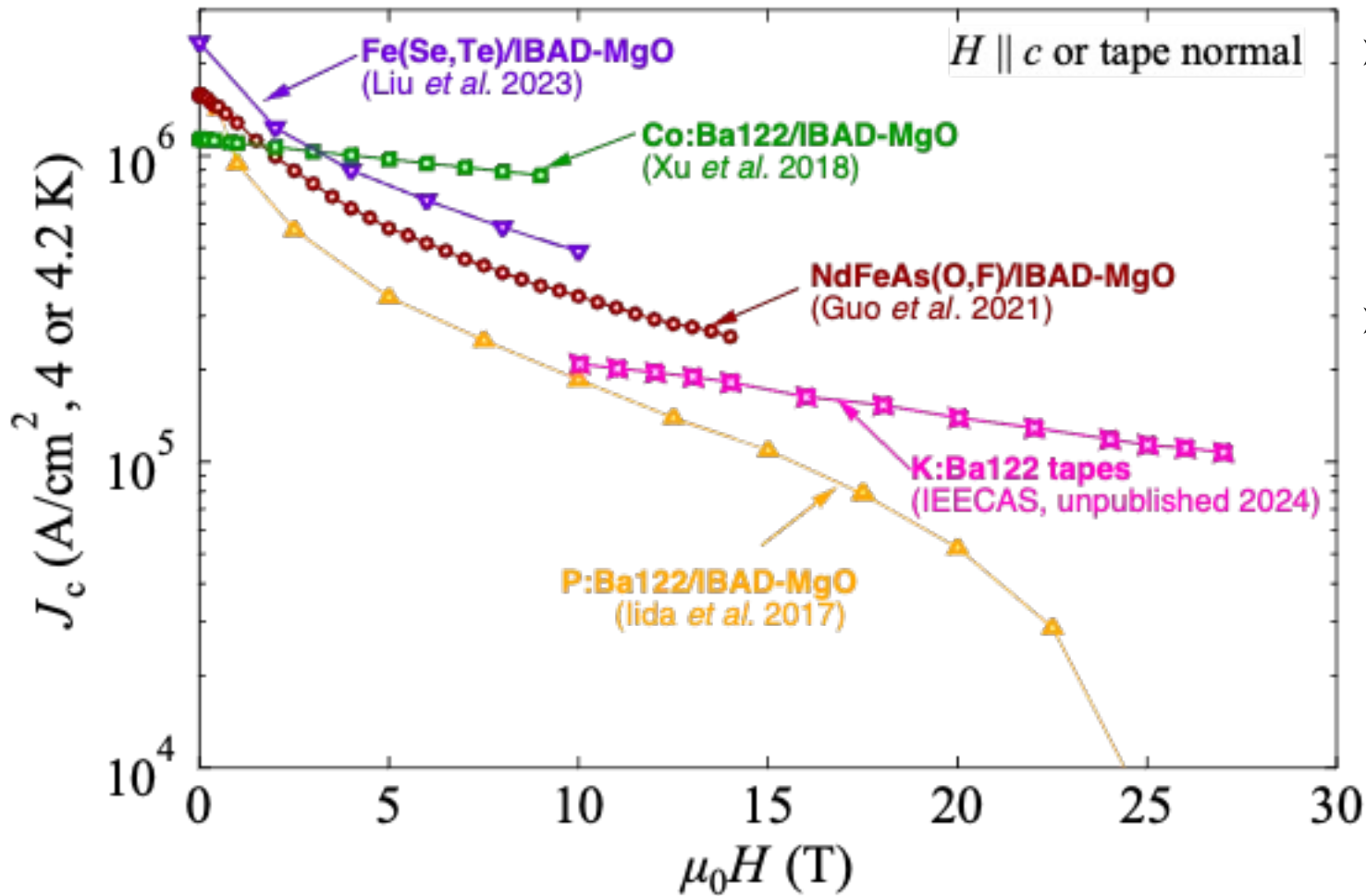


S. Ishida et al., *J. Alloy. Compd.* **961** 171093 (2023).

- The c-axis textured 1144 was realized
- A self-field J_c of reached 12 kA/cm², which is comparable to that of K:Ba122 bulks

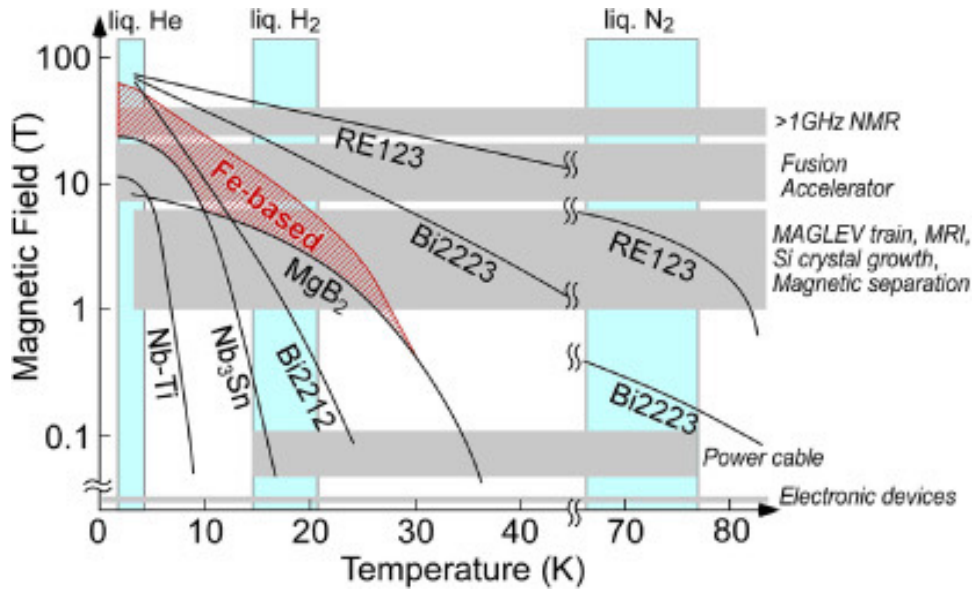


Current status of J_c - H plots for IBs tapes (short samples)



- Fe(Se,Te)/IBAD-MgO showed the highest J_c in low field regime although T_c is the lowest
- In-field J_c of K-doped Ba122 tape was superior to that of the P-doped Ba122/IBAD-MgO

Perspective (still mature... but)



J. Shimoyama *Sust* **27** 044002 (2014).

>20 T at 4.2 K or >10 T at 20-30 K



Viewpoint J. Jaroszynski

Constructing high field magnets is a real tour de force

Moreover, the cost of IBS wire can be four to five times lower than that of Nb₃Sn, making it more expensive than NbTi, but with much higher critical parameters than Nb₃Sn. Attempts to make a superconducting wire started immediately, using either the powder-in-tube (PIT) [11–13] or coated conductor [14, 15] methods.

The cost of IBS wire can be four to five times lower than that of Nb₃Sn, making it more expensive than NbTi, but with much higher critical parameters than Nb₃Sn.

Summary

1. A review of the current status of IBSs has been conducted.
2. A review of various techniques for tuning T_c has been conducted.
3. High-angle grain boundaries (GBs) do block supercurrent flow, but not as severely as in the cuprates. This is a driving force for magnet applications using polycrystalline wires and bulk materials.
4. A strategy for improving the polycrystalline tapes and bulks of K-doped Ba122 has been proposed.
5. J_c - B performances have been improved significantly by APC and thermodynamic approach combined with APC (films).
6. Long length wires and tapes have been developed significantly.

Thank you for your attention!



Iron-based Superconductors: Advances towards applications

Phoenix Seagaia Resort
Convention Center (Miyazaki, Japan)
February 13th – 15th, 2025



<https://smartconf.jp/content/ibs2app>