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**Ca-repaired  $\text{BaZrO}_3$  nanorods/ $\text{YBa}_2\text{Cu}_3\text{O}_{7-x}$  interface for enhanced pinning in  $\text{YBa}_2\text{Cu}_3\text{O}_{7-x}$  nanocomposites with 2-8%  $\text{BaZrO}_3$  doping**

**Judy Wu**

**University of Kansas, USA**

[jwu@ku.edu](mailto:jwu@ku.edu)

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# Acknowledgement

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**Mohan Panth, Victor Ogunjimi, Aafiya, Bibek Gautam, Shihong Chen and Jack Shi**

**University of Kansas, USA**

**Mary Ann Sebastian and Timothy J. Haugan**

**U.S. Air Force Research Laboratory, US**

**Di Zhang and Haiyan Wang**

**Purdue University, USA**



# Outline

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## **Effect of strain on APC/REBCO nanocomposites**

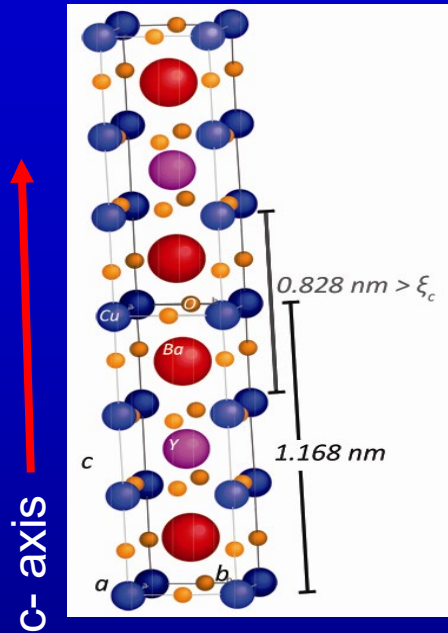
- **Role of strain field in controlling APC morphology, dimension, orientation, etc**
- **Effect of APC/HTS interface on APC pinning efficiency**

## **Development of multilayer (ML) approach to dynamically adjust APC/HTS interface strain**

- **Improving APC/YBCO interface**
- **Enhancing APC pinning efficiency**
- **Large  $I_c$  in ML films**

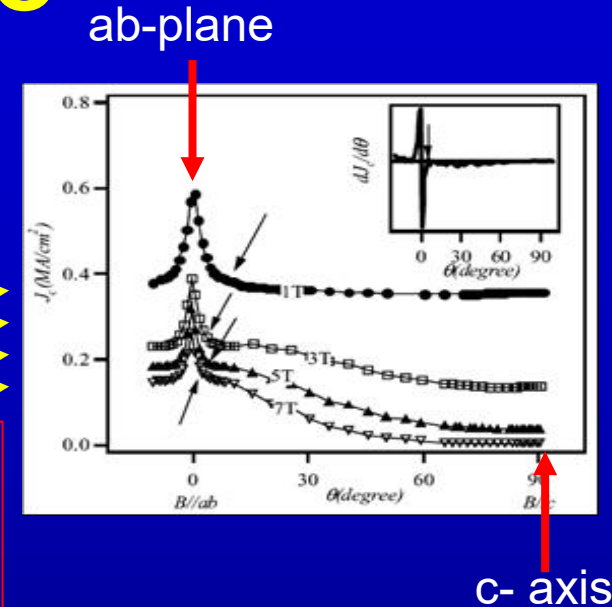
## **Summary and future perspectives of APC/HTS nanocomposites for applications**

# APCs needed in REBCO



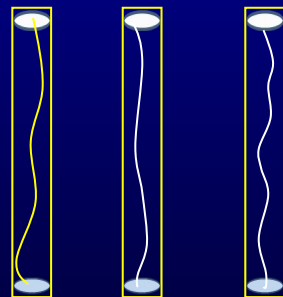
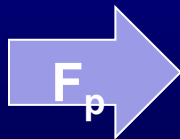
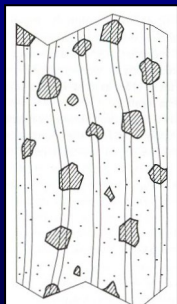
Layered structure, relatively short  $\xi_c$  of YBCO intrinsically provide pinning for  $H \parallel a-b$

With high  $F_p/L$ , columnar defects provide a desirable geometry to address weak pinning at  $H \parallel c$ -axis

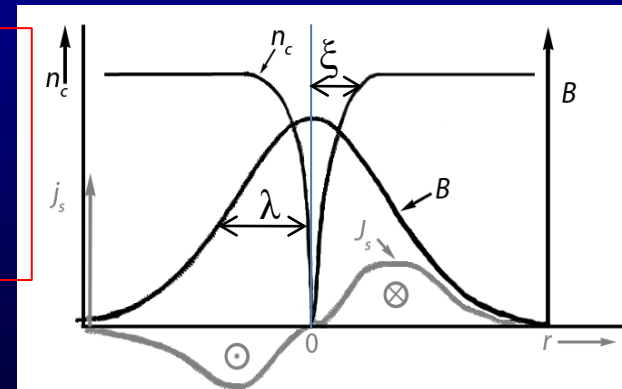


$\xi_{ab} \sim$  a few nm in YBCO determines the APC dimension

$$H_{c1} < H < H_{c2}$$



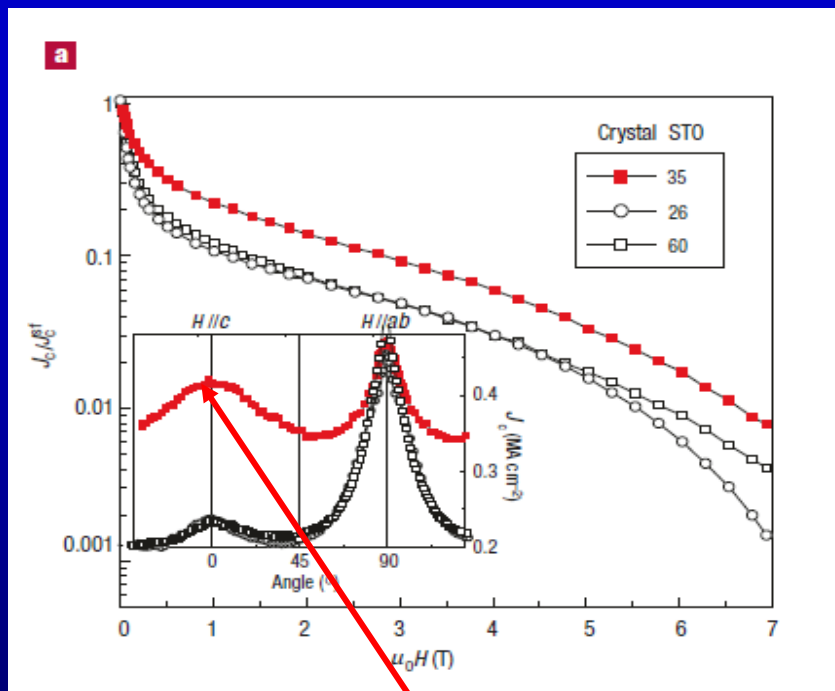
$F_p \propto$  sharpness of the pinning potential energy across the APC/HTS interface



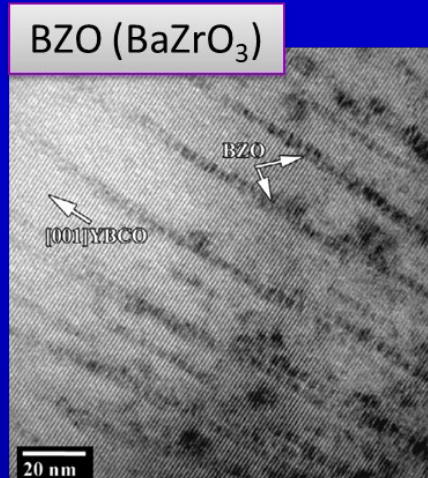
R.P. Huebener, *Magnetic Flux Structures in Superconductors*. Springer (2001); W. Buckel, R. Kleiner, *Superconductivity*, Wiley-VCH, (2004); Blatter G, Feigel'man M V, Geshkenbein V B, Larkin A I and Vinokur V M 1994 Vortices in high-temperaturesuperconductors Rev. Mod. Phys. 66 1125

# BaZrO<sub>3</sub> (BZO) 1D APCs

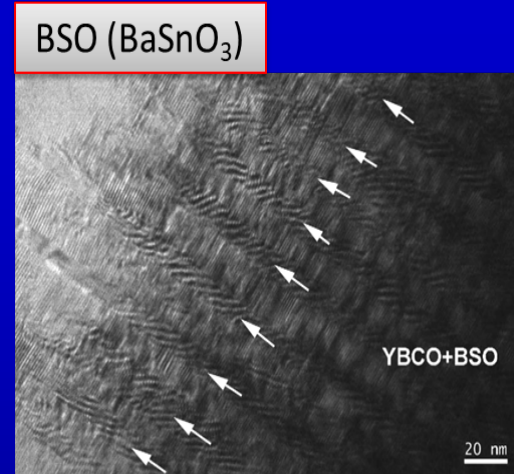
# Other 1D APCs



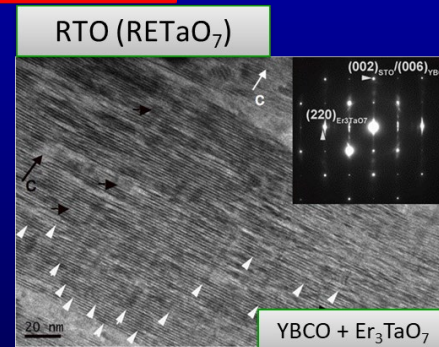
J. MacManus-Driscoll et al, *Nat. Mat.* **3**, 439 (2004)



A. Goyal, et al. *Supercond Sci Technol.* **18**, 1553 (2005)



C. V. Varanasi, et al., *Appl Phys Lett* **93**, 092501 (2008).

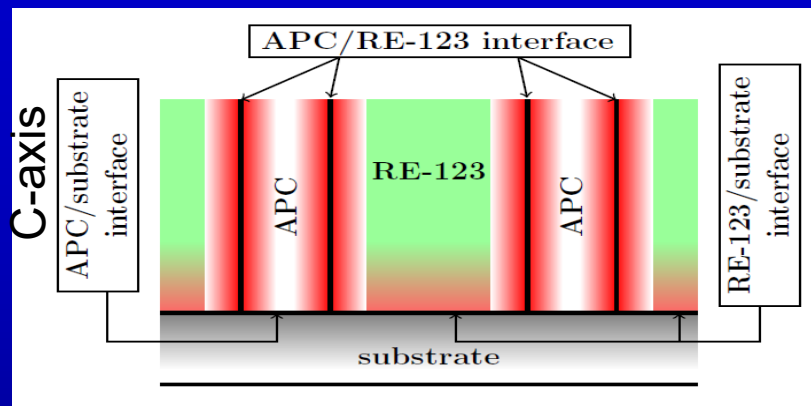


S. A. Harrington, et al. *Supercond Sci Technol.* **22**, 022001 (2009).

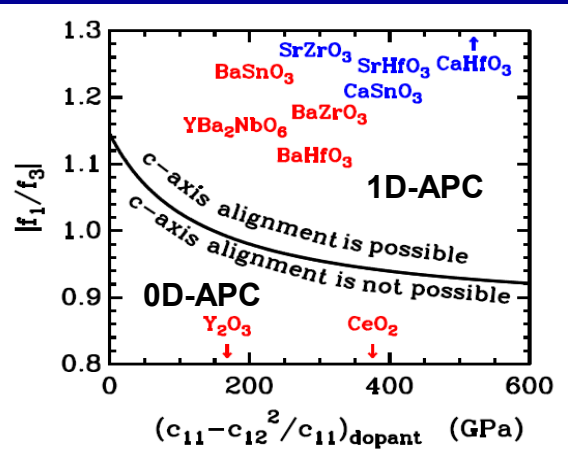
- C-axis aligned BZO 1D APCs provide strong correlated pinning shown as a  $J_c$  peak at H//c-axis
- Accommodation field  $H^* \sim n^* \Phi_0$  could be estimated from 1D APC areal density  $n^*$

# Strain field initiated from 1D APC/RE-123 interface plays a critical role in self-assembly of the 1D-APCs

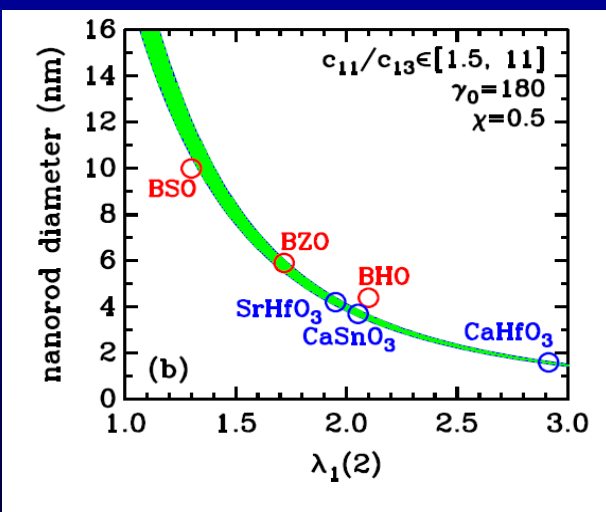
Elastic strain energy model



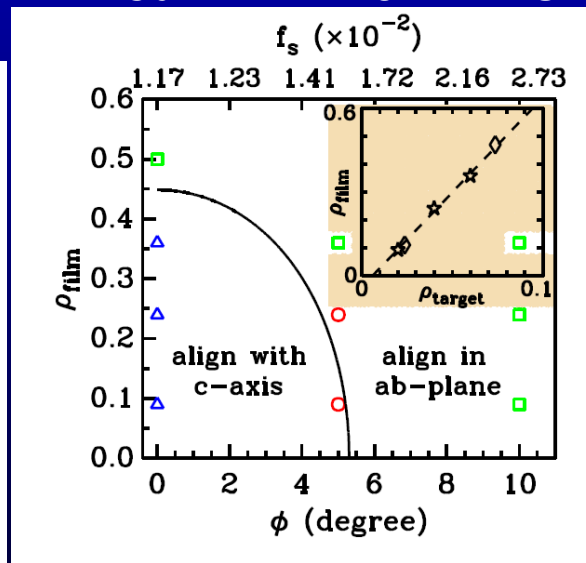
## APC material selection



## APC dimension



## Mixed 1D+2D+3DAPCs

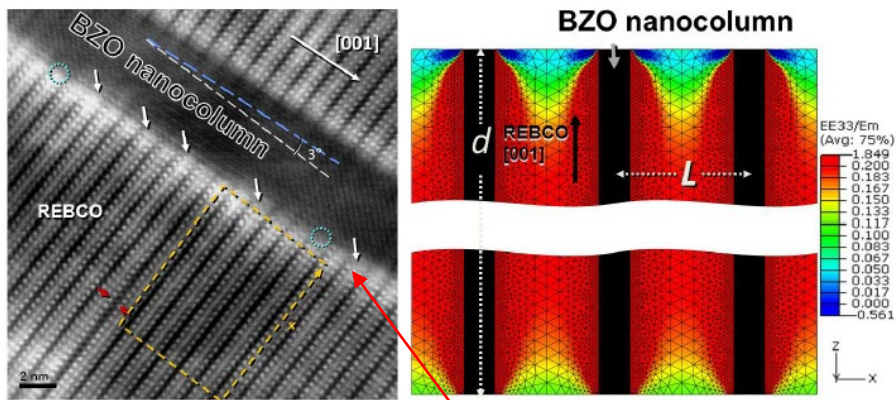


- Shi and Wu, *Philosophic Magazine* **92**, 2911 (2012); **92**, 4205 (2012);
- Wu and Shi, *SUST* **30**, 103002 (2017) in *SUST Special Issue on Artificial Pinning Centers*
- Wu, Gautam and Ogunjimi, in *Superconductivity*, ed. by Kosmas Prassides, Chiara Tarantini, Anna Palau, Petre Badica, Alok K Jha, Tamio Endo and Paulo Mele, Springer (2020). Page 29-52.

# The bad news of the strain field

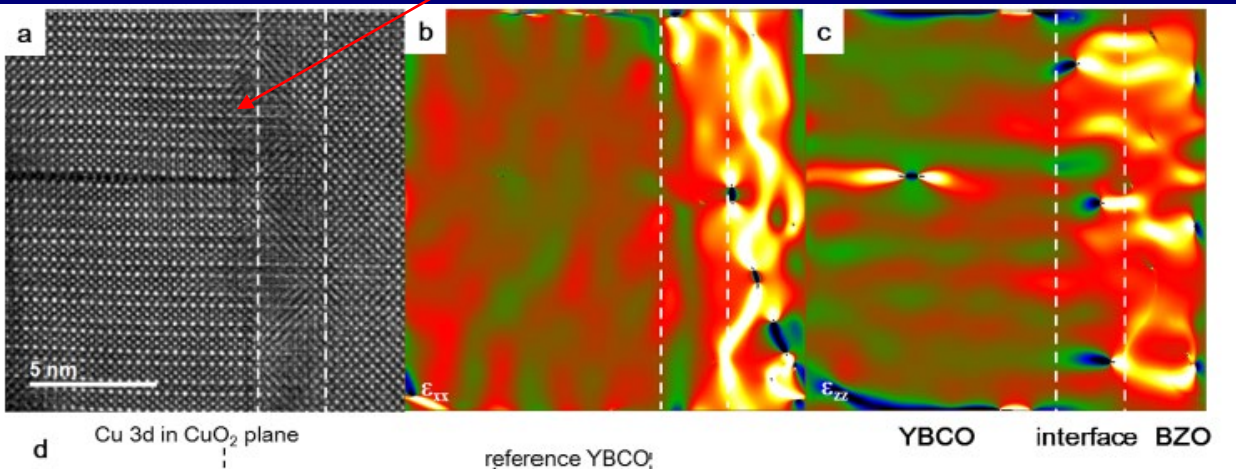
Strain field initiated from 1D APC/RE-123 interface due to a large BZO/YBCO lattice mismatch of  $\sim 7.7\%$

C. Cantoni et al. *ACS Nano* 6, 4783 (2011).



BZO 1D-APC/YBCO interface is defective

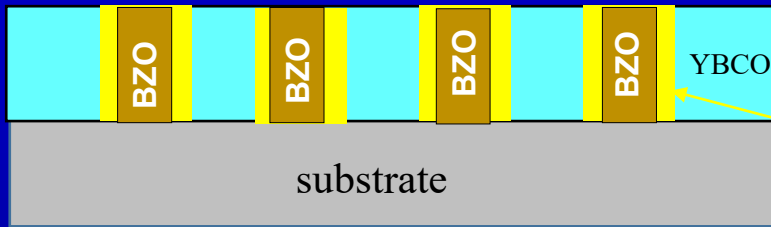
- A defective, oxygen-deficient YBCO column around the BZO/YBCO interface
- This raises a question on the impact on the pinning efficiency of BZO 1D-APCs



T. Horide et al. *ACS Nano* 11, 1780 (2017).

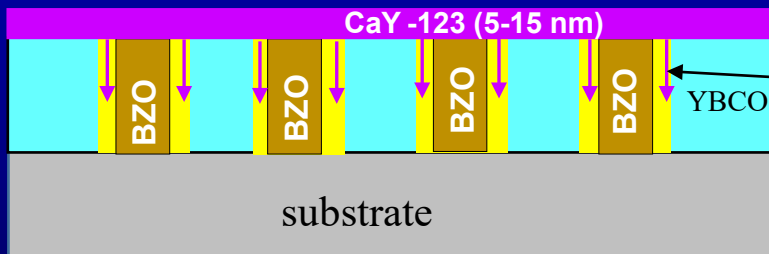
# Multilayer approach: dynamic control of the BZO/YBCO interface

## Step 1: deposition of BZO/YBCO



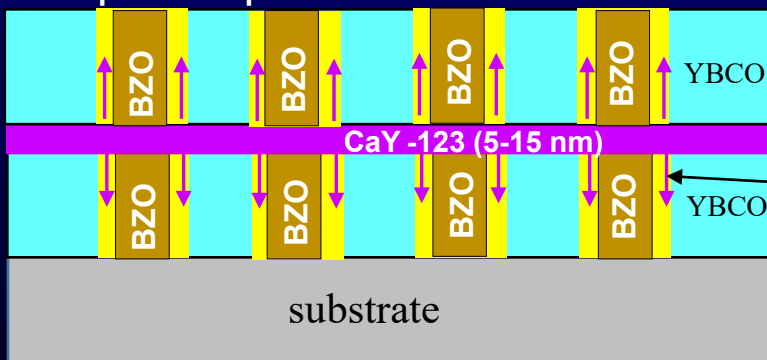
- BZO 1D-APCs form similarly to the case single-layer BZO/YBCO nanocomposite films
- **Tensile strained** BZO/YBCO interface due to 7.7% lattice mismatch

## Step 2: deposition of $\text{Ca}_{0.3}\text{Y}_{0.7}\text{Ba}_2\text{Cu}_3\text{O}_7$



Cu substitution by 30% larger Ca ions favored energetically by the tensile strained BZO/YBCO interface, leading to enlarged YBCO c-axis and reduced lattice mismatch at the BZO/YBCO interface

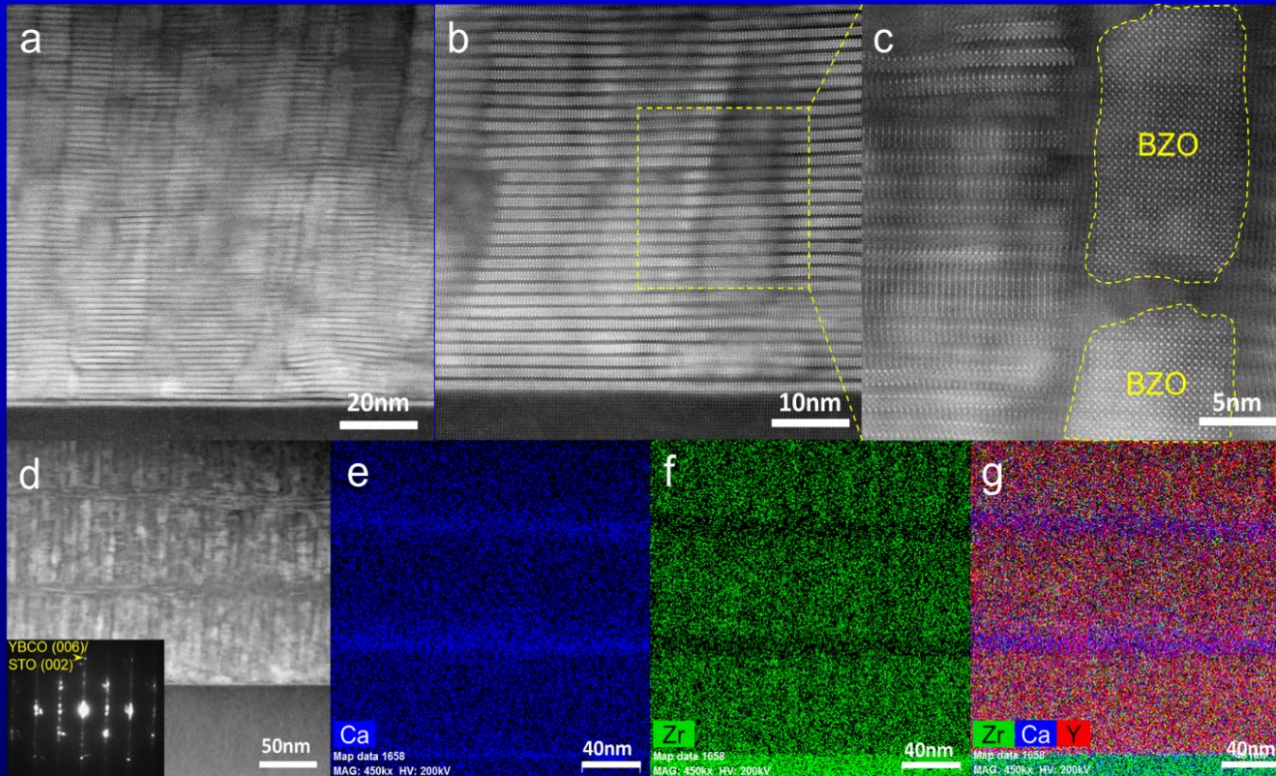
## Step 3: deposition of BZO/YBCO



- 2-8% of BZO/YBCO with layer thickness of 50-250 nm tested for Ca diffusion
- The ML deposition can be repeated for thick films



# STEM/EDS characterization

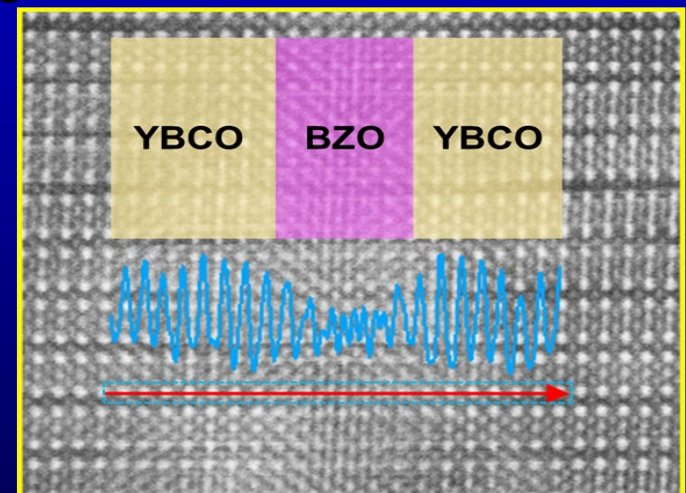
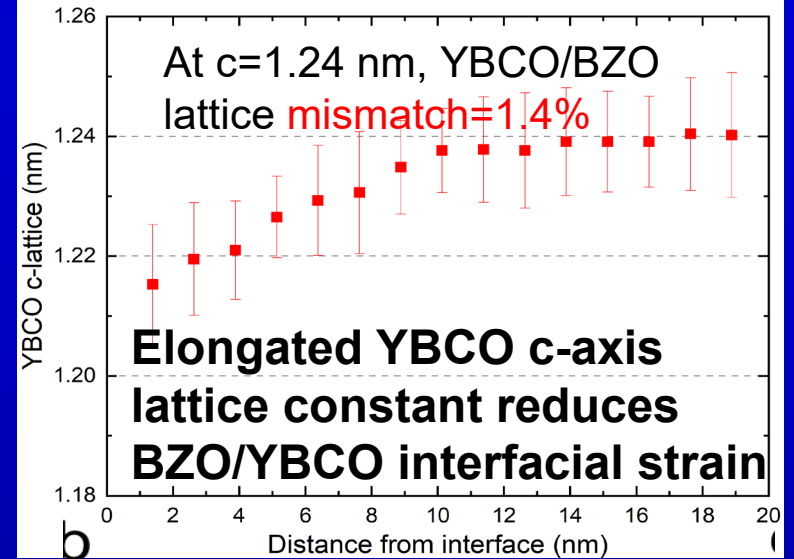
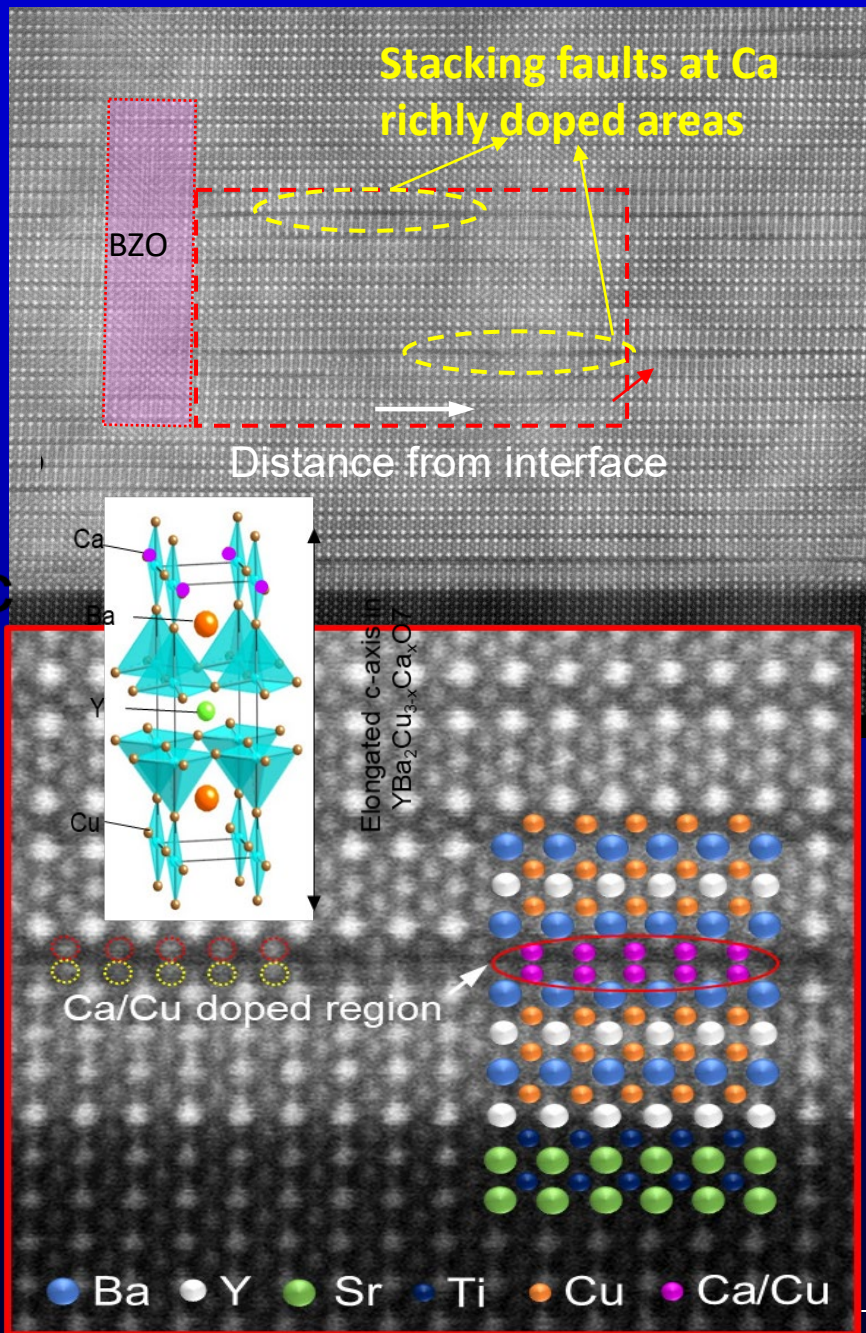


6% BZO/YBCO  
single-layer (SL)  
nanocomposite film

6% BZO/YBCO  
multilayer (ML)  
nanocomposite film

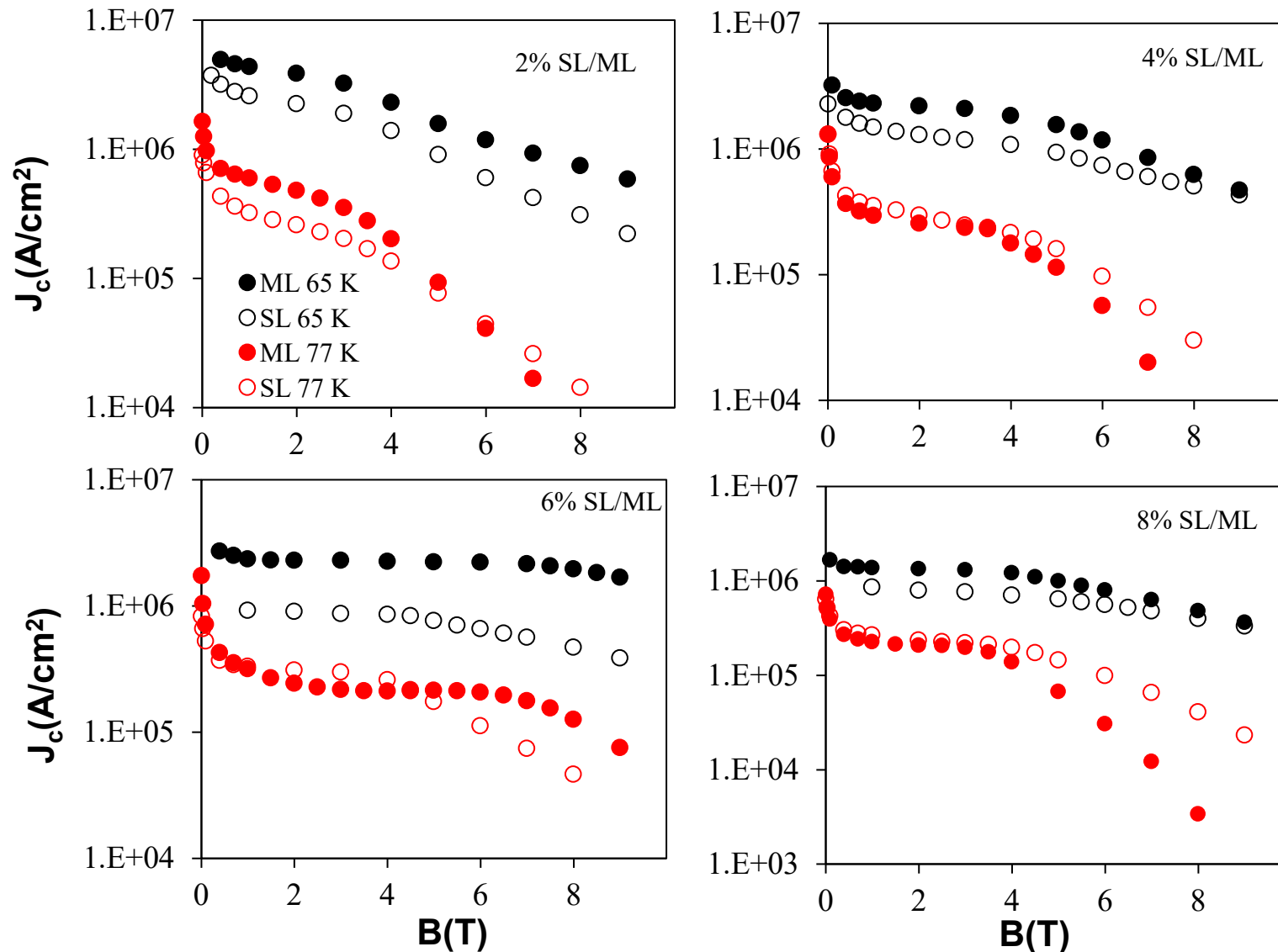
Ogunjimi et al, SUST 34,  
104002 (2021); Wu, et al,  
SUST 35, 034001 (2022)

- BZO 1D-APCs have comparable diameters and areal concentrations in SL and ML samples
- Minor Ca diffusion from CaY-123 spacers to YBCO during PLD growth is clearly visible from STEM/EDS elemental maps



**Ca/Cu replacement on YBCO's Cu-O planes leads to stacking faults formation and hence the c-lattice elongation, which prevents formation of defects on BZO/YBCO interface**

# $J_c$ in 2-8% BZO/YBCO ML and SL films



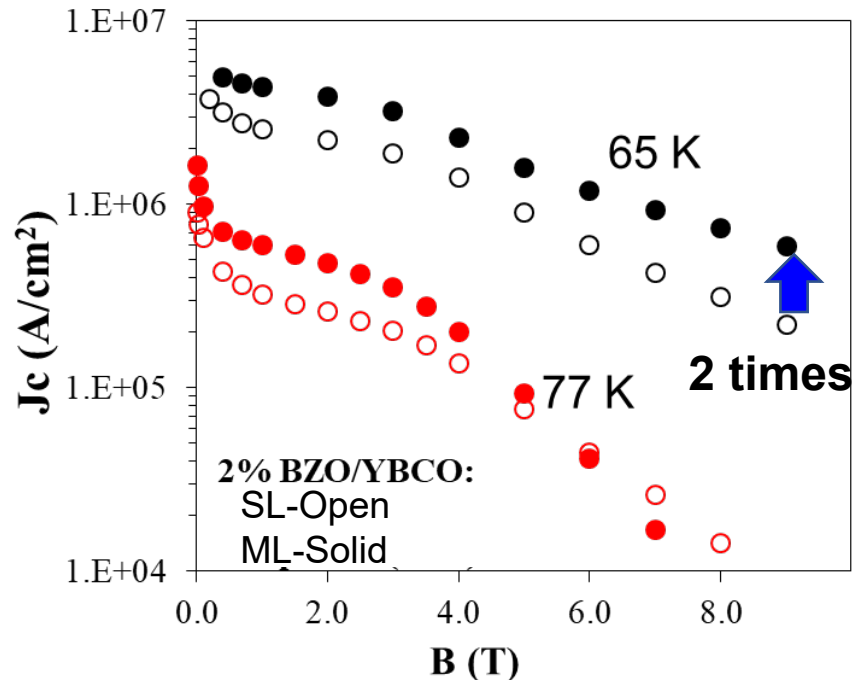
**Improved  $J_c$  has been observed in all ML films doped with 2-8% BZO doping**

Panth et al, IEEE Trans. Appl. Supercond. **32** P1-8 (2022)

## 2% BZO/YBCO

$T_c \sim 88.5$  K (SL)

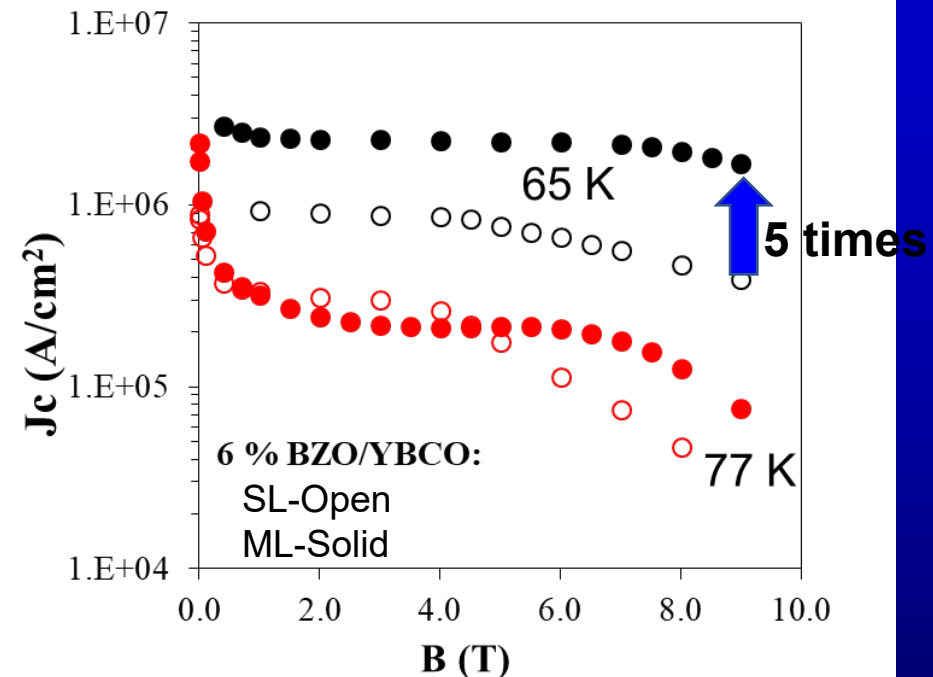
$T_c \sim 87.5$  K (ML)



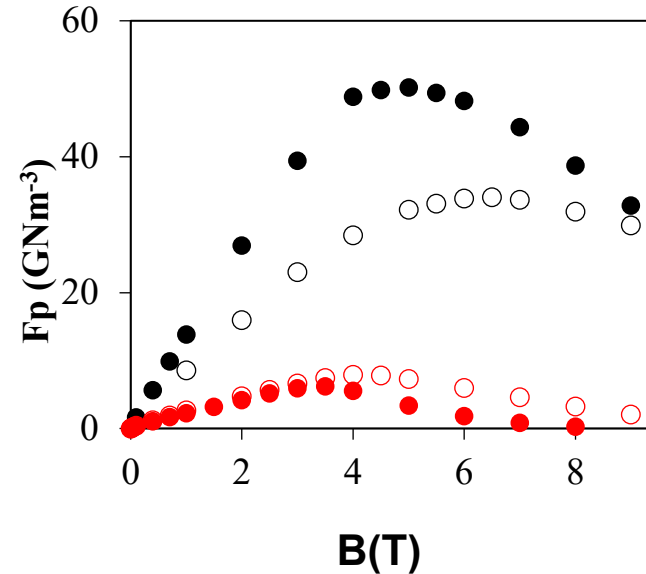
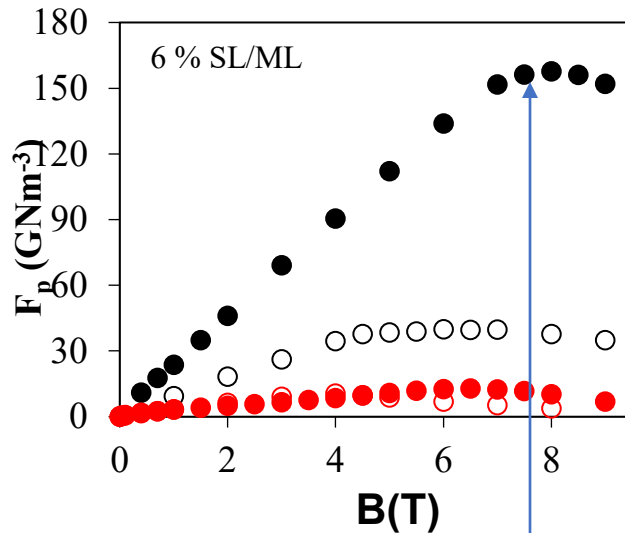
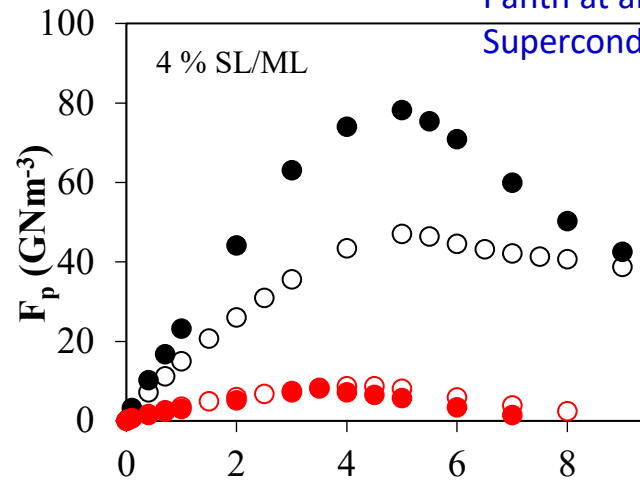
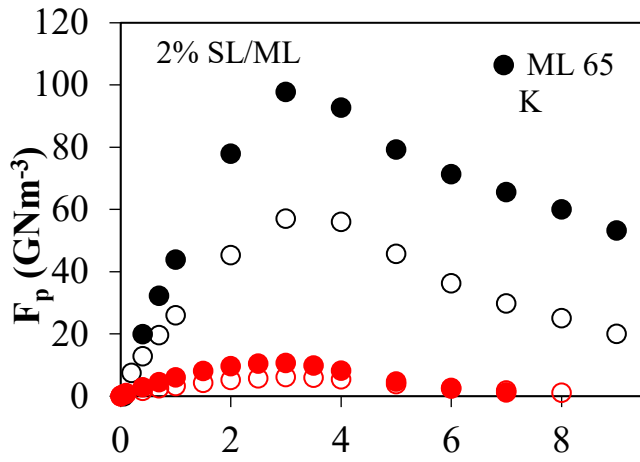
## 6% BZO/YBCO

$T_c \sim 86.9$  K (SL)

$T_c \sim 84.5$  K (ML)

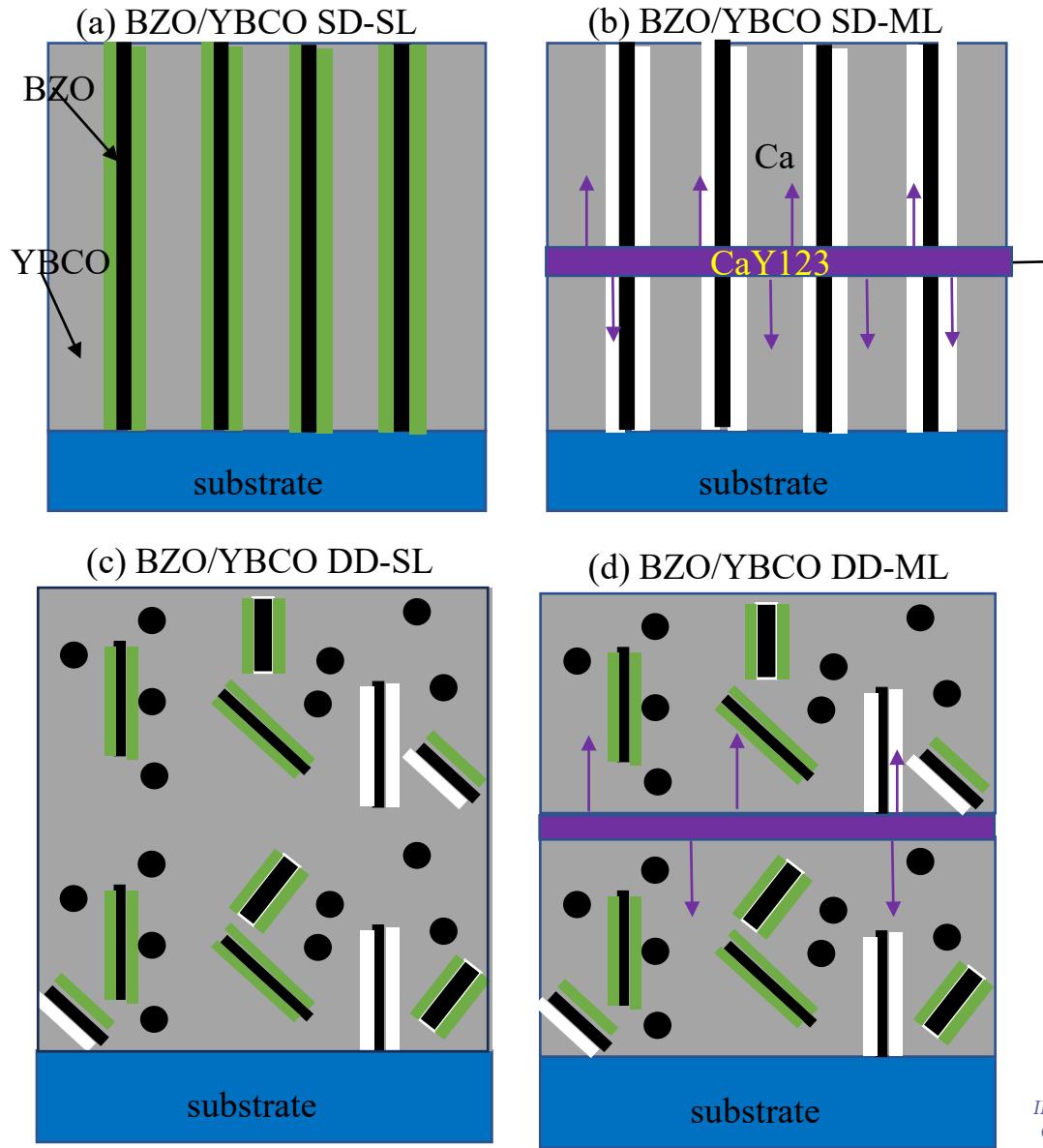


- Despite slightly lower  $T_c$  values, enhanced  $J_c$  (B) was observed in ML samples with a coherent BZO/YBCO interface
- At 65 K,  $J_c$  is enhanced over the entire B field range up to 9.0 T



- At 65 K, the peak  $F_{pmax} \sim 157$  GN/m<sup>3</sup> in 6% ML BZO/YBCO is 4.4 times of that in the SL 6% BZO/YBCO sample
- $B_{max}$  is increased by 60% to 8.0 T in 6% ML BZO/YBCO, but there is still a room for further improvement considering  $B^* \sim 9.2$  T

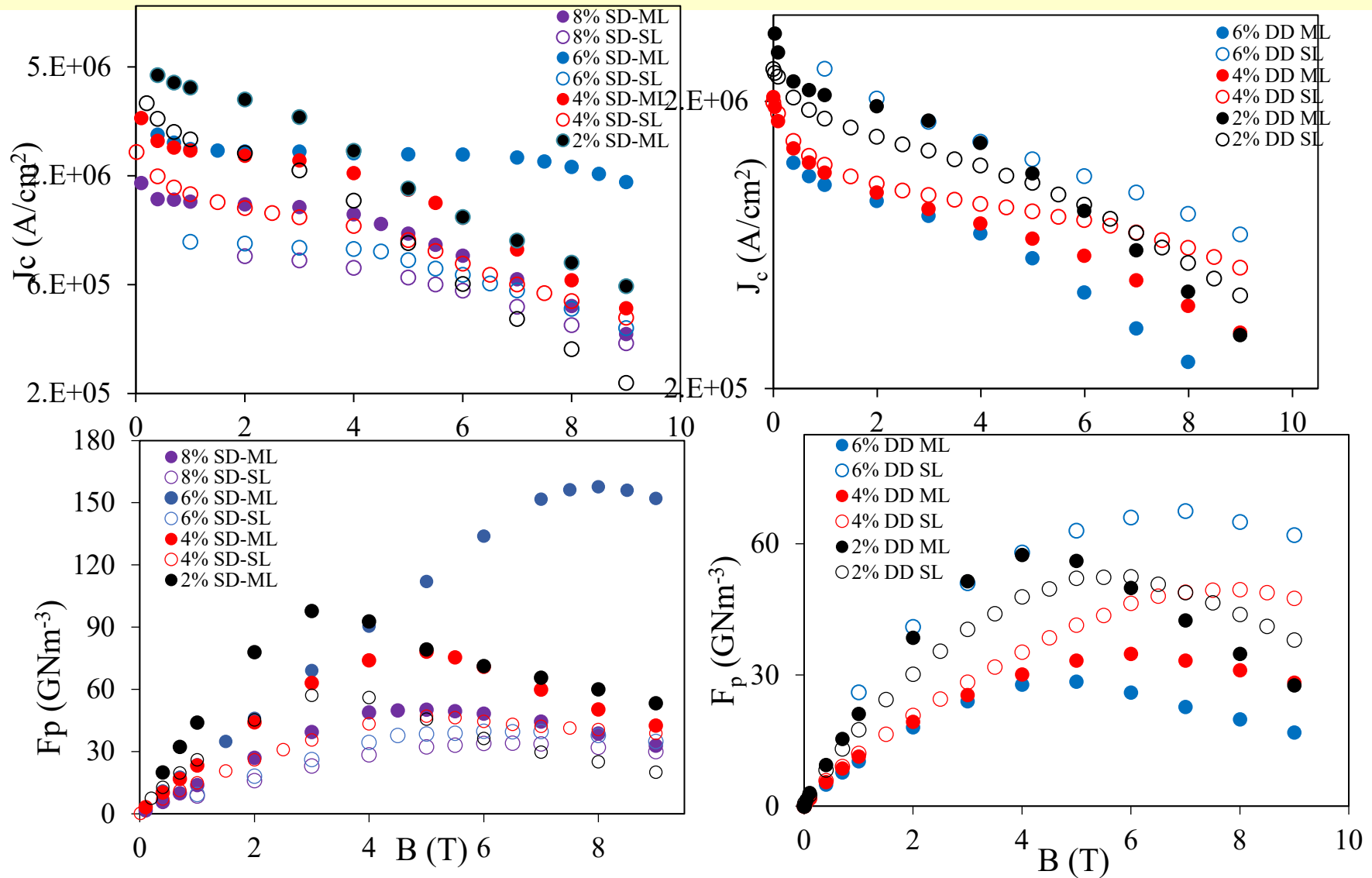
# Effect of strain field on the Ca ion diffusion in $\text{BaZrO}_3\text{-Y}_2\text{O}_3$ doped $\text{YBa}_2\text{Cu}_3\text{O}_{7-x}/\text{Ca}_{0.3}\text{Y}_{0.7}\text{Ba}_2\text{Cu}_3\text{O}_{7-x}$ multilayer nanocomposite films

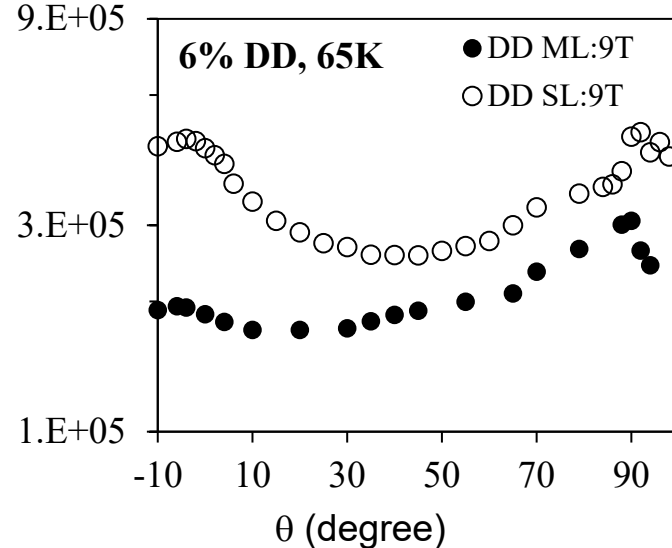
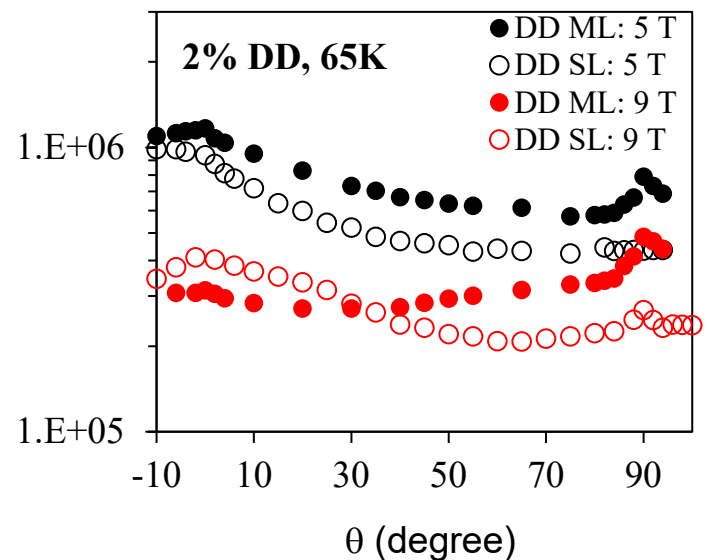
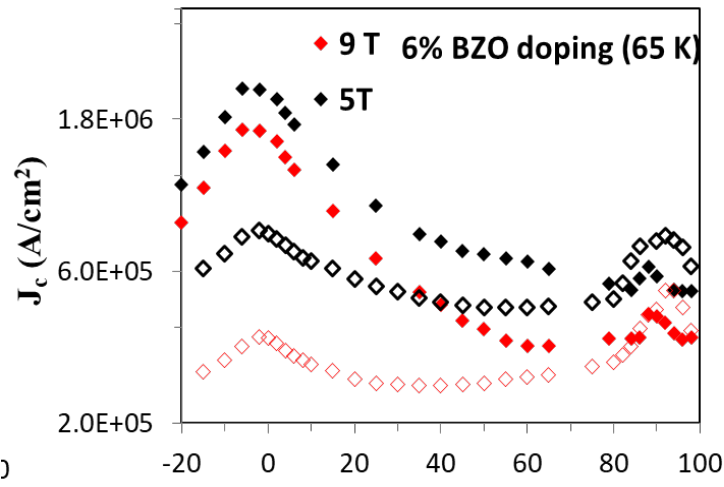
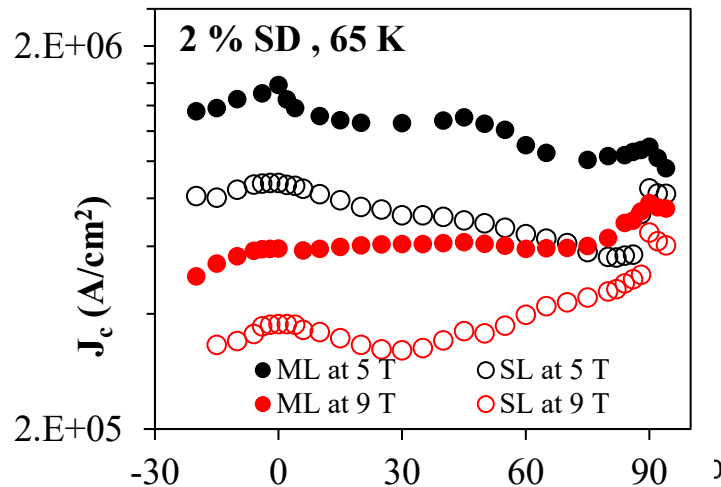


- The tensile strain at the BZO/YBCO interface makes Ca-diffusion along the interface and Ca/Cu substitution energetically favorable
- In DD ML samples with >4% BZO, BZO 1D-APCs become segmented with random orientations, which releases the modulated BZO/YBCO interfacial strain and changes the Ca-diffusion

Panth et al, IOP MRX 10  
P046001 (2023)

# Improved pinning only observed on 2%BaZrO<sub>3</sub>-Y<sub>2</sub>O<sub>3</sub> double-doped ML films, confirming the effect of strain on Ca-difusión and APC pinning





**Enhanced pinning efficiency of the BZO 1D APCs in BZO/YBCO ML samples also extends to a broad angular range up to 85 degree with respect to the B orientations especially at high B fields**



# Summary

- Using a ML method, a coherent BZO 1D APC/YBCO interface can be obtained through Ca/Cu replacement induced stacking faults that leads to reduction of the BZO/YBCO lattice mismatch from 7.7% to 1.4% and prevents defects formation at the BZO/YBCO interface
- Coherent 1D APC/RE-123 interface is the key to achieve high pinning efficiency of 1D-APCs including high  $F_p(H)$ ,  $B_{max}/B^*$ , and isotropic  $J_c(\theta)$  in high B fields
- Significantly enhanced pinning efficiency has been obtained for BZO 1D-APCs in 2-8% BZO/YBCO SD-ML samples. The  $F_{p,max} \sim 157 \text{ GN/m}^3$  (65K, 6vol% BZO doping) is 4.4 times of that of the SL counterpart without the interface repair