

$\text{YBa}_2\text{Cu}_3\text{O}_{7-x}$ film with $\text{Ba}_2\text{Y}(\text{Nb},\text{Ta})\text{O}_6$ nano- inclusions for high field applications

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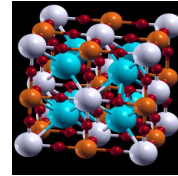


G. Celentano - Glasgow, September 3rd 2019

OUTLINE

Introduction

- Interest in high field applications for REBCO;
- Nb- and Ta-based APC for REBCO: state-of-the-art

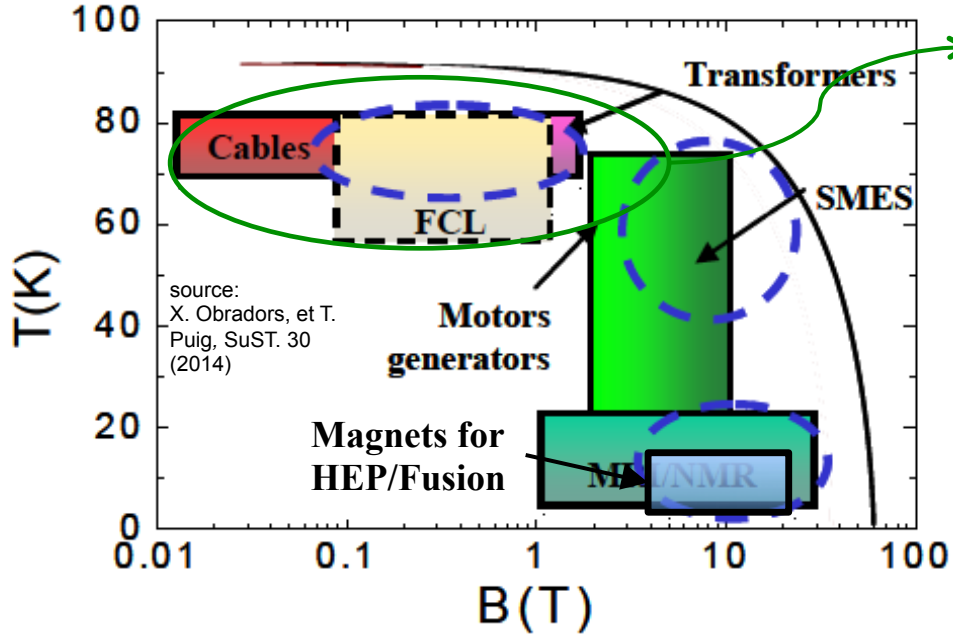


Mixed doping by $\text{Ba}_2(\text{Y,Ta})\text{O}_6 + \text{Ba}_2(\text{Y,Nb})\text{O}_6$: 2.5 mol.% + 2.5 mol.%

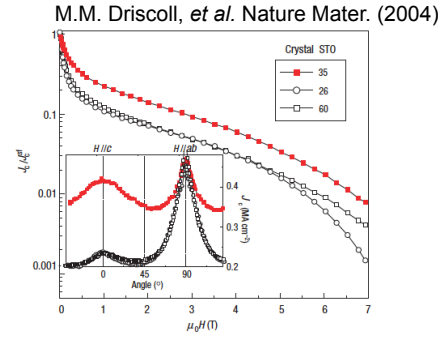
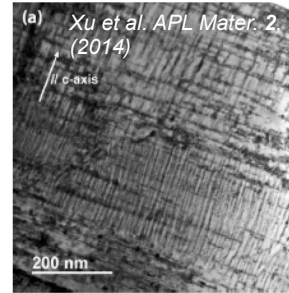
- BYTO single vs mixed BYTO + BYNO doping;
- Defects landscape tunability by growth rate;

Conclusions & Perspectives

Intro – High fields: new perspectives of REBCO applications



Effective technology for control of pinning and J_c optimization @ LN2 and low/mid field



APC by incorporation of $BaMO_3$ (M= Zr, Hf, Sn)
Self-assembled columnar structures/c – axis correlated defects

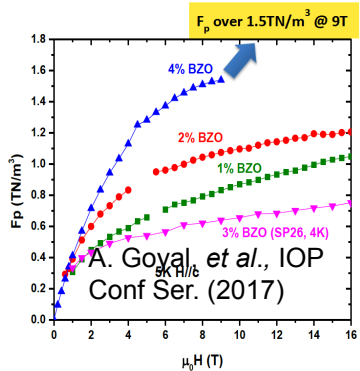
Nuclear fusion and accelerators requests are extremely demanding

Nb₃Sn technology cannot fulfill these needs →

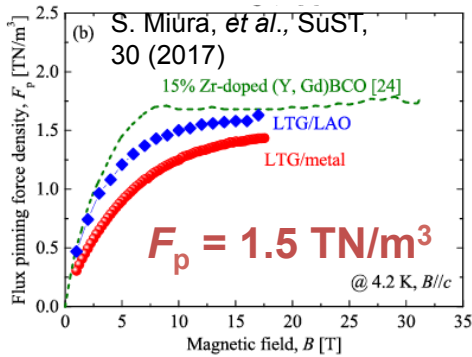
For **REBCO** this is a unique opportunity to extend its applicability to high field magnets sector

Intro – What we know @ Low T / High field conditions for REBCO

APC approach is still effective: APC + additional defects spontaneously generated by APC/YBCO interfacial strain accommodation

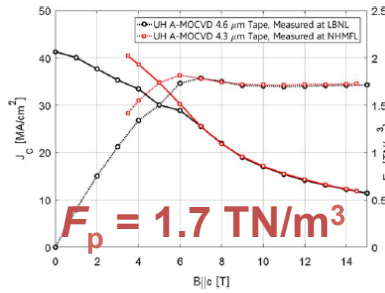


PLD-YBCO with BZO

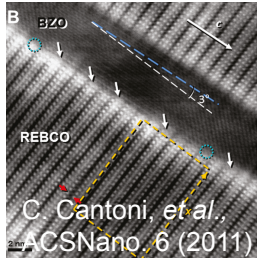
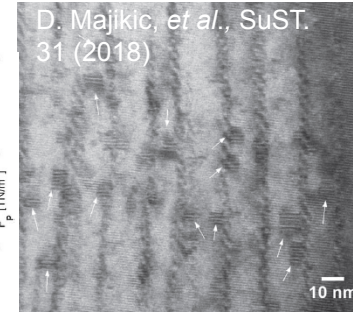


PLD-SmBCO with BHO

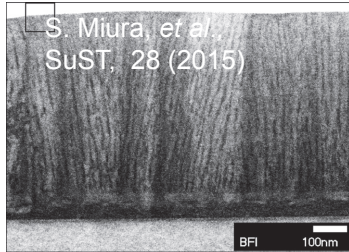
MOCVD-(Y,Gd)BCO with Zr



Very thin rods/columns + segregated RE₂O₃



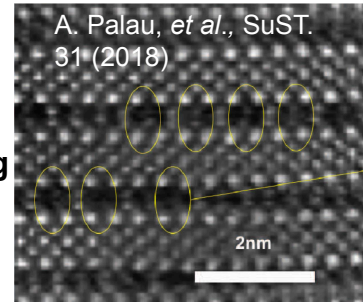
Oxygen disorder in YBCO at interface



Careful control of rods size, density and inclination distribution

Cluster of Cu-O vacancies in stacking faults

CSD-YBCO with BZO np

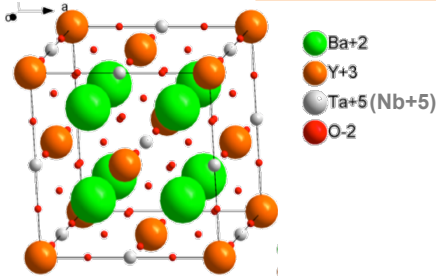


Little is known so far:
 REBCO poorly investigated in low T/high field conditions
 More studies are needed

Intro: Ba_2YTaO_6 and Ba_2YNbO_6 doping: great performances at LN2



Double perovskite; cubic - Fm3m; $a = 0.84$ nm;



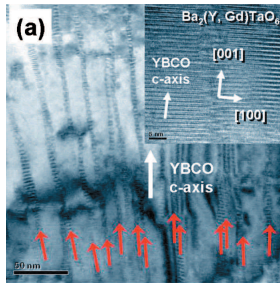
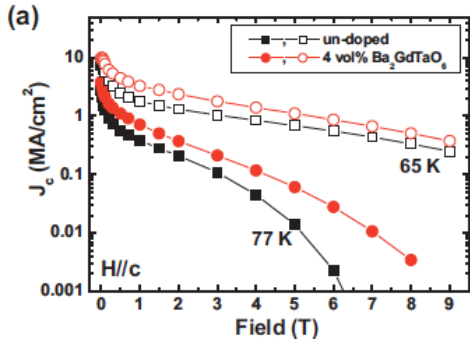
- Great phase stability (**chemically inert w.r.t. YBCO**);
- large mismatch w.r.t. YBCO: in plane $\sim 9.4\%$; c-axis $\sim 8.3\%$;
- Nb⁵⁺ = Ta⁵⁺ = 0.78 Å (Zr⁴⁺ = 0.86 Å)
- Ta₂O₅ (Nb₂O₅) lower melting T than ZrO₂;
- **larger ion mobility than Zr is expected;**

G. Ercolano et al. SuST 23 (2010)
 G. Wee et al. PRB 81 (2010)
 G. Ercolano et al. SuST 24 (2011)



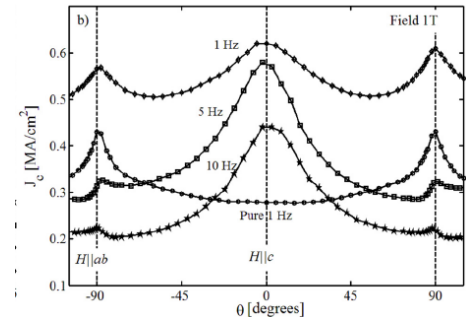
G. Wee et al. PRB 81 (2010)

G. Ercolano et al. JAP (2014)



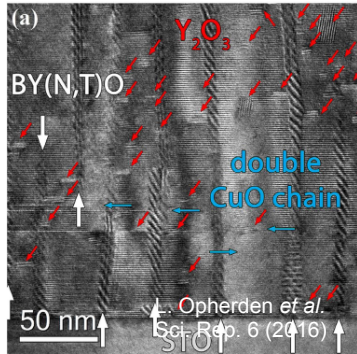
Splayed columns,
dense and fine structure
 $d \approx 6 - 7$ nm,
 $B_\phi \approx 5 - 9$ T

dense and fine nanorods
 $d \approx 10$ nm, $B_\phi \approx 2 - 5$ T



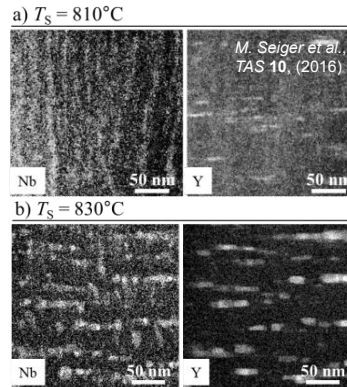
Intro: Nb and Ta- based double doping

More complex defect landscape **Mixed $Ba_2YNbO_6 + Ba_2YTaO_6$ doping**

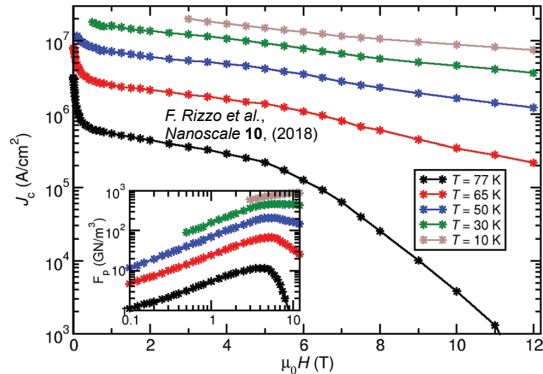
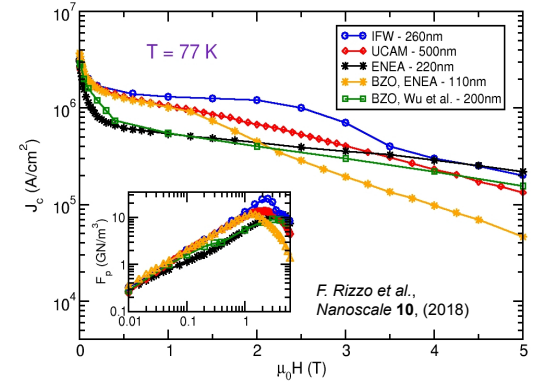


Complex landscape:
 $Ba_2Y(Nb,Ta)O_6$
 continuous columns + Y_2O_3 nanoparticles

Tunable: switch to plate-like defects



Reproducible: similar performance for samples grown in different Labs



Good T and B - behavior: $Ba_2Y(Nb,Ta)O_6$ promising for low T and high fields regimes

OUTLINE

Mixed doping by $\text{Ba}_2(\text{Y,Ta})\text{O}_6$ + $\text{Ba}_2(\text{Y,Nb})\text{O}_6$: 2.5 mol.% + 2.5 mol.%

- BYTO single vs mixed BYTO + BYNO doping;
- Defects landscape tunability by growth rate;

PLD growth of $\text{YBa}_2\text{Cu}_3\text{O}_{7-x}$ + tantalate/niobiate based APCs

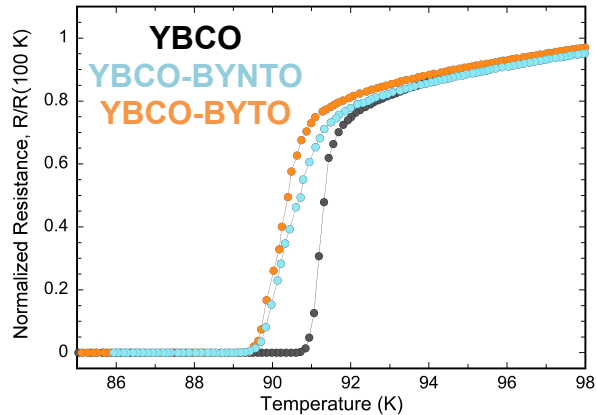
Single (BYTO) vs balanced mixed doping

YBCO composite targets

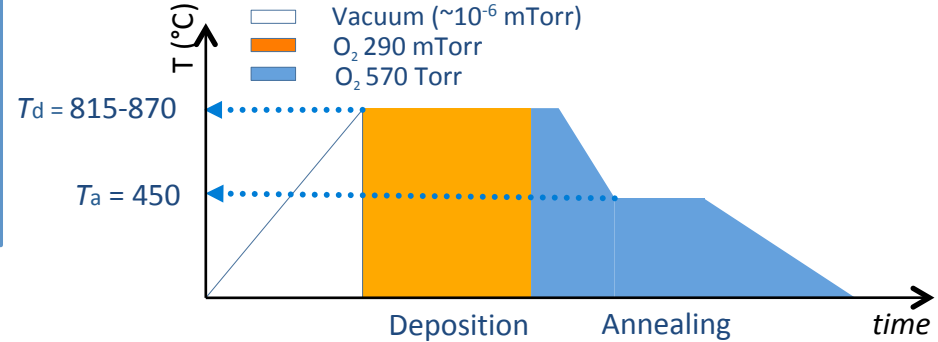
YBCO : Ba_2YTaO_6 5 mol. %

YBCO : Ba_2YTaO_6 2.5 mol. % + Ba_2YNbO_6 2.5 mol.

Best $T_c \approx 89 \text{ K}$ @ $T_d = 840 \text{ }^\circ\text{C}$



Film Deposition conditions

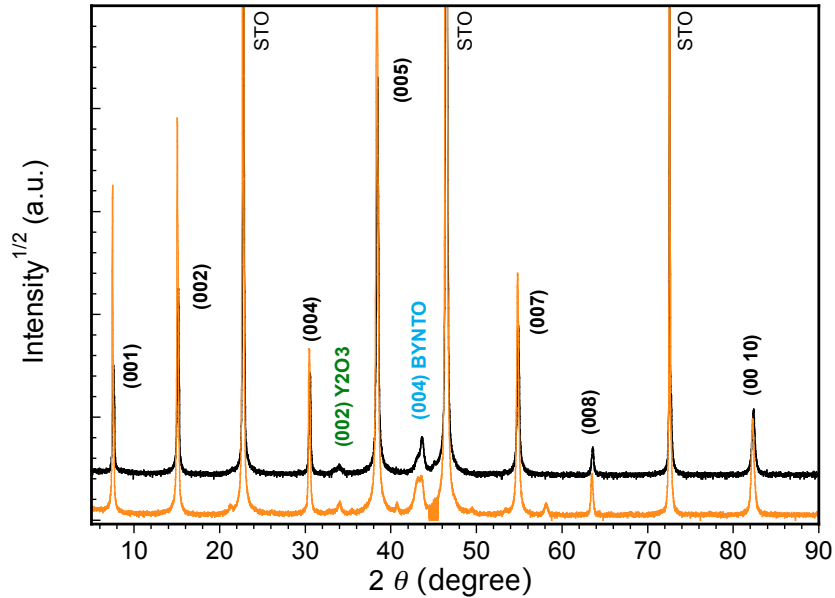


Film thickness $\approx 200 \text{ nm}$
Substrate: (001) SrTiO_3
Growth rate, $\rho \approx 0.3 \text{ nm s}^{-1}$

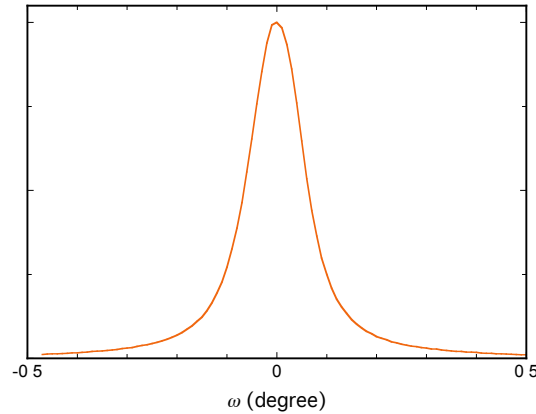
PLD setup

XeCl Excimer Laser
 $\lambda = 308 \text{ nm}$
 $f_L = 10 \text{ Hz}$
fluence $\approx 1.5 - 2 \text{ J/cm}^2$

BYTO 2.5 mol.% + BYNO 2.5 mol.% / BYTO 5 mol.%: structural properties



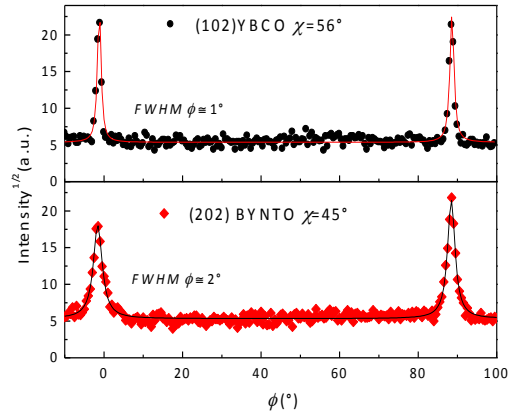
- good YBCO epitaxial growth
- BYNTO grows cube-on-cube with YBCO
- presence of Y_2O_3 related peaks



FWHM

RC (005) YBCO = 0.12°

RC (004) BYNTO = 0.6°

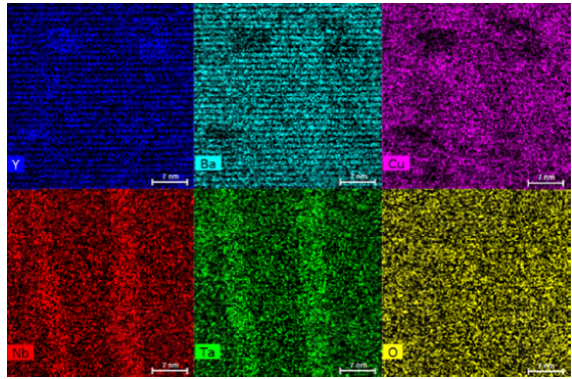


FWHM

ϕ - scan YBCO (102) = 1°

ϕ - scan BYNTO (202) = 2°

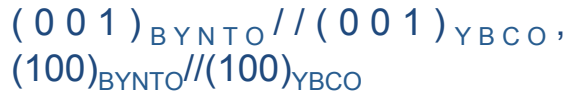
BYTO 2.5 mol.% + BYNO 2.5 mol.%: TEM/EDX



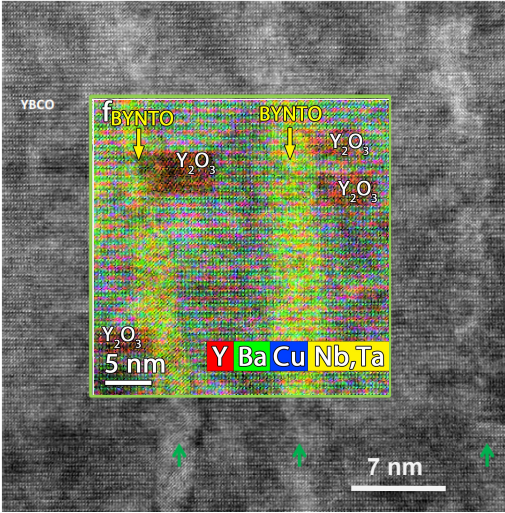
EDX elemental maps:

Excess of Ta and Nb is present in the columns

- Continue splayed columns are present



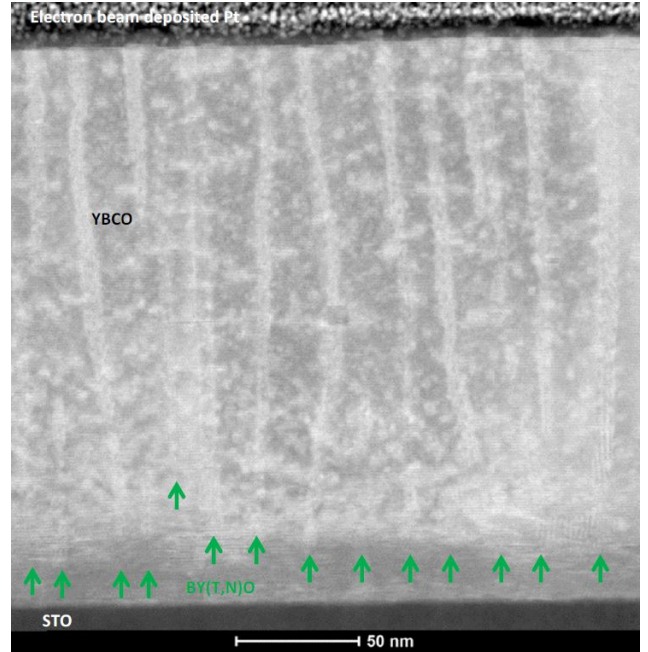
- High density of Y_2O_3 nanoparticles is recognized



Y_2O_3 nanoparticles

- size ≤ 10 nm;

- structural relationship: $(001)_{\text{Y}_2\text{O}_3} // (001)_{\text{YBCO}}, (110)_{\text{Y}_2\text{O}_3} // (100)_{\text{YBCO}}$

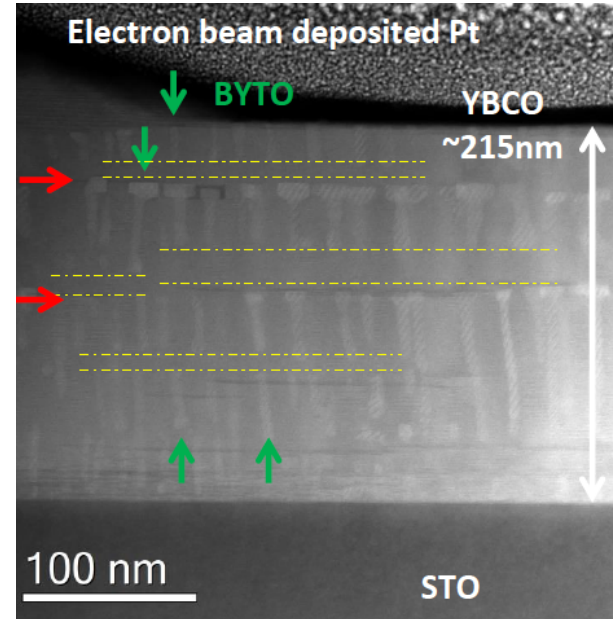
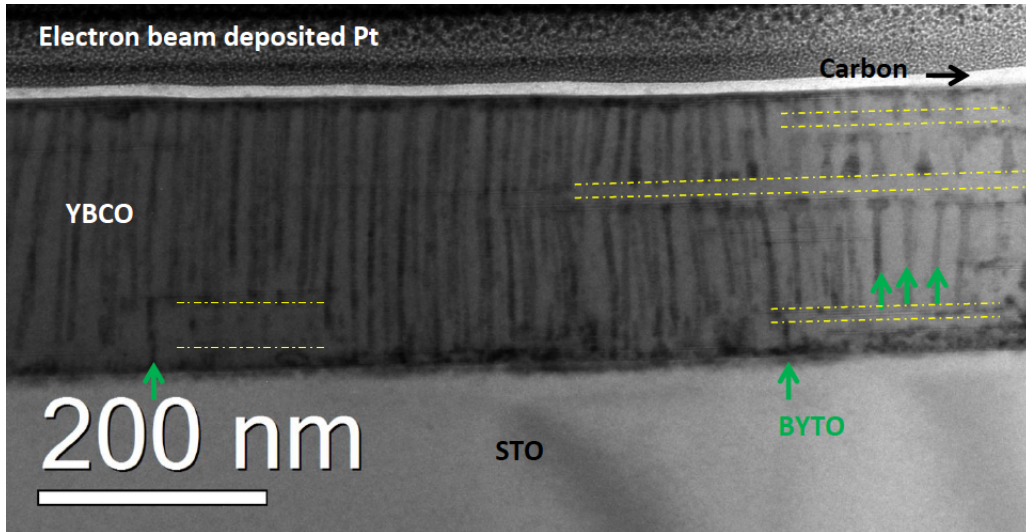


Column size $d \sim 5$ nm
Inter-column distance ~ 20 nm
($n \sim 2500 \mu\text{m}^{-2}$, $B_\phi \approx 5.2$ T)

BYTO 5 mol.%: TEM/EDX

Two type of columnar structures:

- **continuous** through the full YBCO thickness
- and **truncated**, some of them with hammerhead



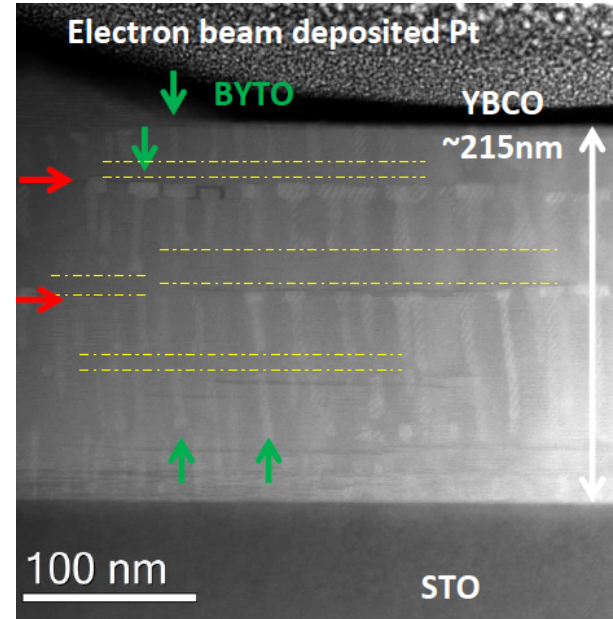
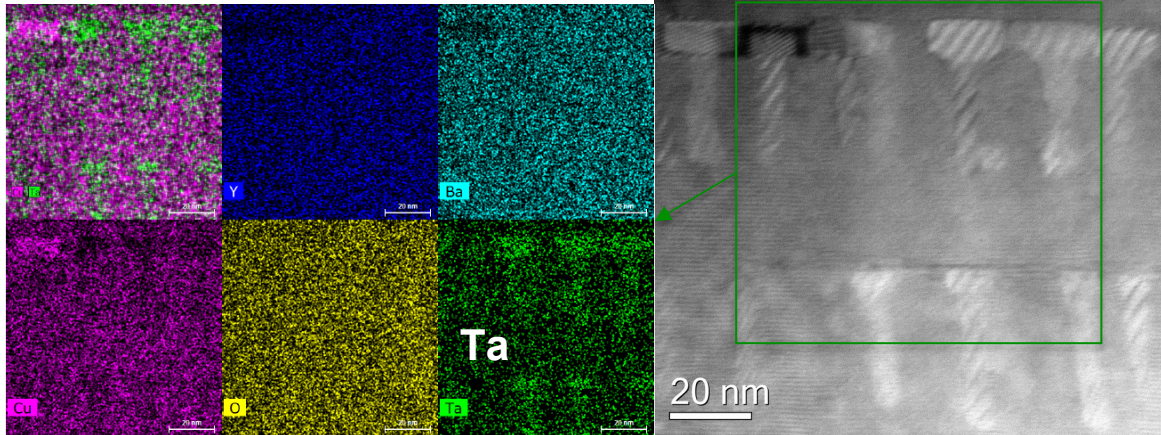
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BYTO 5 mol.%: TEM/EDX

Two type of columnar structures:

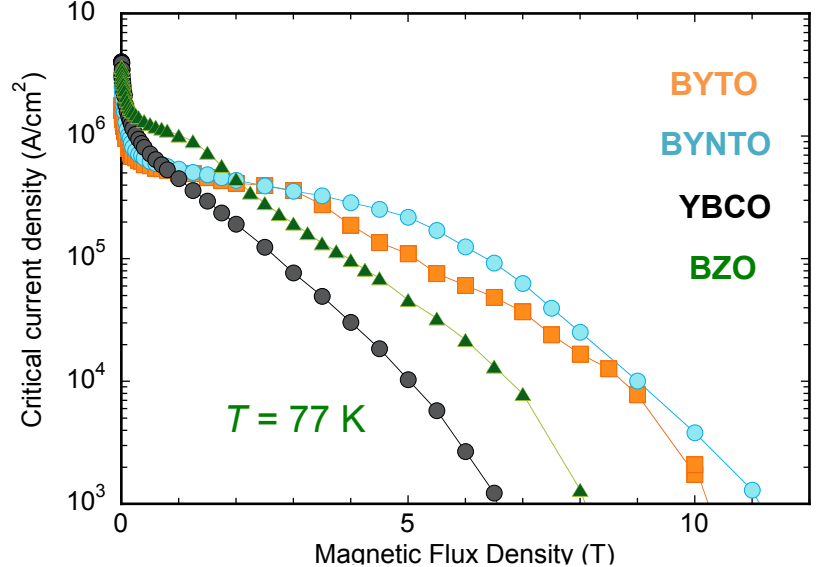
- **continuous** through the full YBCO thickness
- and **truncated**, some of them with hammerhead

EDX elemental maps:
Excess of Ta is present in the column hammerheads



Column size $d \sim 5$ nm
Inter-column distance ~ 20 nm
($n \sim 2500 \mu\text{m}^{-2}$, $B_{\phi} \approx 5.2$ T)

BYTO 2.5 mol.% + BYNO 2.5 mol.% / BYTO 5 mol.%: J_c behaviour @ 77 K

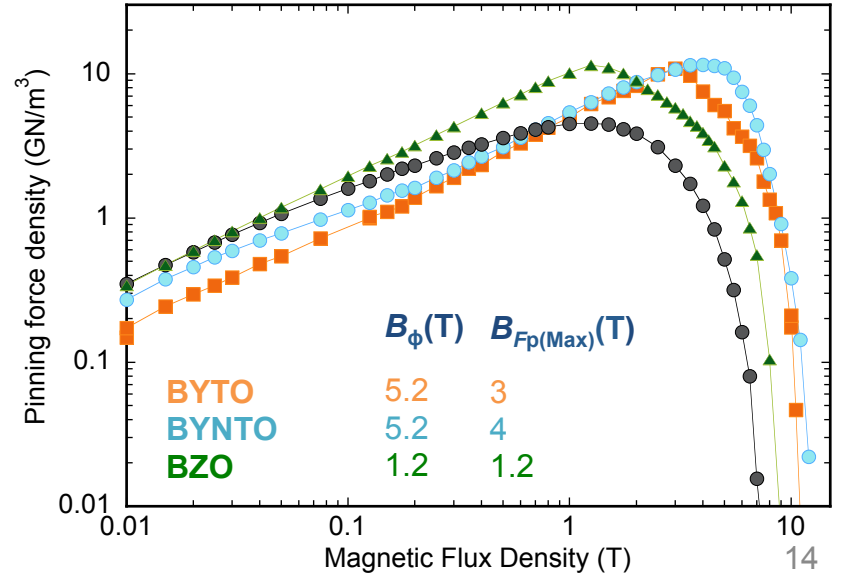


77 K	BYTO	BYNTO	YBCO	BZO
$J_c(0)$	1.6	3.2	4.0	3.7
$\alpha, J_c \approx B^{-\alpha}$	0.23	0.28	0.44	0.27
B_{irr}	10.2	11.1	6.7	8
$F_p(max)$	10.9	11.5	4.5	11.4

Comparison with YBCO and YBCO + 5 mol.% BZO deposited by PLD in similar conditions

$d_{BZO} \approx 5 - 7\text{ nm}$, $n_{BZO} \approx 600\ \mu\text{m}^{-2}$ ($B_\phi \approx 1.2\text{ T}$)
 A. Augieri, et al., JAP. 108 (2010)

BYNTO best mid field performances @ 77 K

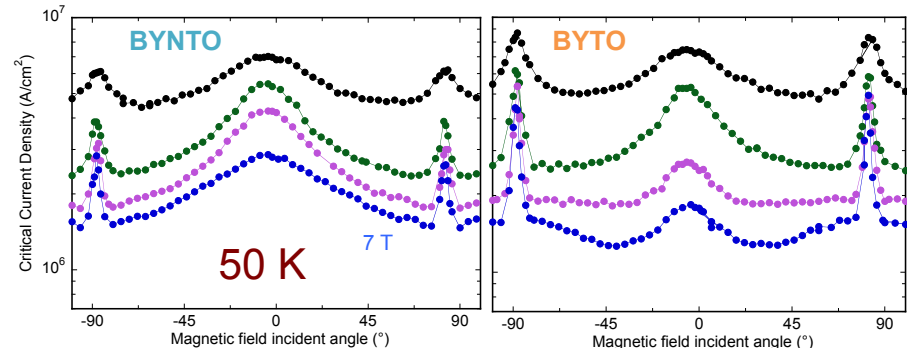
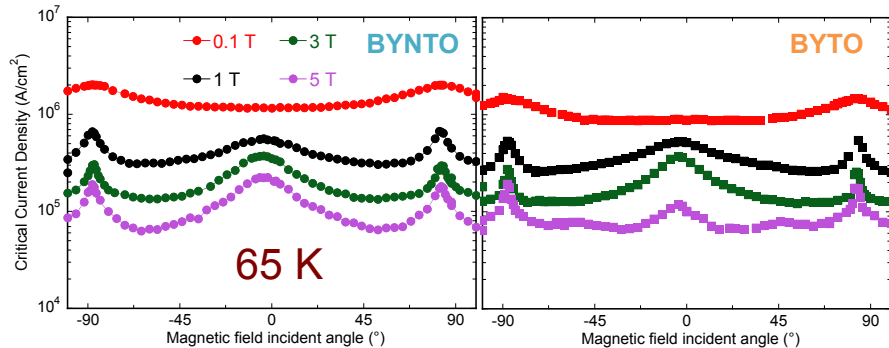
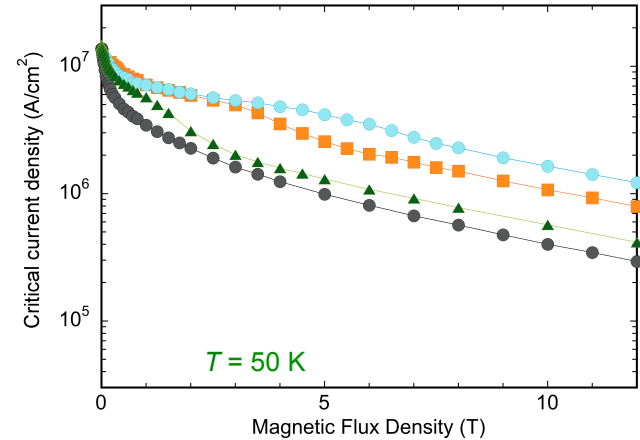
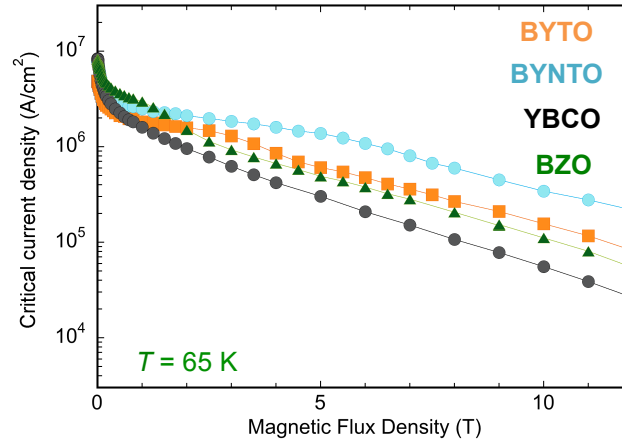


BYTO 2.5 mol.% + BYNO 2.5 mol.% / BYTO 5 mol.%: J_c behaviour @ intermediate T

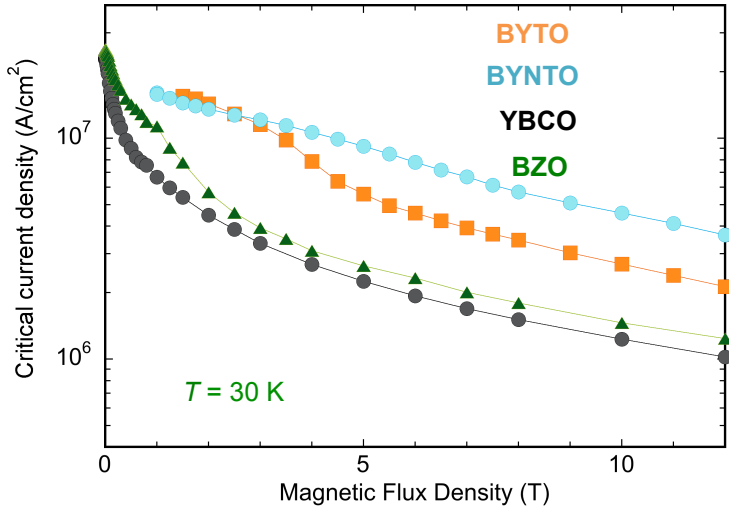
- Both **BYTO** and **BYNTO** improve YBCO
- **BYNTO** shows a stronger J_c retention in high field

$J_c(\theta)$: **BYNTO** is more efficient

- Broader peaks and more intense at higher fields

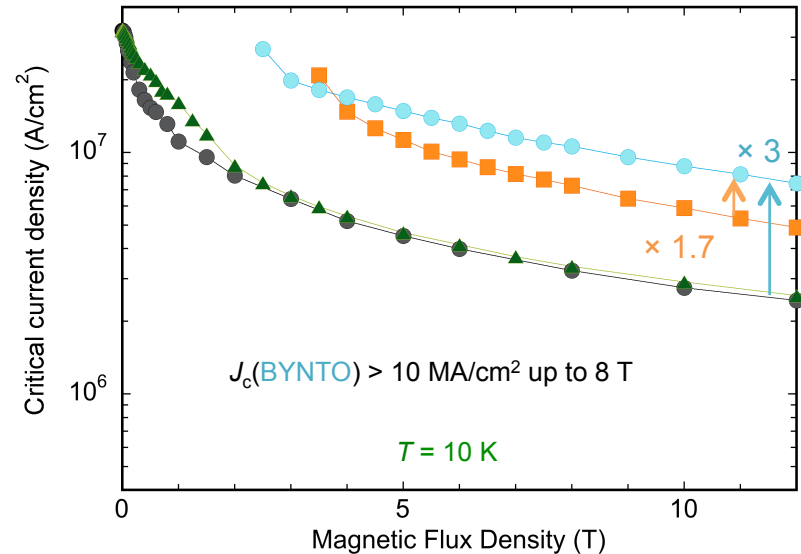


BYTO 2.5 mol.% + BYNO 2.5 mol.% / BYTO 5 mol.%: J_c behaviour @ low T

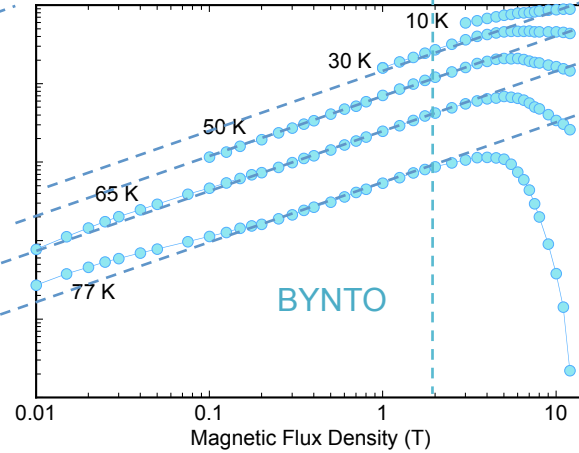
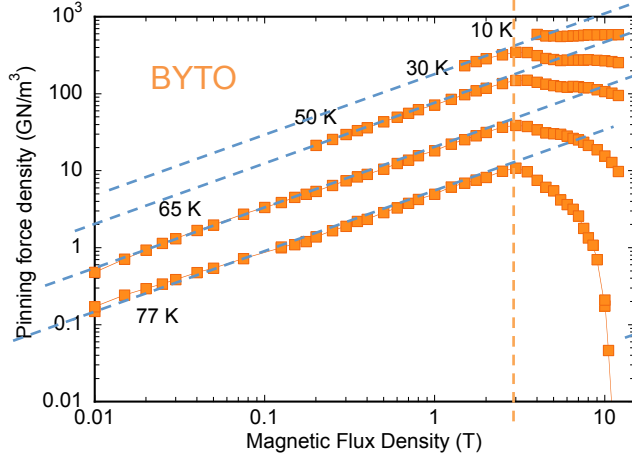


- BYNTO & BYTO similar low field behaviour (up to $B \approx 3 - 4$ T)
- BYNTO has **better high-field** behaviour than BYTO (@ 10 K, 12 T $J_c(\text{BYNTO}) = 1.7 \times J_c(\text{BYTO})$)

Both **BYTO** and **BYNTO** largely improve YBCO performances in **whole T - and B - range**
(@ 10 K, 12 T $J_c(\text{BYNTO}) = 3 \times J_c(\text{YBCO})$)



BYTO 2.5 mol.% + BYNO 2.5 mol.% / BYTO 5 mol.%. Pinning Force Density, F_p



power law exponents
are **T -independent**

Matching field effect

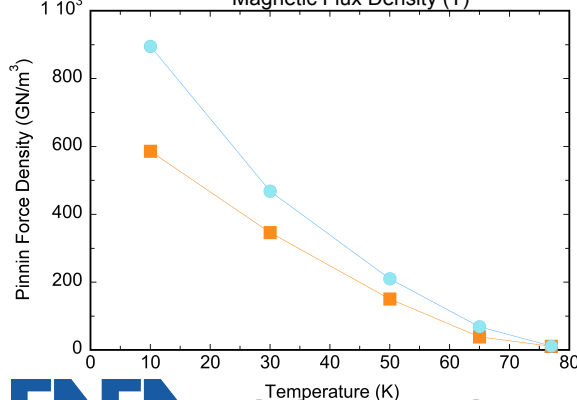
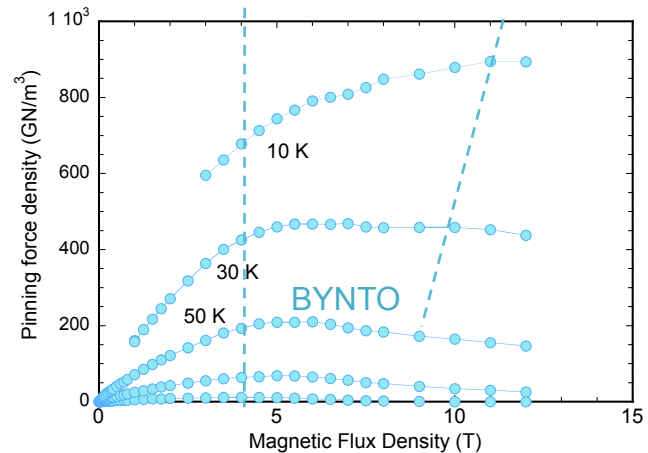
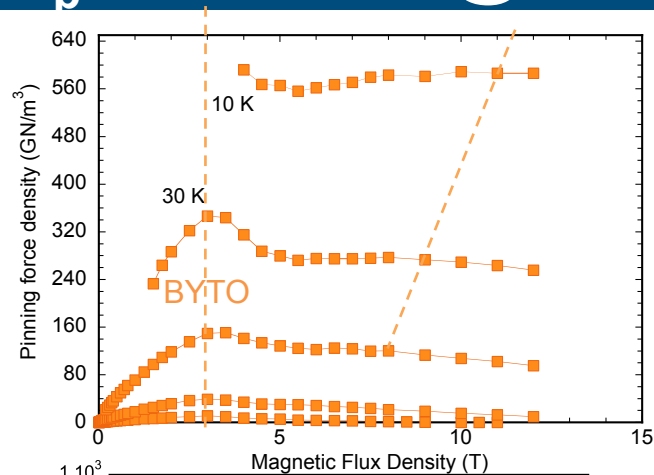
BYTO/BYNTO
columnar
systems is
effective in the
whole T -range
and dominate for
 B [0 – B_ϕ]

BYTO	77 K	65 K	50 K	30 K	10 K
$\alpha, J_c \approx B^{-\alpha}$	0.23	0.26	0.22	0.26	---

BYNTO	77 K	65 K	50 K	30 K	10 K
$\alpha, J_c \approx B^{-\alpha}$	0.28	0.24	0.21	0.26	---

BYTO 2.5 mol.% + BYNO 2.5 mol.% / BYTO 5 mol. %:

F_p behaviour @ low T



BYNTO	77 K	65 K	50 K	30 K	10 K
$F_p(\text{max})$ (GN/m ³)	12	69	210	469	895
$B@F_p(\text{max})$	4	5	6	7	11

BYNTO/BYTO presence of a second high field F_p component emerging at low T

BYNTO has a second high field F_p component which makes it more performing wrt to BYTO at low T

Conclusions 1/2

Mixed doping by $\text{Ba}_2(\text{Y,Ta})\text{O}_6 + \text{Ba}_2(\text{Y,Nb})\text{O}_6$: 2.5 mol.% + 2.5 mol.%

- BYTO single vs mixed BYTO + BYNO doping;

BYNTO exhibits the best J_c in extended T and B ranges

This results from a **synergetic combination** of:

- **density** of columns;
- **size** of columns;
- **continuity**;
- **splay**;
- **Y_2O_3 nanoparticles** decorating BYNTO columns;
- **CuO intergrowth** density;

key factors (by comparison with BYTO)

This landscape provide an **effective contribution to vortex pinning at low $T < 30$ K**

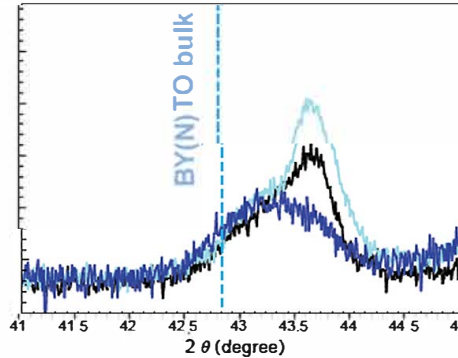
Can the defect landscape be tuned?

Mixed doping BYTO 2.5% + BYNO 2.5%: analysis of the film growth rate

Film growth rate (ρ) tuned in the range $\rho \approx 0.02 - 1.8 \text{ nm s}^{-1}$ by:

- laser repetition rate;
- laser wavelength;

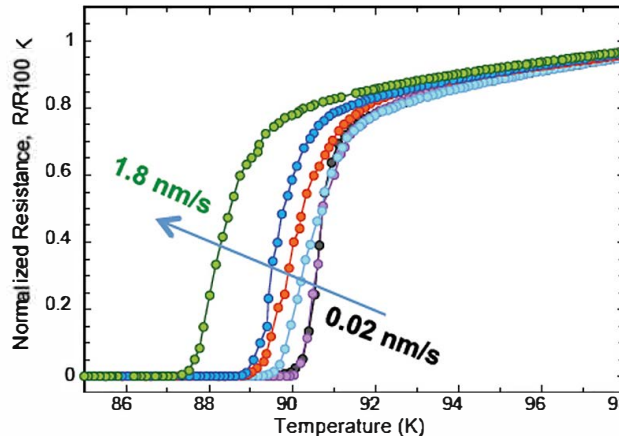
rate per pulse @248 nm $\approx 3 \times$ @308 nm



Growth rate (nm/s)	0.02	0.3	1.4
$c_{\text{YBCO}} (\text{Å} \pm 0.007)$	11.696	11.692	11.714
$a_{\text{BY(N)TO}} (\text{Å} \pm 0.01)$	8.31	8.30	8.36
FWHM (005)	0.12	0.13	0.13

PLD setup
XeCl/KrF Excimer Laser
 $\lambda = 308 \text{ nm}$
 $\lambda = 248 \text{ nm}$
 $f_L = 1 - 15 \text{ Hz}$
fluence $\approx 1.5 - 2 \text{ J/cm}^2$

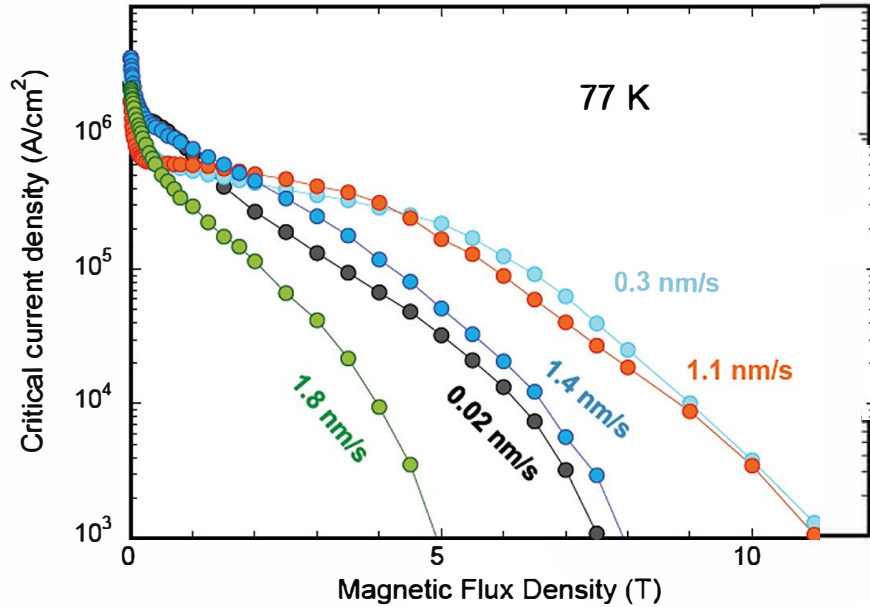
Film thickness $\approx 200 \text{ nm}$



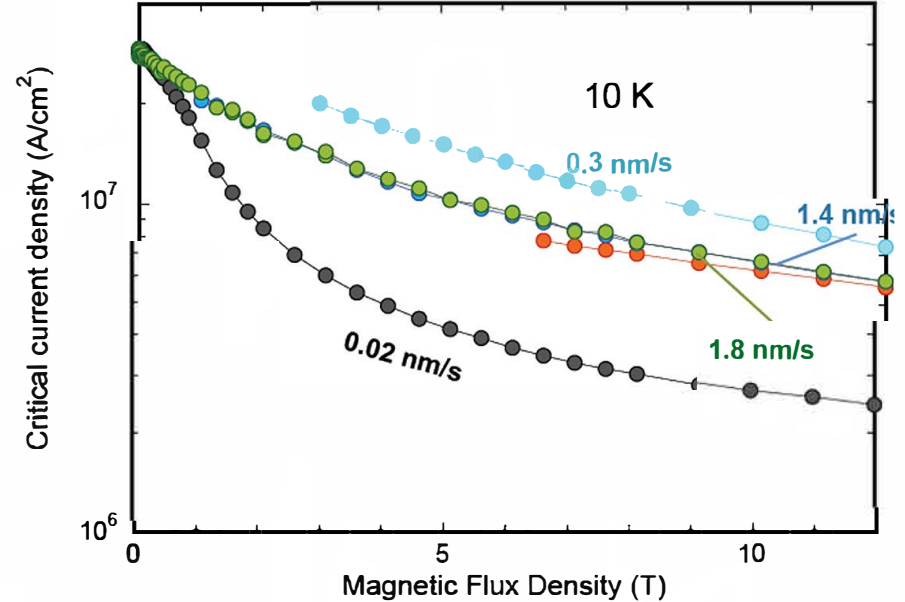
With higher rates:

- higher strain in YBCO;
- change in BYNTO;
- lower T_c ;

more details will be provided by F. Rizzo in his talk, today



@ 77 K, best J_c in field behaviour for rates within 0.3 - 1.1 nm/s with extended plateau and highest H_{irr}



@ 10 K, $J_c(B)$ for high rates very similar (1.4 nm/s and 1.8 nm/s fully overlap!)
 $J_c(12\text{ T}) = 7.4, 5.8, 6.0, 6.0\text{ MA/cm}^2$

Conclusions 2/2

Mixed doping by $\text{Ba}_2(\text{Y,Ta})\text{O}_6 + \text{Ba}_2(\text{Y,Nb})\text{O}_6$: 2.5 mol.% + 2.5 mol.%

- BYTO single vs mixed BYTO + BYNO doping;

BYNT0 exhibits the best in extended angular, T and B ranges

The landscape provides an **effective contribution to vortex pinning at low $T < 30$ K**







- Defects landscape tunability by growth rate;

Very low rate (0.02 nm/s ←): *continuous columns with reduced density and increased diameter + Y_2O_3 nanoparticles*

high rate (→ 1.8 nm/s): *ab-plane platelets + c-axis rods*

crossing 0.3 - 1.1 nm/s

high performances in the whole T-range

Growth rates (nm/s)	High T	low T
low rates		
intermediate (0.3-1.1)		
high rates		

Thank you for your attention

